

# Considerations for 3D EBSD

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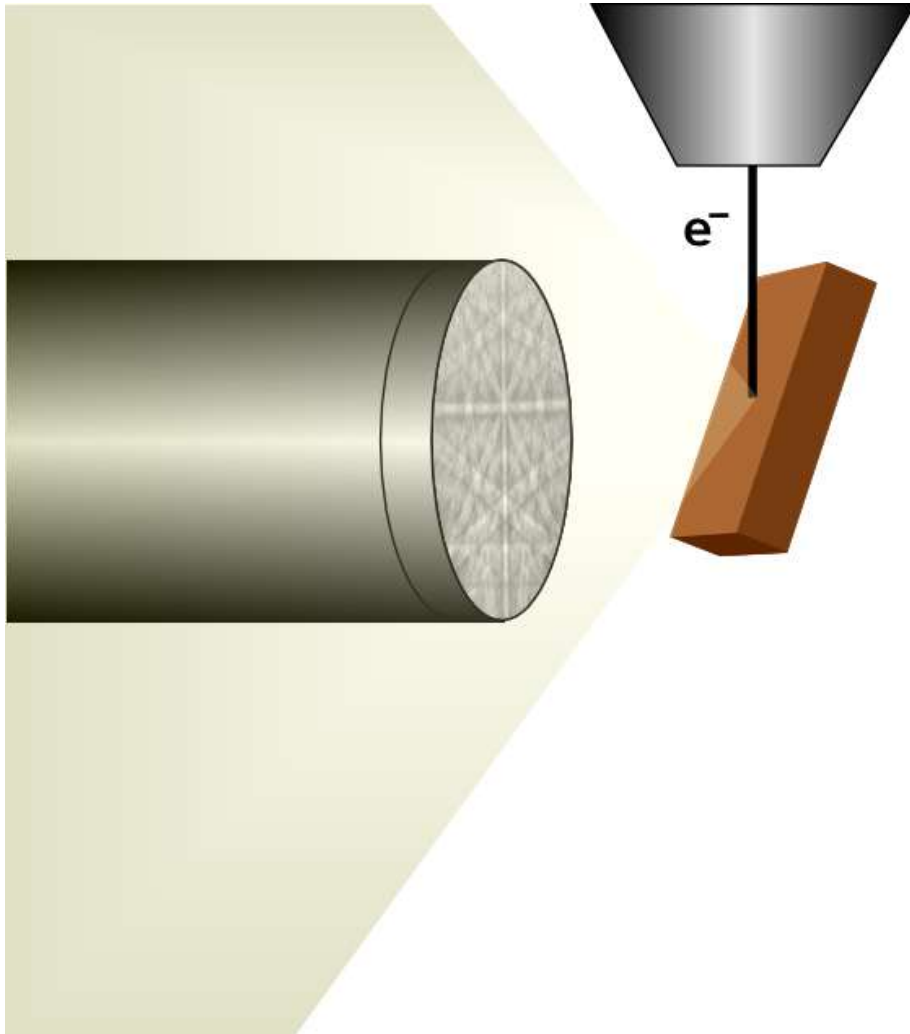
# **EBSD – Probably the Best Measurement in the World**

Austin Day, *Microscopy & Microanalysis*, 11, 502-503 (2005)

# OIM 3D

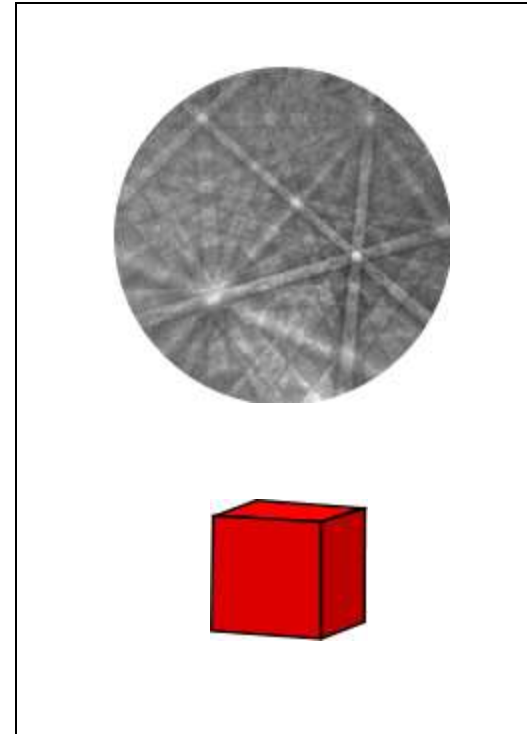
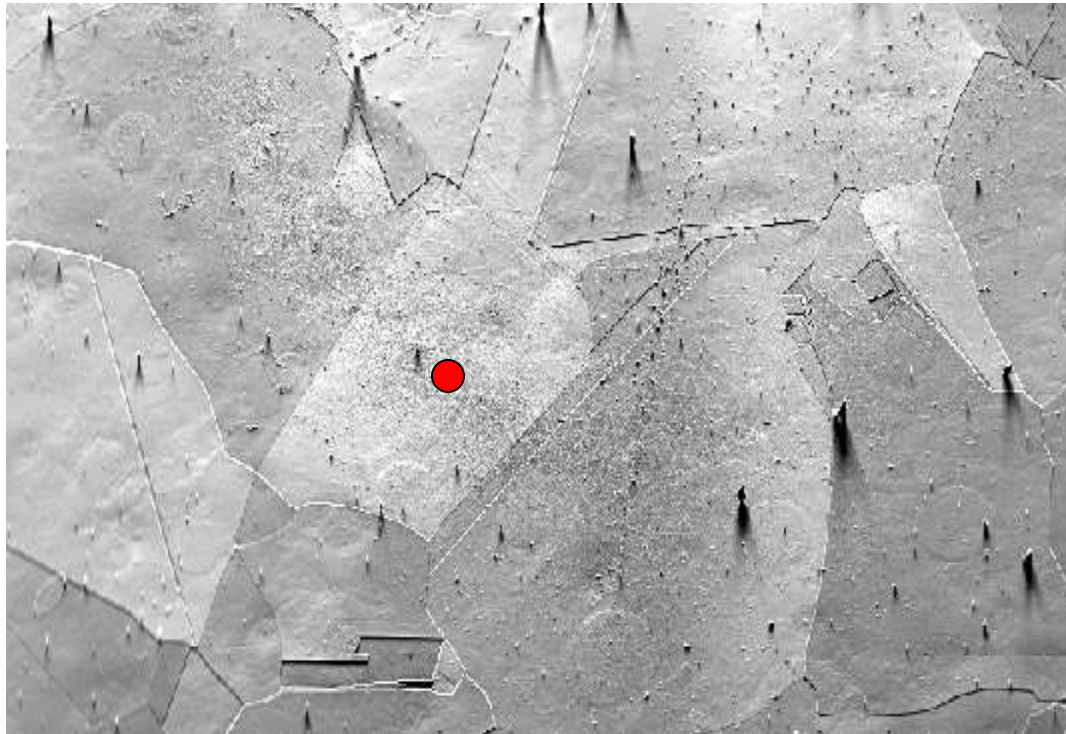
- **Introduction to EBSD**
- **3D Data Acquisition – Serial Sectioning**
- **Practical considerations for EBSD**
- **CdTe PV Case Study**
- **Conclusions**

# EBSD

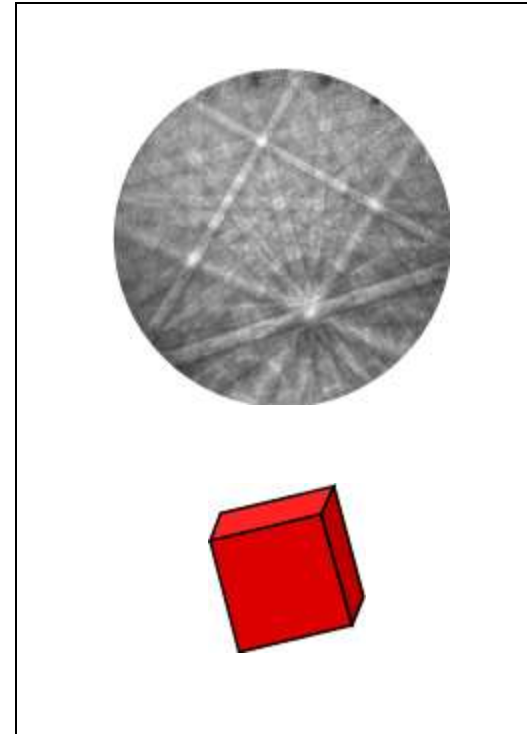
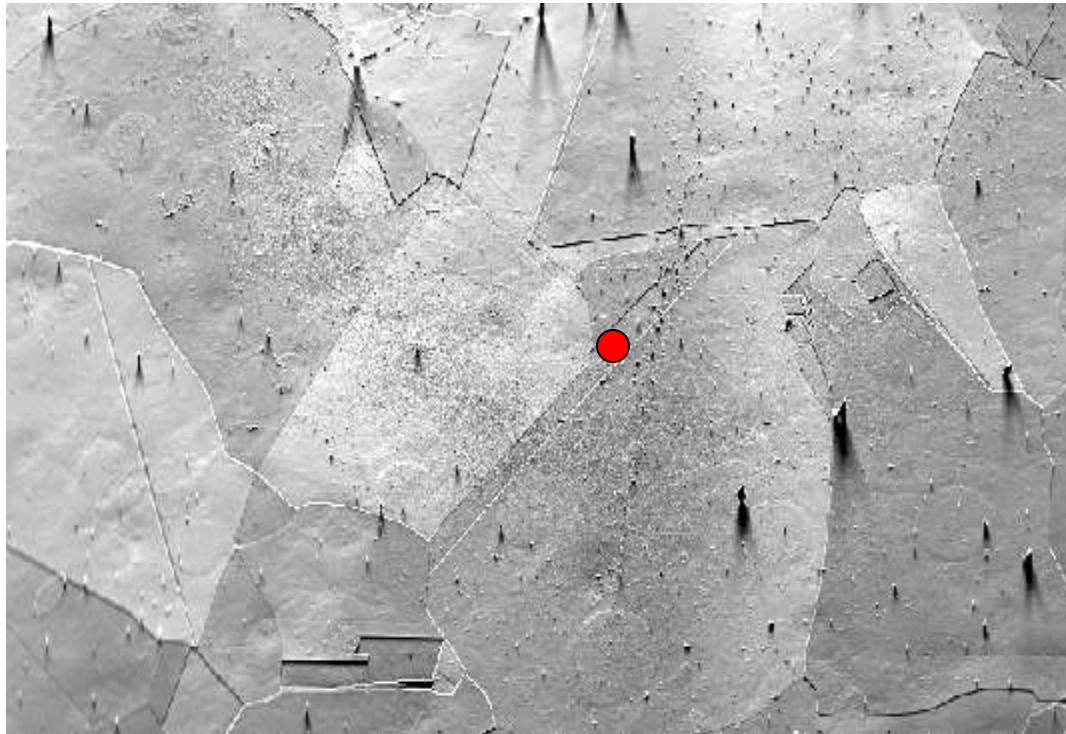


Electron backscatter diffraction patterns (or EBSPs) are obtained in the SEM by focusing a stationary electron beam on a crystalline sample. The sample is tilted to approximately 70 degrees with respect to the horizontal. The diffraction pattern is imaged on a phosphor screen. The image is captured using a low-light CCD camera. The bands in the pattern represent reflecting planes in the diffracting crystal volume. Thus, the geometrical arrangement of the bands is a function of the orientation of the diffracting crystal lattice.

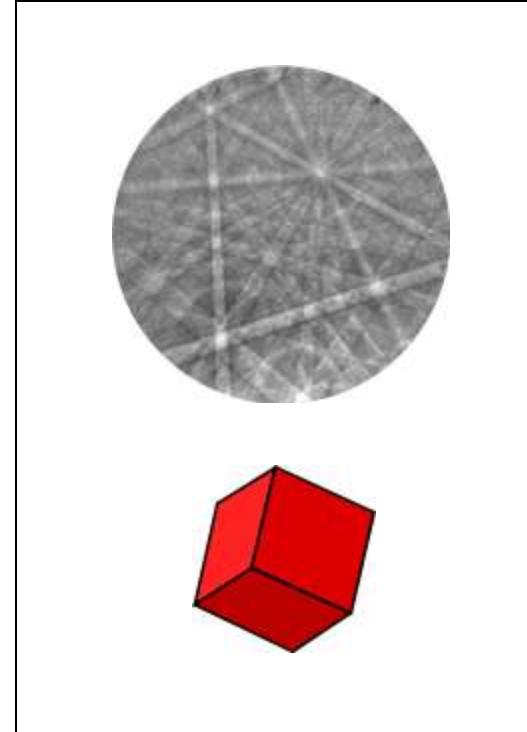
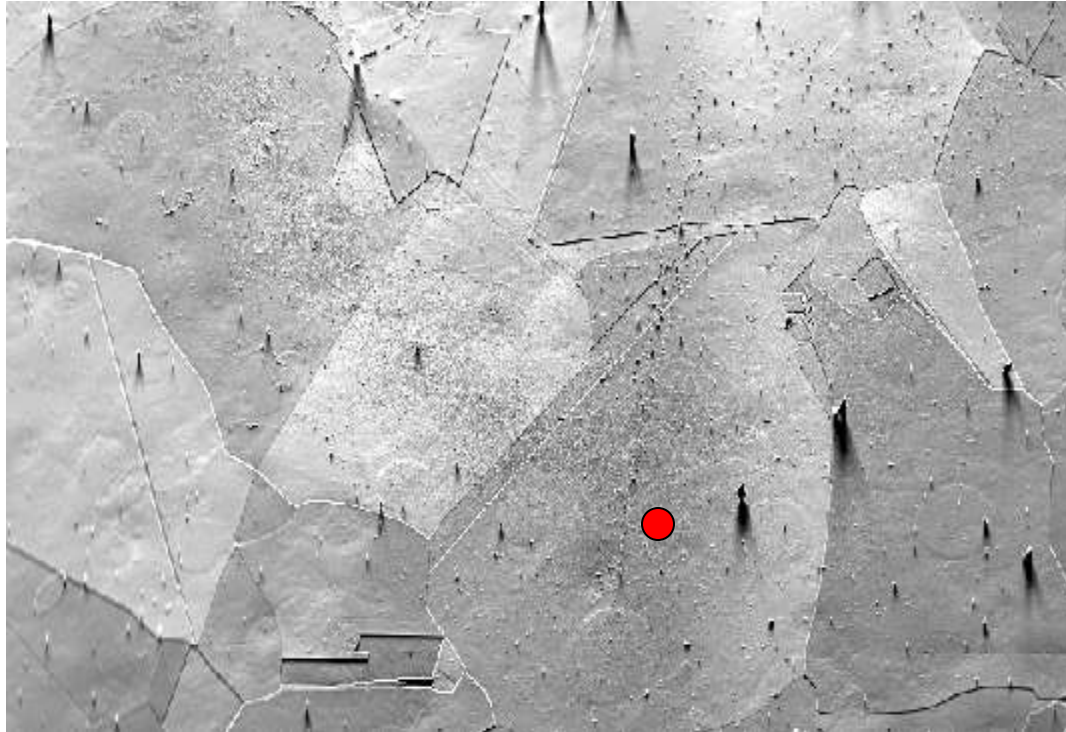
# EBSD



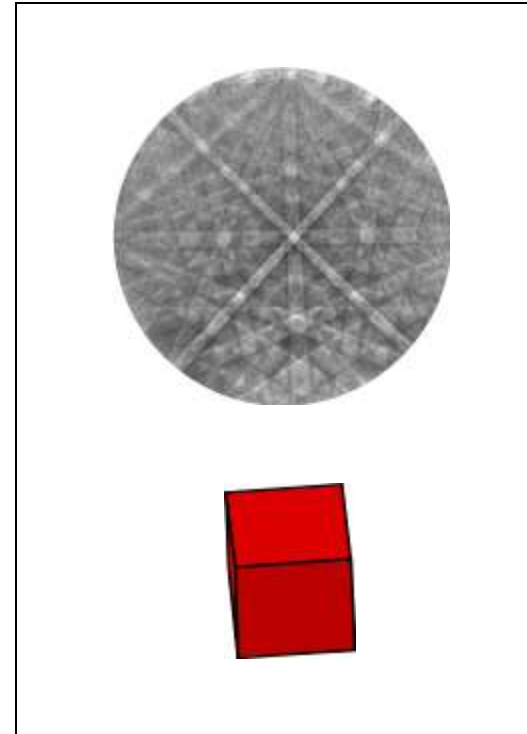
# EBSD



# EBSD

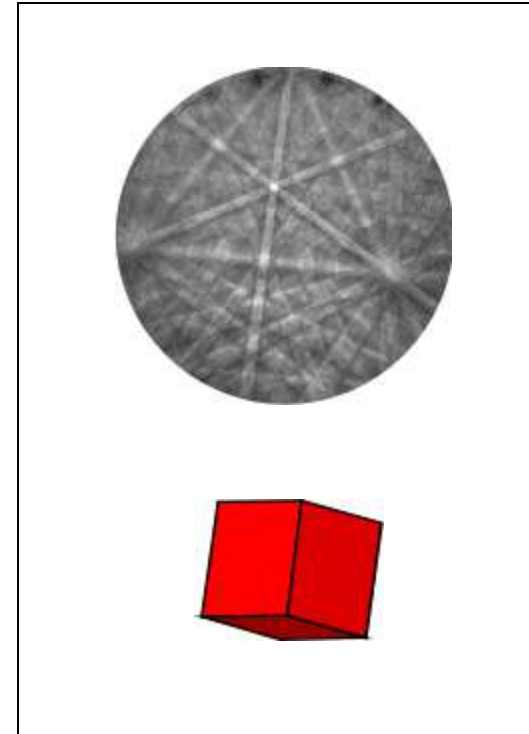
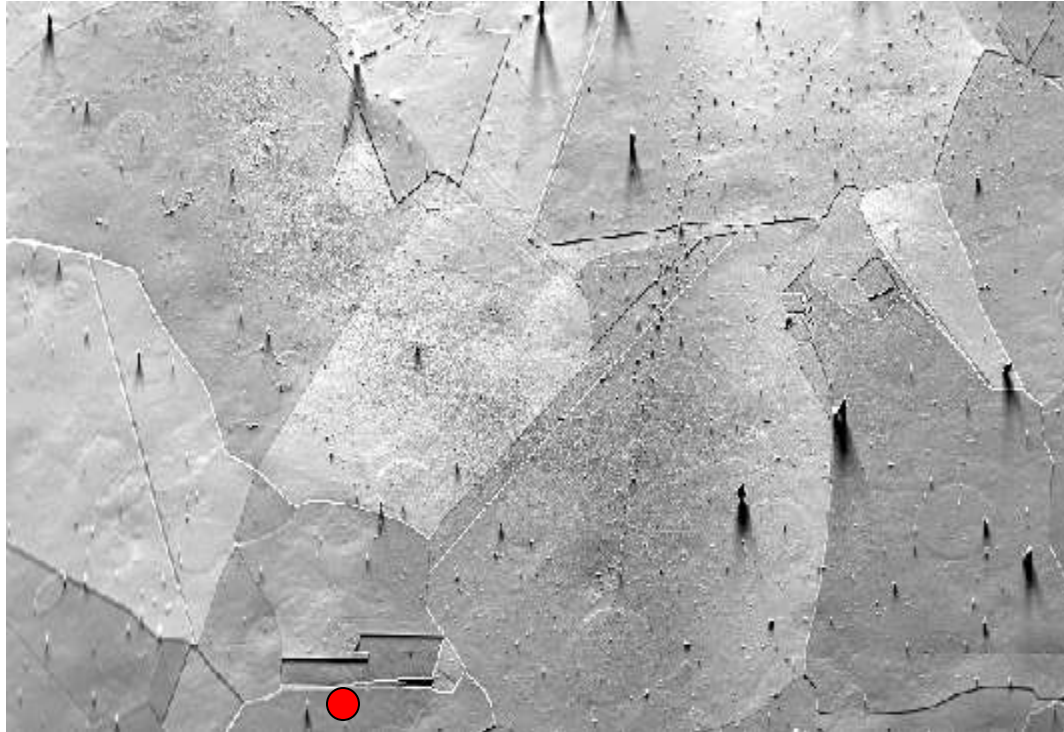


# EBSD

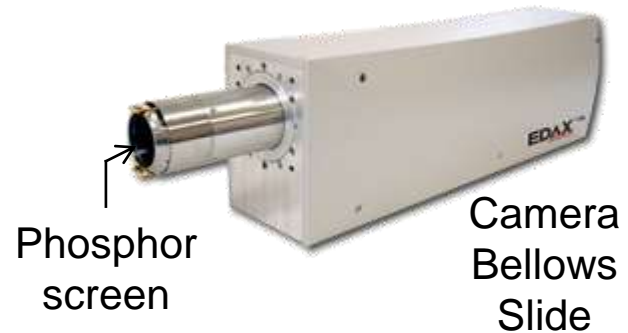




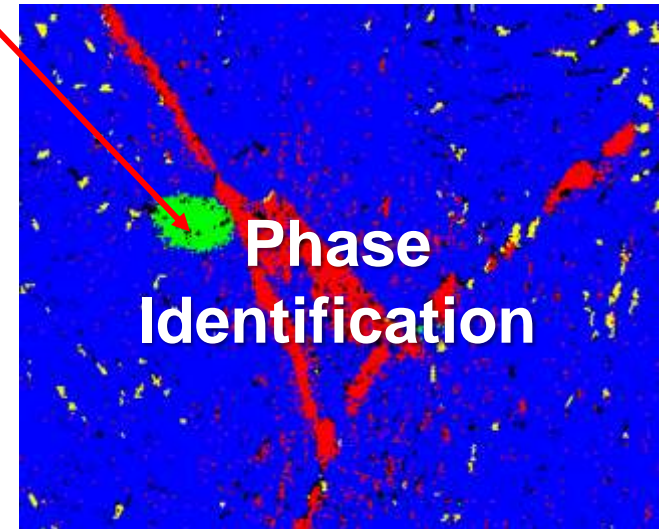
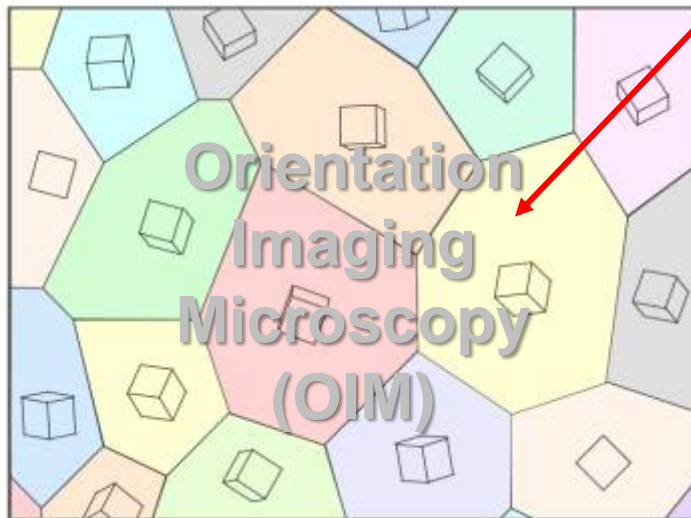
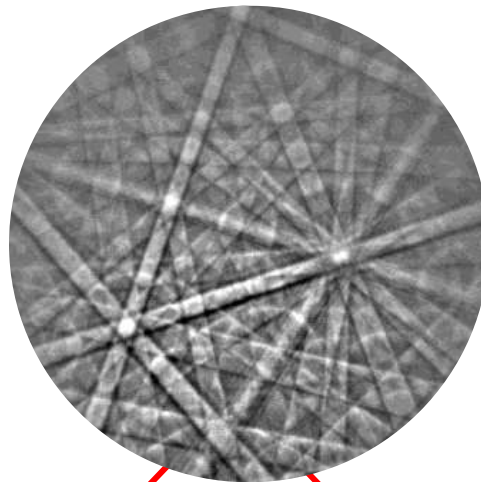
# EBSD



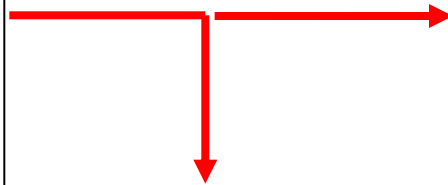
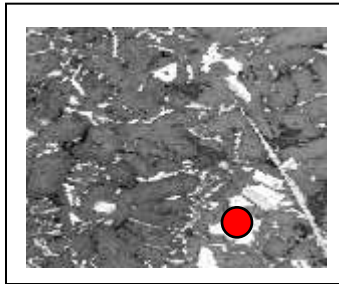
# EBSD - Hardware



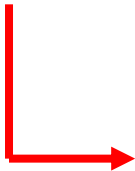
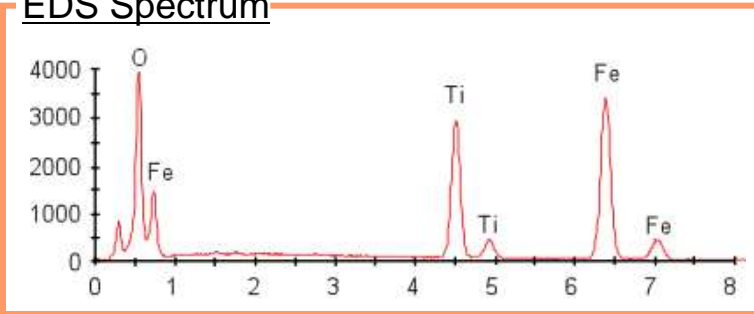
# EBSD – Two Main Application Areas



# EBSD – Phase ID



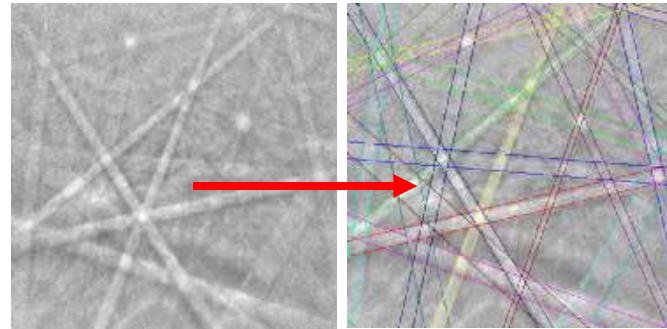
EDS Spectrum



Database



EBSD



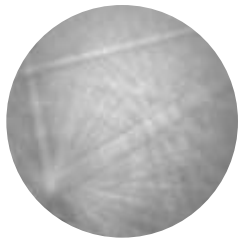
$\text{Fe}_2\text{TiO}_4$



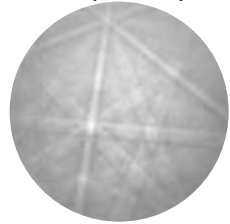
Candidate Phases

$\text{Fe}_2\text{TiO}_4$	Cubic	$a=11.297$
Fe-Ti-O	Cubic	$a=11.31$
$\text{Fe}_4(\text{TiO}_4)_3$	Tetragonal	$a=9.3, c=9.5$
$\text{FeTiO}_3$	Trigonal	$a=5.0884, c=14.093$
$\text{Fe}_2\text{TiO}_4$	Cubic	$a=8.5352$
$\text{Fe}_3\text{Ti}_3\text{O}_{10}$	Orthorhombic	$a=7.789, b=10.008, c=3.74162$
$\text{FeTiO}_3$	Orthorhombic	$a=5.026, b=5.174, c=7.245$
$\text{Fe}_2\text{Ti}_3\text{O}_9$	Hexagonal	$a=2.8667, b=4.5985$

# Phase ID



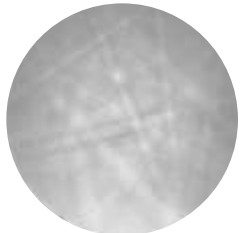
Al(ZrNi)



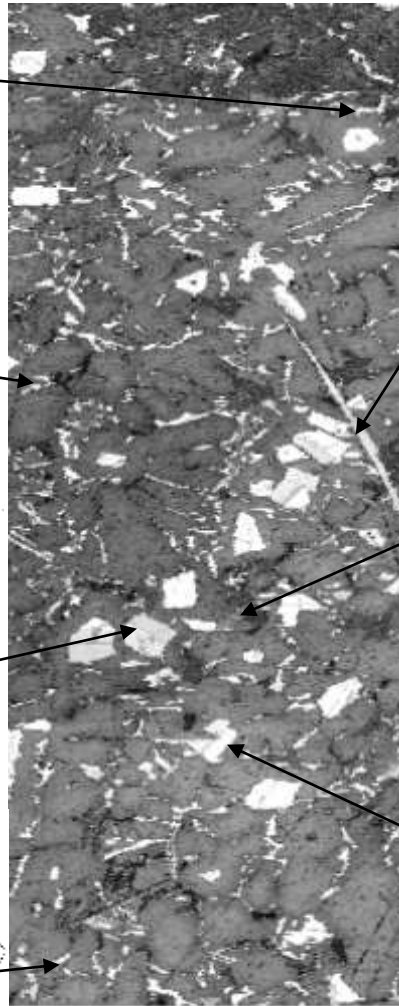
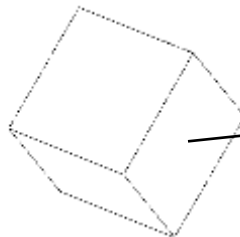
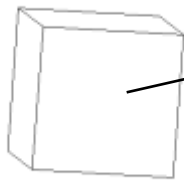
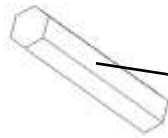
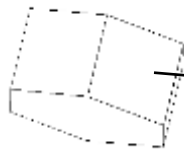
Al<sub>7</sub>Cu<sub>4</sub>Ni



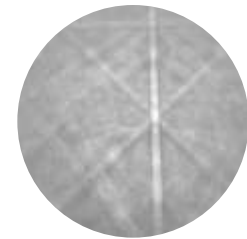
Si



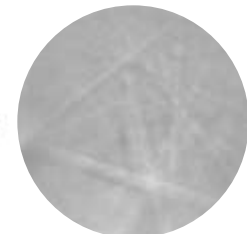
AlP



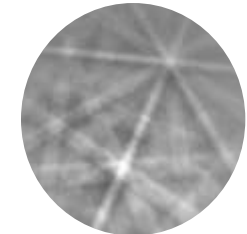
40.00  $\mu\text{m}$  = 40 steps



AlTi

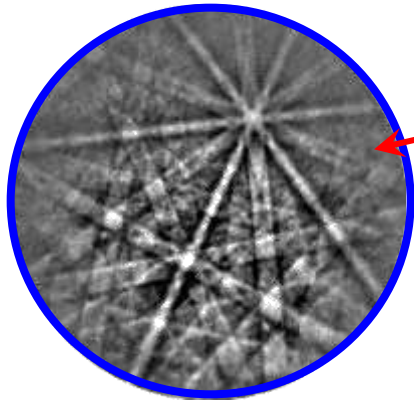


Al<sub>9</sub>(NiFe)

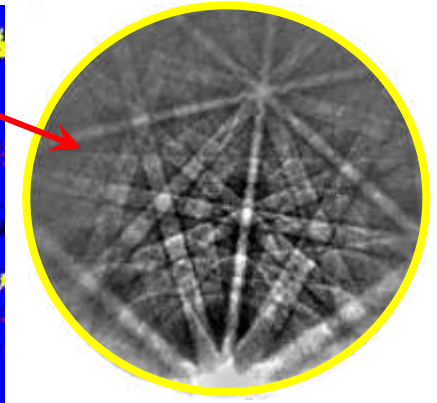
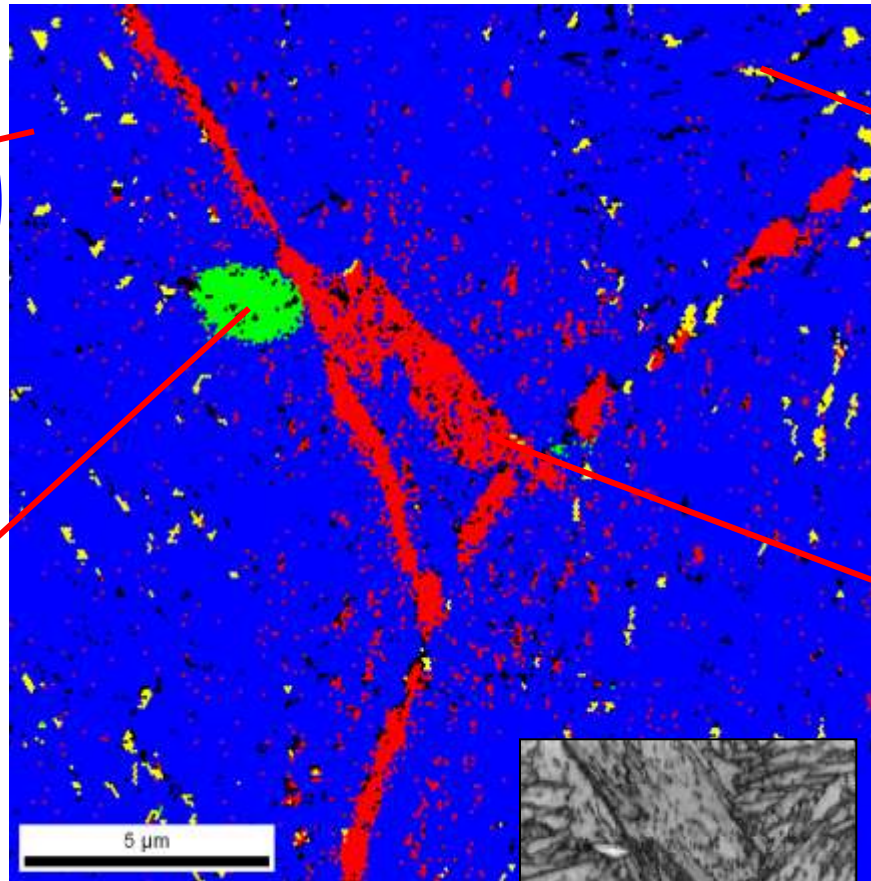


Al

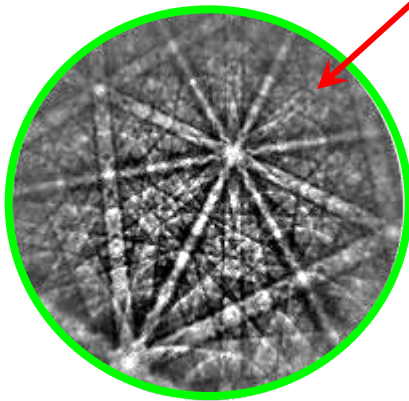
# Phase ID



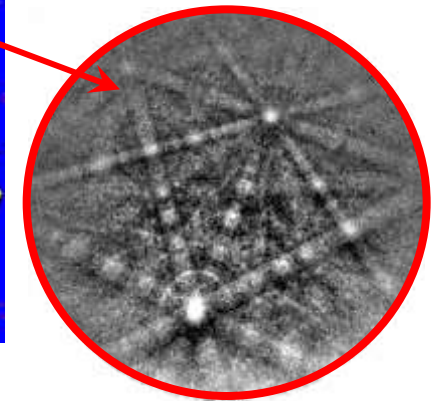
Ferrite



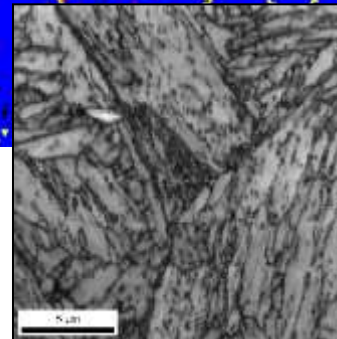
Austenite



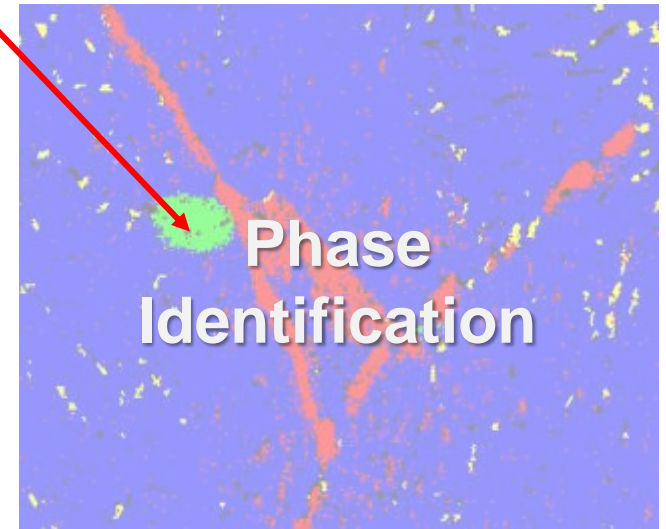
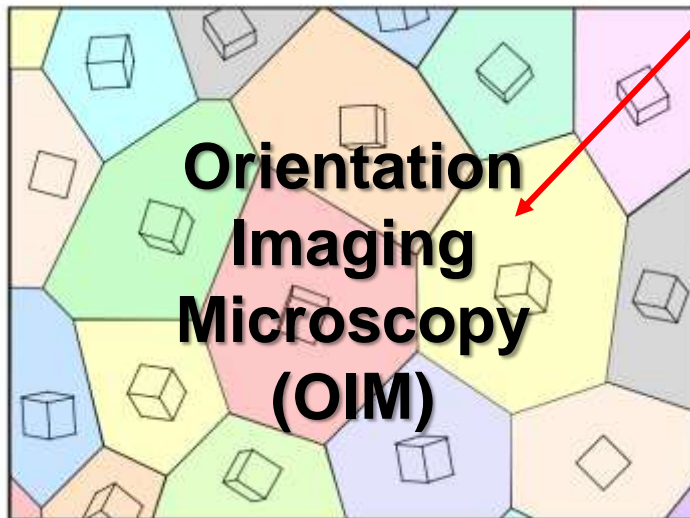
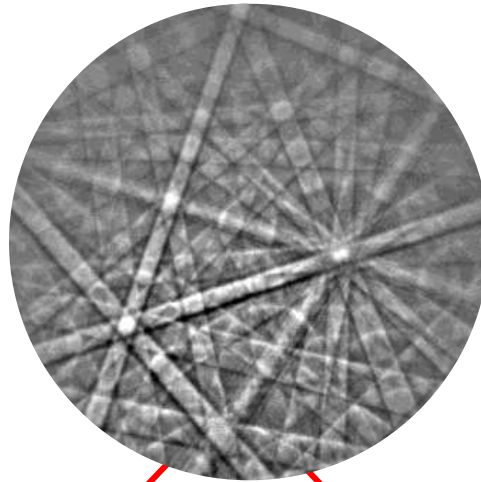
Niobium Carbide



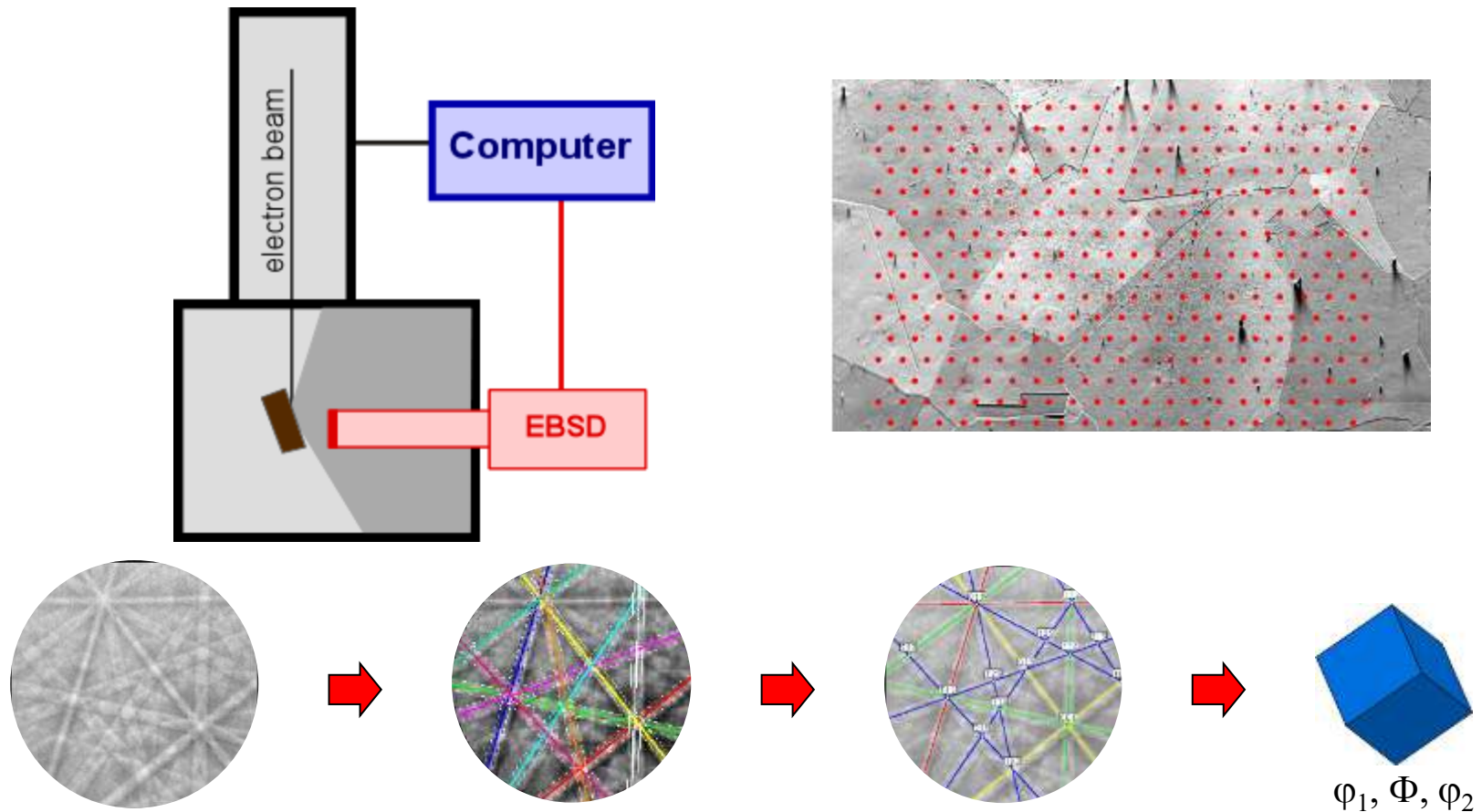
Chromium Carbide



# EBSD – Two Main Application Areas



# Orientation Imaging Microscopy - OIM

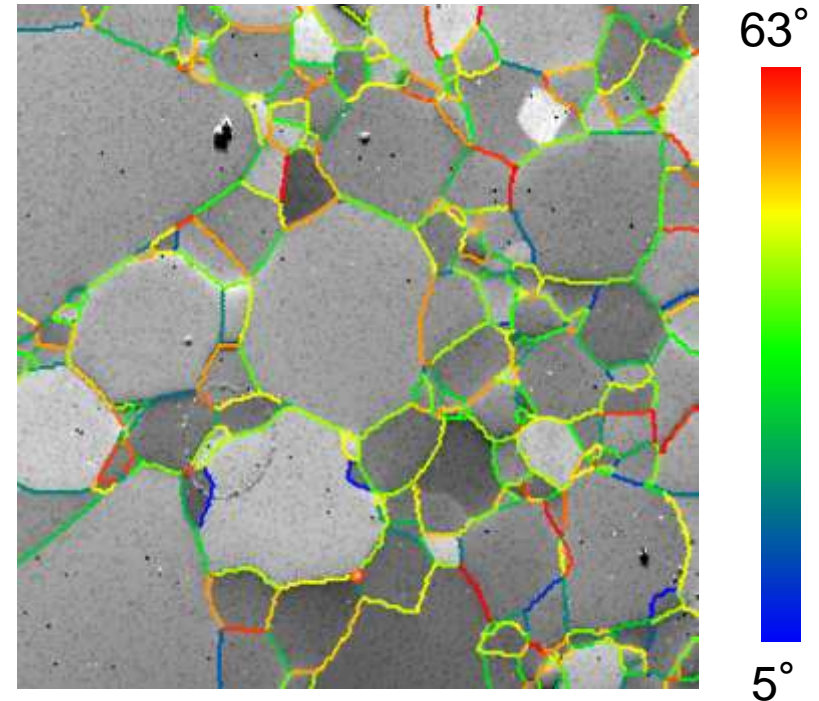
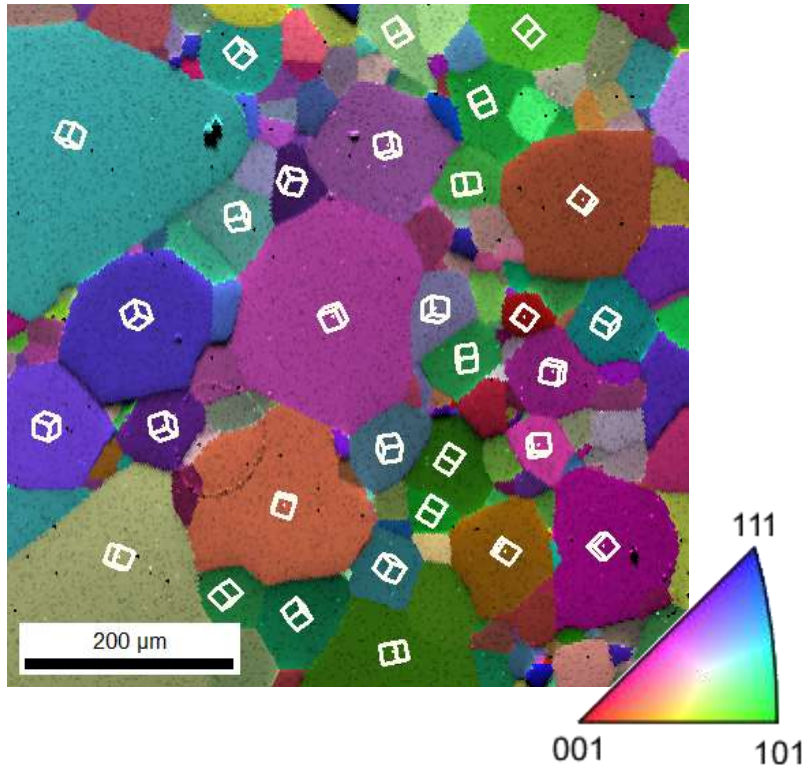


In an OIM scan the beam is stepped across the sample surface in a regular grid. At each point the EBSP is captured and automatically indexed and the orientation and other information recorded (such as the quality of the EBSP, an indexing reliability factor, the secondary detector intensity, EDS data...)

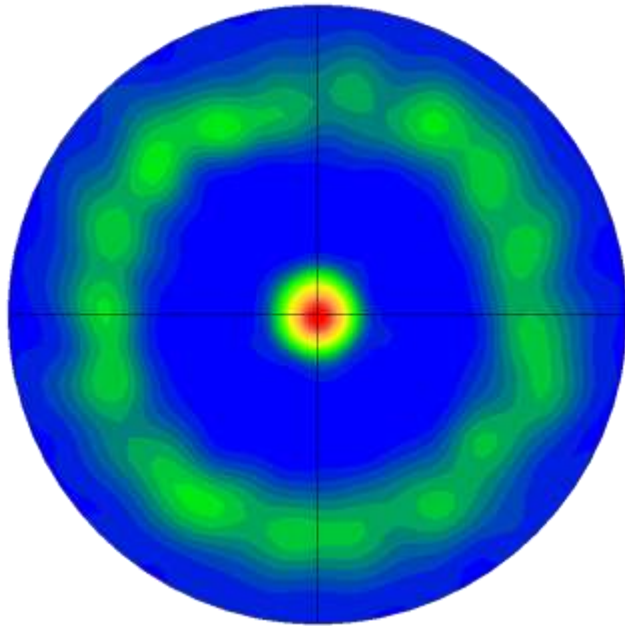


# Orientation Maps

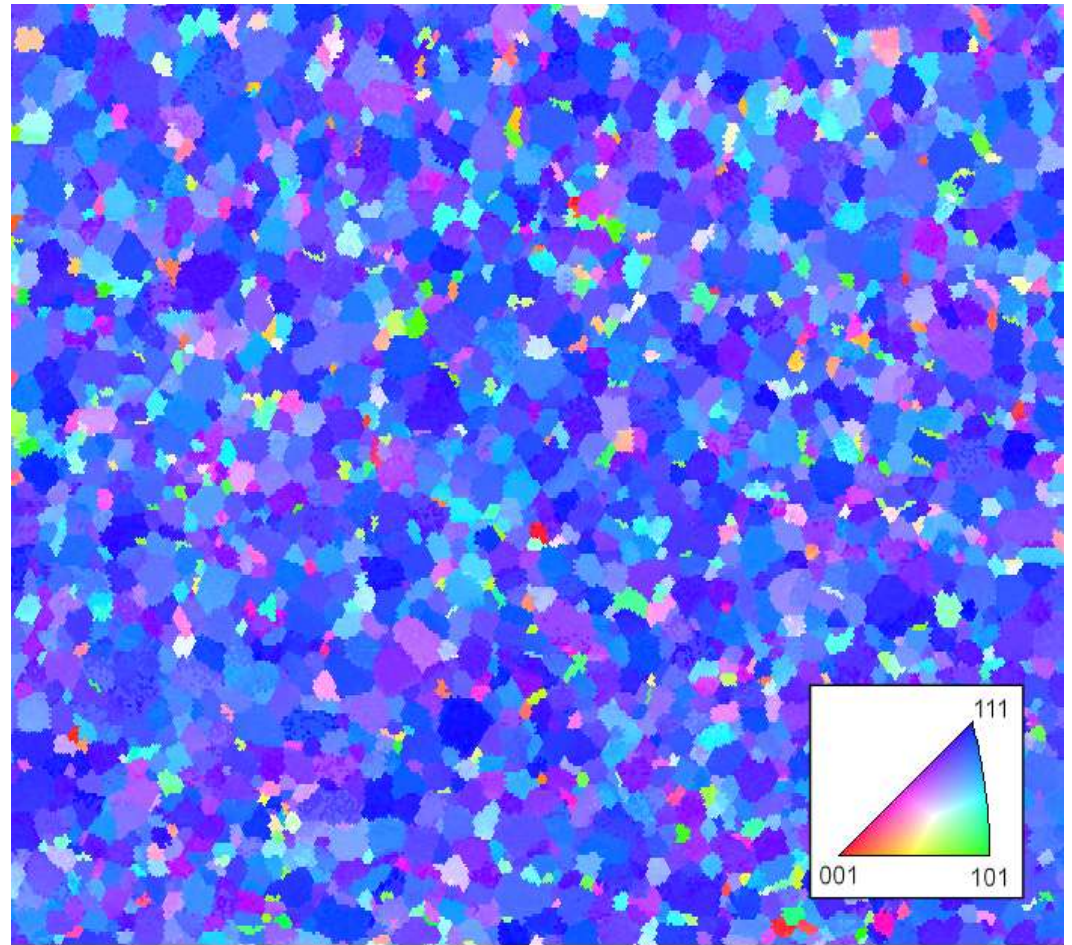
- An Orientation Map is generated by shading each point in the OIM scan according to some parameter reflecting the orientation at each point.
- A **Grain Boundary Map** can be generated by comparing the orientation between each pair of neighboring points in an OIM scan. A line is drawn separating a pair of points if the difference in orientation between the points is within a specified range.



# Texture



111 Pole Figure and corresponding OIM Map from an Aluminum Thin Film

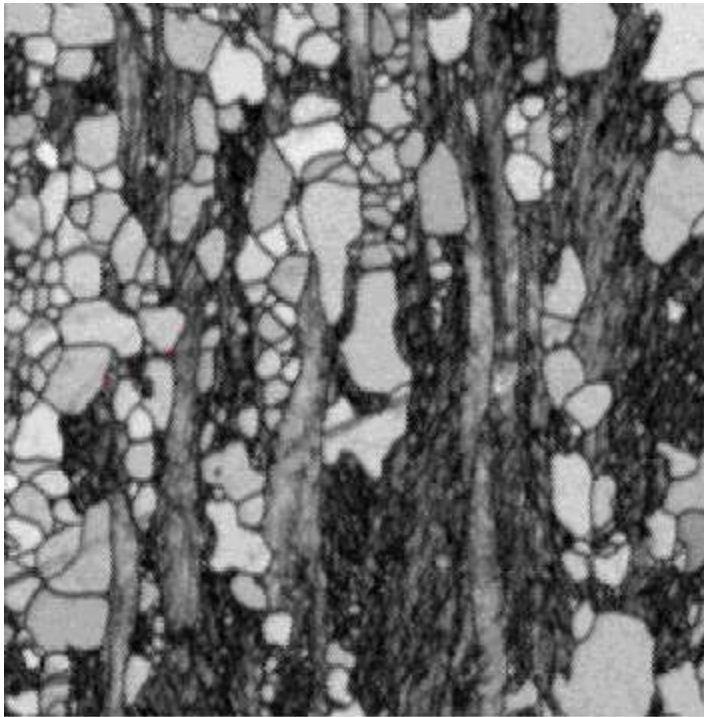


10 μm

# Residual Strain

## Pattern Quality

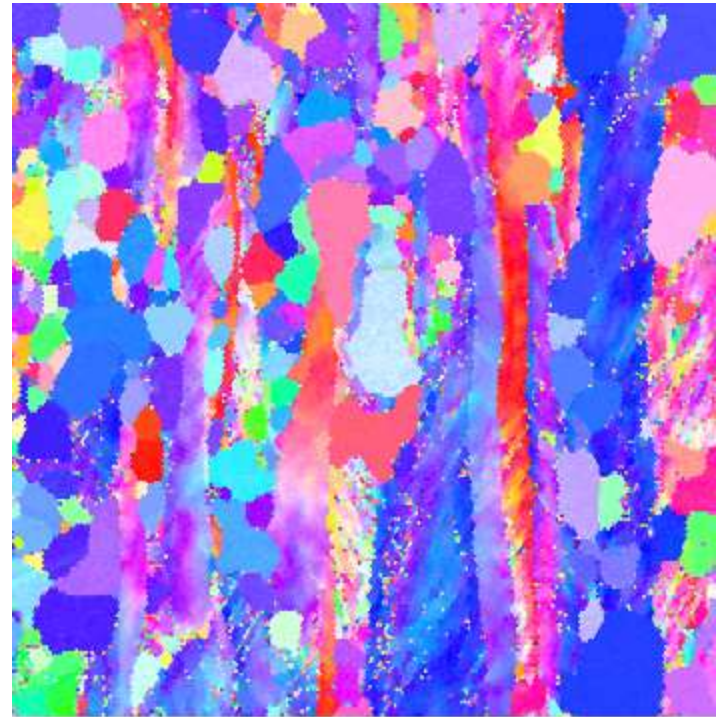
- Dark regions are strained
- Light regions are recrystallized



37.50  $\mu\text{m}$  = 50 steps IQ 10.4...57.8

## Local Misorientation

- Subtle changes in color

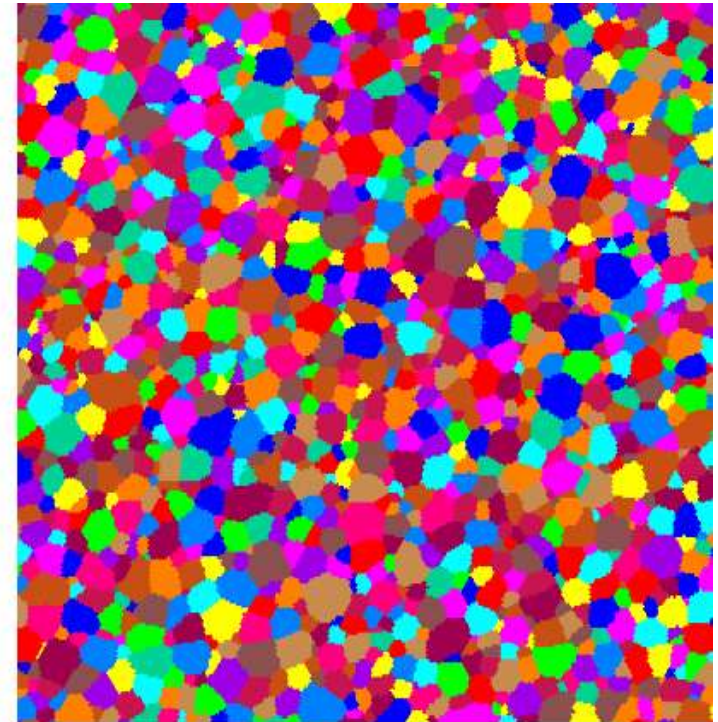
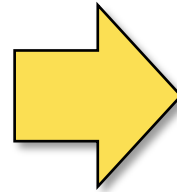
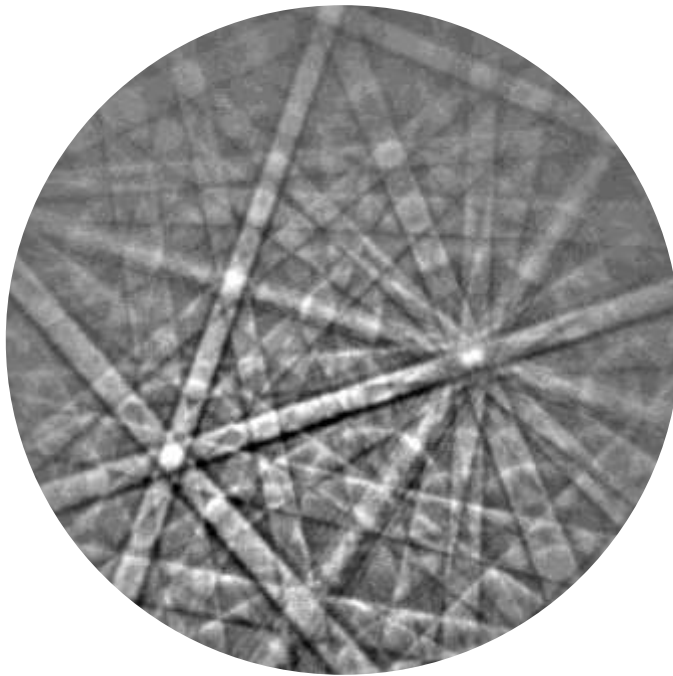


37.50  $\mu\text{m}$  = 50 steps IPF

# Specs

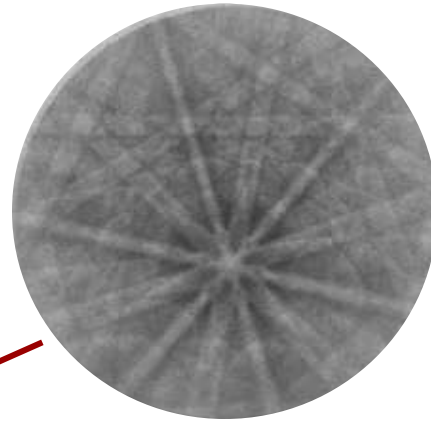
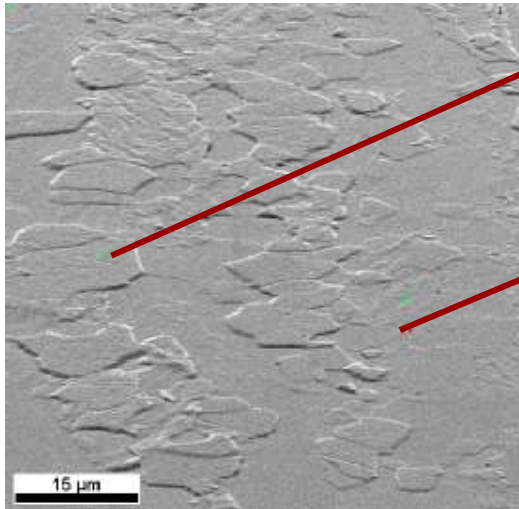
- **Electron Backscatter Diffraction**

- Spatial resolution ( $\sim 20\text{nm}$ )
- Angular resolution ( $\sim 0.3^\circ$ )
- Automation ( $\sim 500\text{pps}$ )

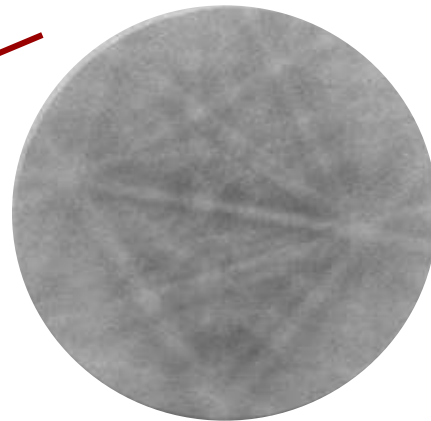


# Sample Conditions

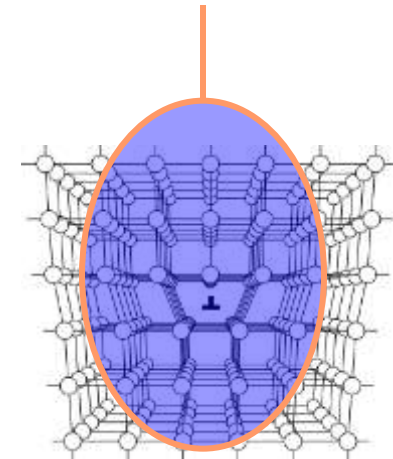
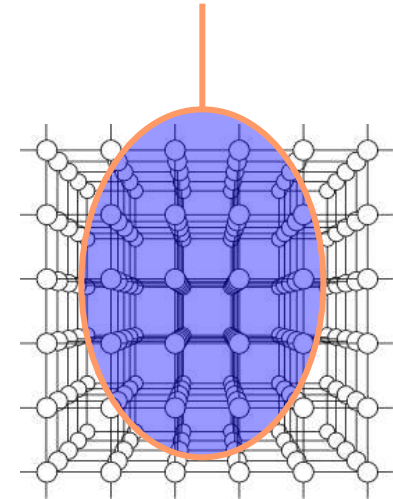
Plastic Strain  
Elastic Strain  
Lattice Distortions  
(Probe size)



Recrystallized



Deformed

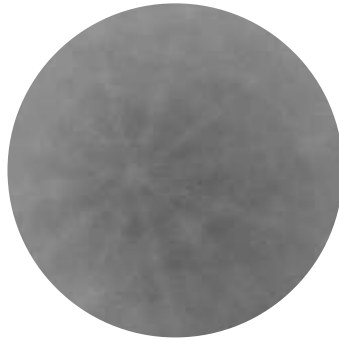


# Sample Preparation

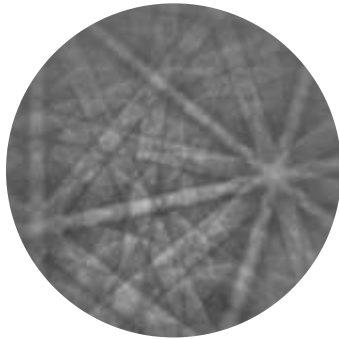
Surface Strain  
Topography  
Oxidation layers  
Conductive coating



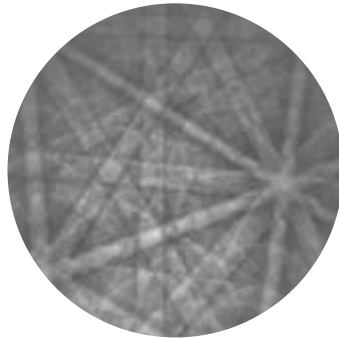
1200 um SiC



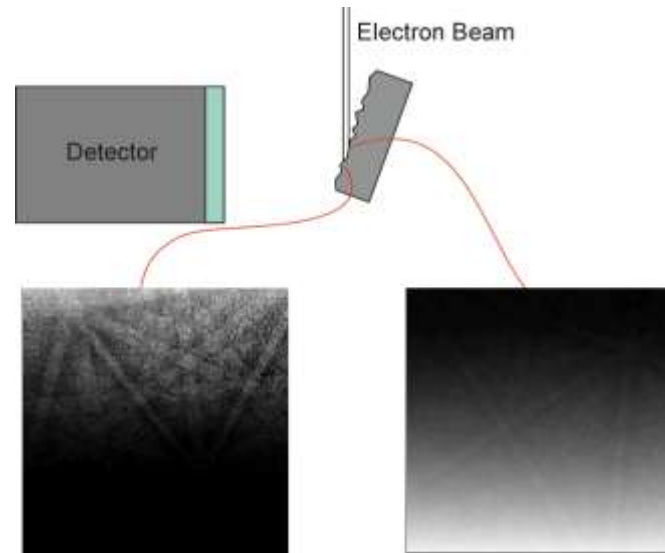
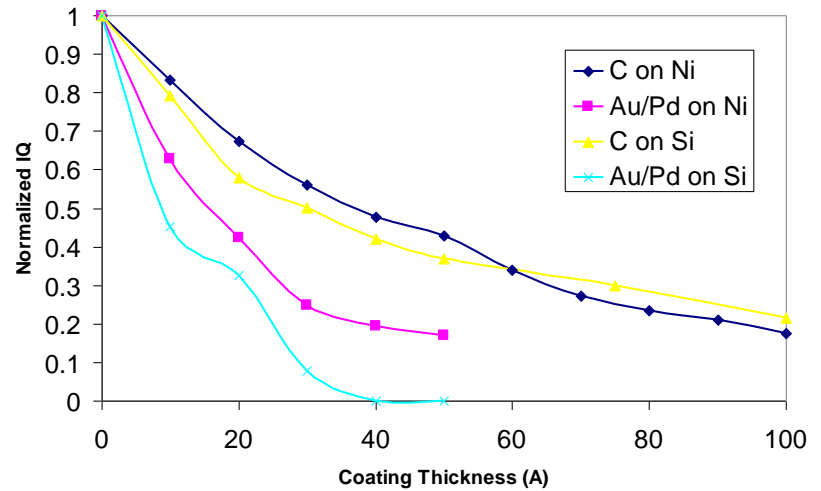
1 um Al<sub>2</sub>O<sub>3</sub>



0.3 um Al<sub>2</sub>O<sub>3</sub>



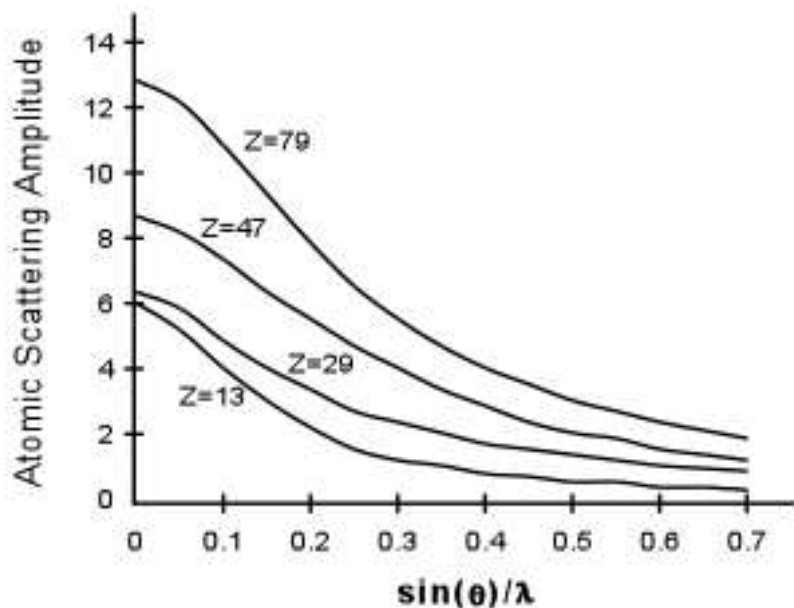
0.05 um SiO<sub>2</sub> 10 min



# Atomic Scattering Factor

$$F_{hkl} = \sum_j f_j(\theta) \exp(-2\pi i [hu_j + kv_j + lw_j])$$

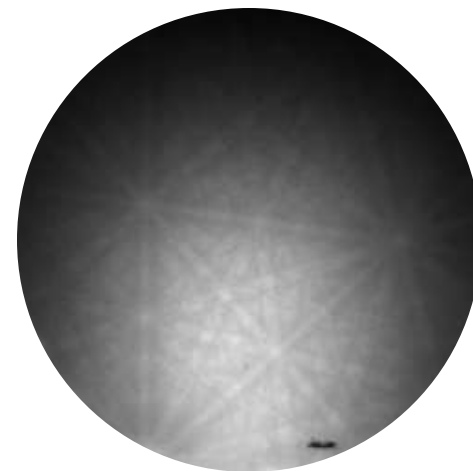
$$f(\theta) = \frac{m_o e^2}{2h^2} \left[ \frac{\lambda}{\sin \theta} \right]^2 (Z - f_x)$$



Aluminum



Tantalum



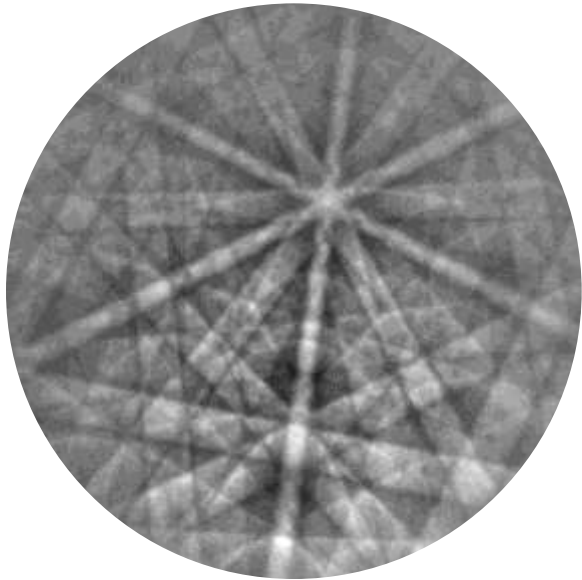
EBSPs obtained at 1.92s exposure time (no image processing)

# Microscope Conditions

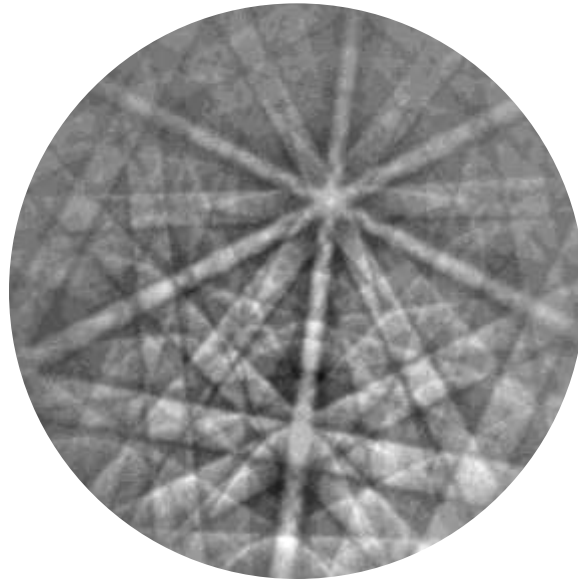
Probe Size (FE vs. LaB<sub>6</sub> vs. Tungsten)

Operating Conditions (**Current**, Voltage, Vacuum)

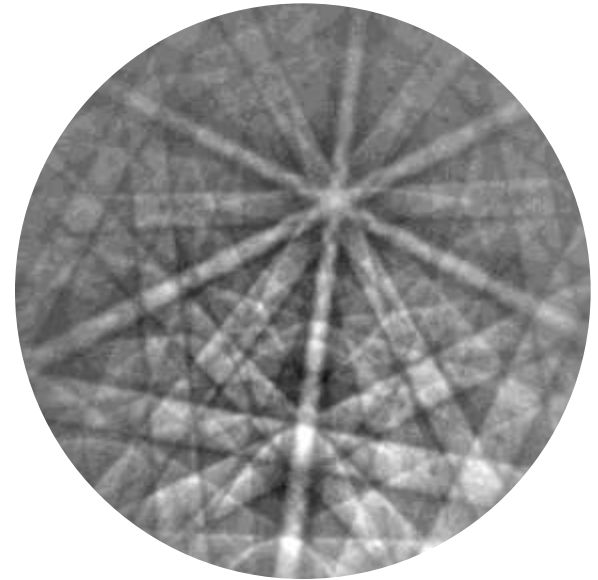
Video Settings (Exposure, gain, contrast & brightness)



**0.6nA Beam Current**  
**4.62 Seconds**



**2.4nA Beam Current**  
**1.56 Seconds**



**9.45nA Beam Current**  
**0.6 Seconds**

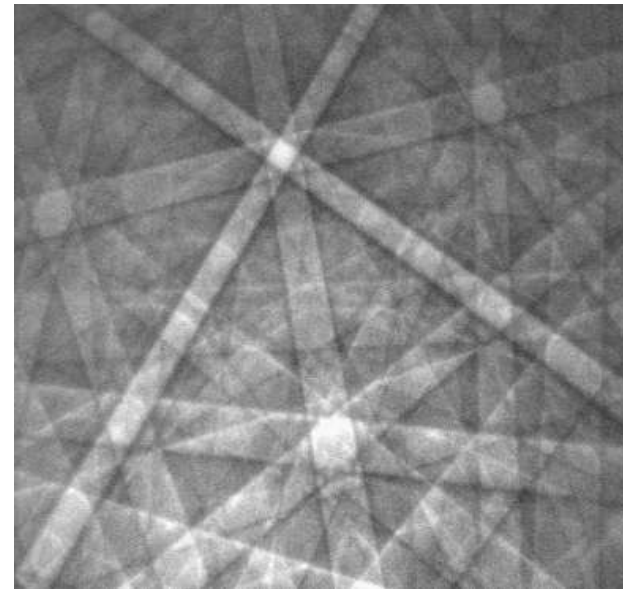
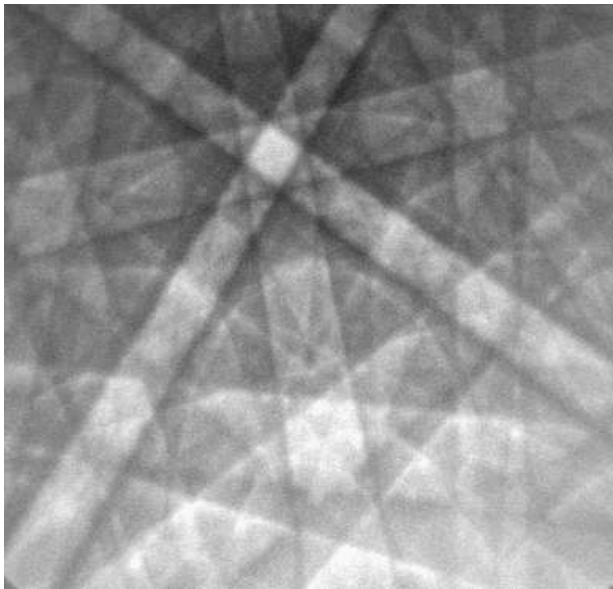


# Microscope Conditions – Voltage Effect

Probe Size (FE vs. LaB<sub>6</sub> vs. Tungsten)

Operating Conditions (Current, **Voltage**, Vacuum)

Video Settings (Exposure, gain, contrast & brightness)

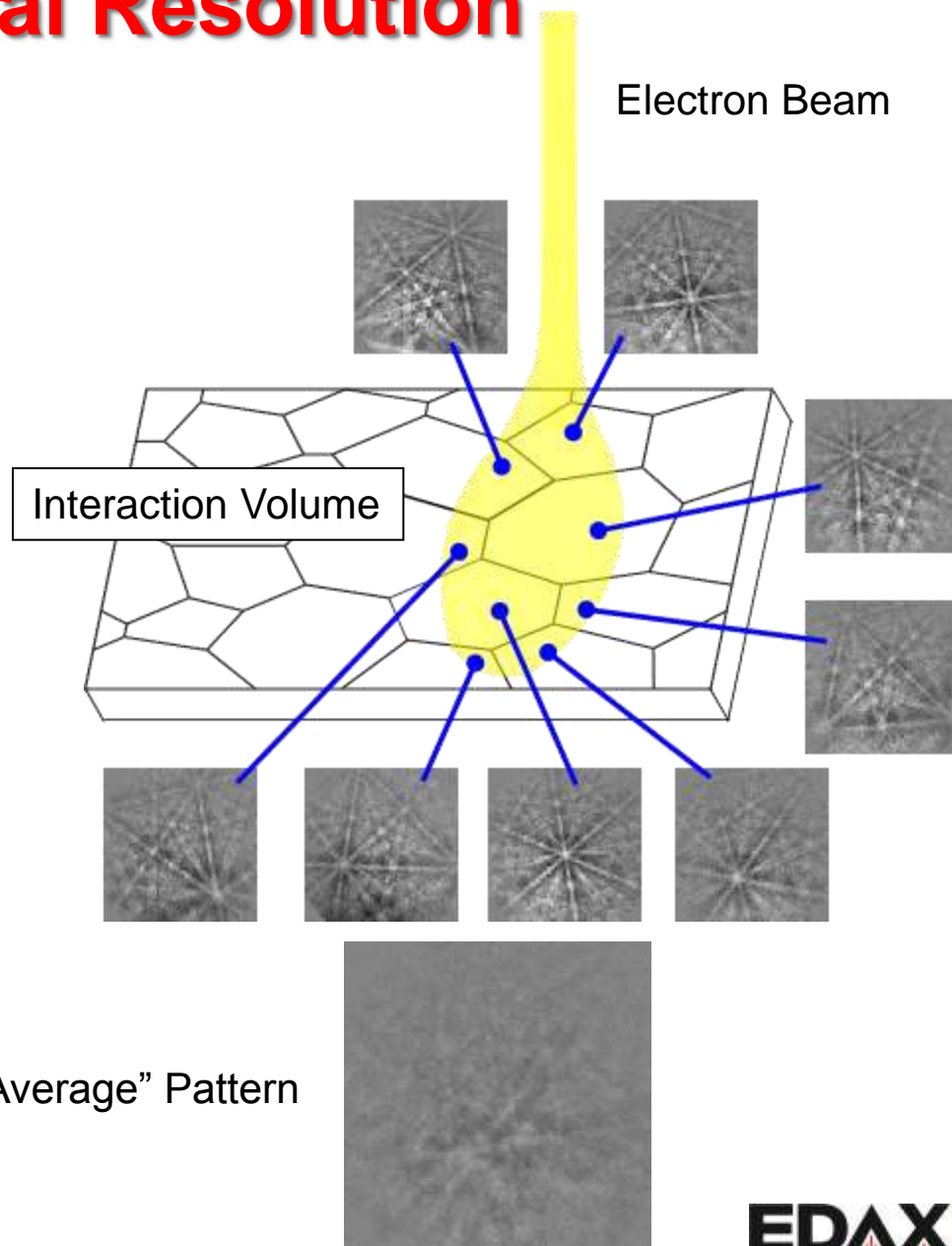


Si at 10 kV and 30 kV

# EBSD Spatial Resolution

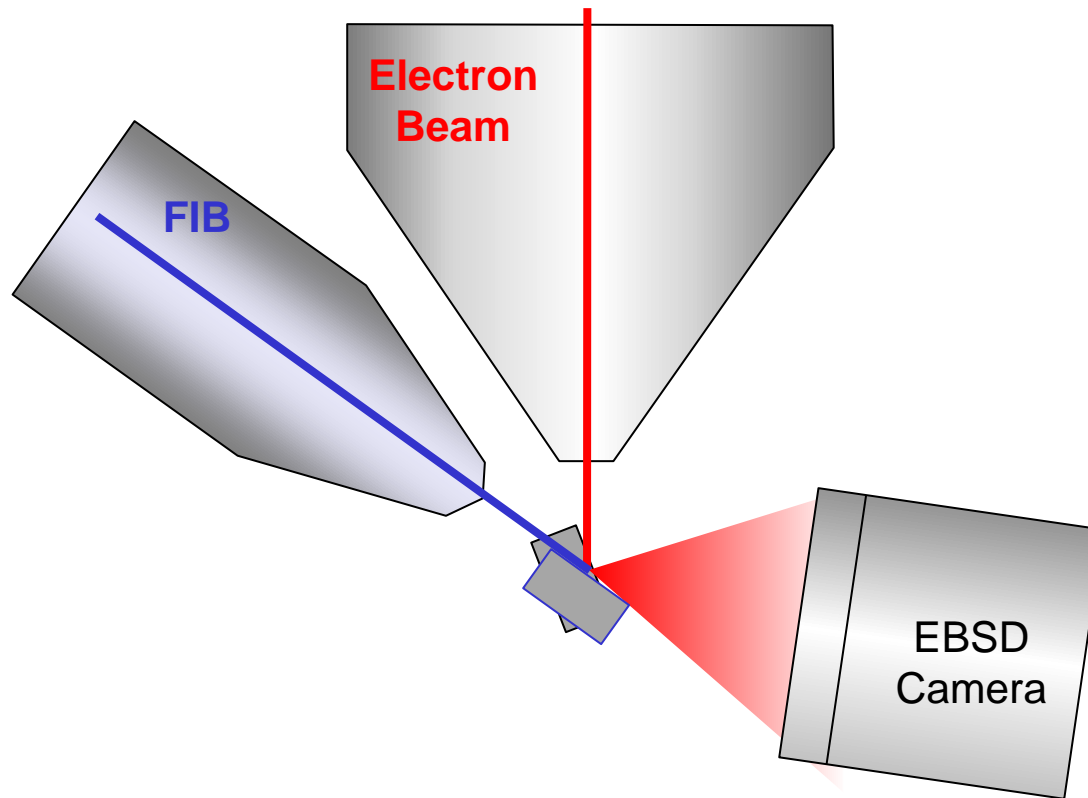
- **Interaction Volume**

- Spot Size  
Current generally increases with increasing spot size
- Current  
~5nA (1-10nA)
- Voltage  
Seems to effect depth more than lateral resolution  
5-30 kV
- Material  
Higher Z materials are better reflectors
- Operating Conditions  
Drift vs. Speed



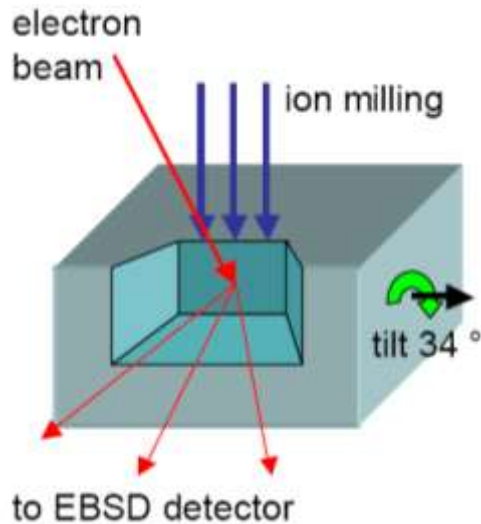
# Serial Sectioning

- Use of Combined SEM/FIB with EBSD



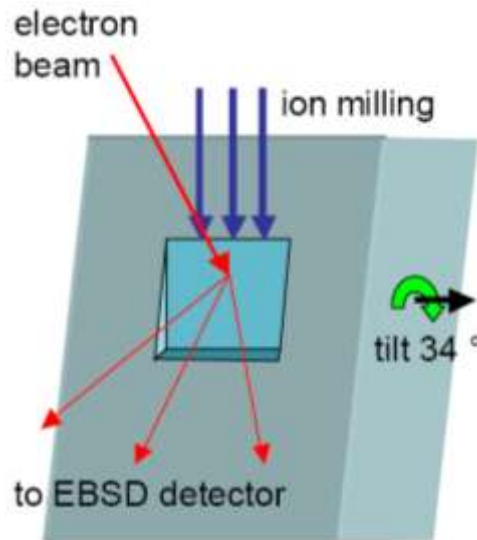
# In-Situ Milling

**milling strategy:**  
grazing-incidence  
edge-milling



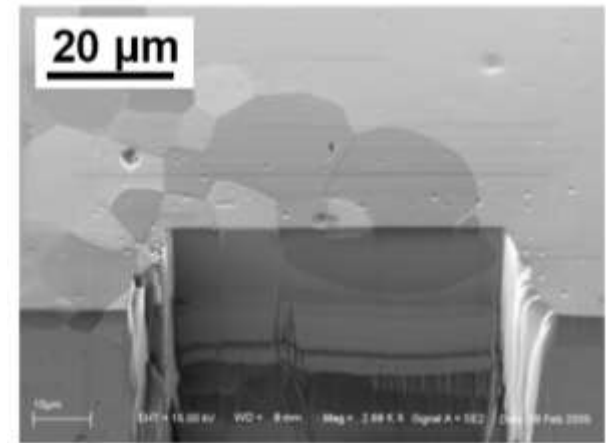
feature has to be  
at edge

**milling strategy:**  
low-incidence  
surface-milling



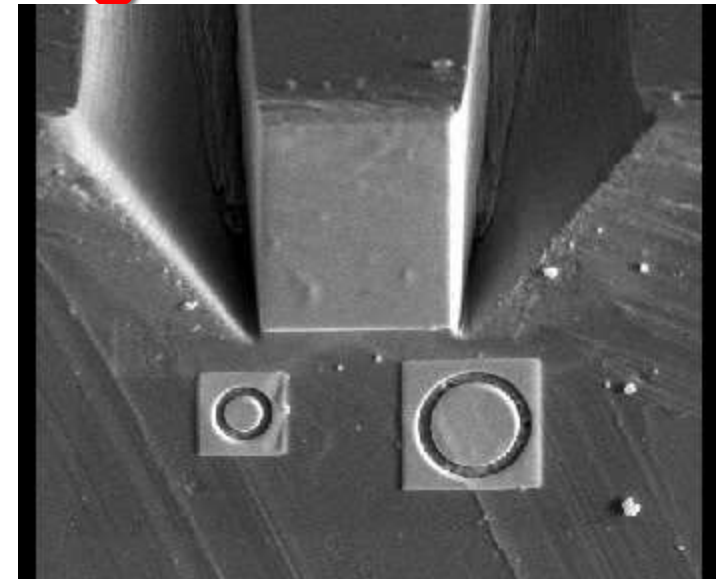
large milling areas  
required to avoid  
shadowing of EBSD

edge-milling  
Fe-Si



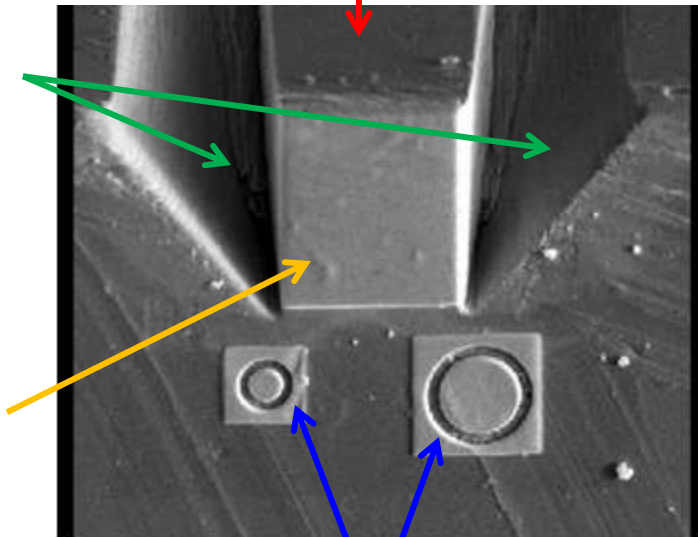
# Serial Sectioning

SEM View



**EBSD surface** –  
orientation mapping

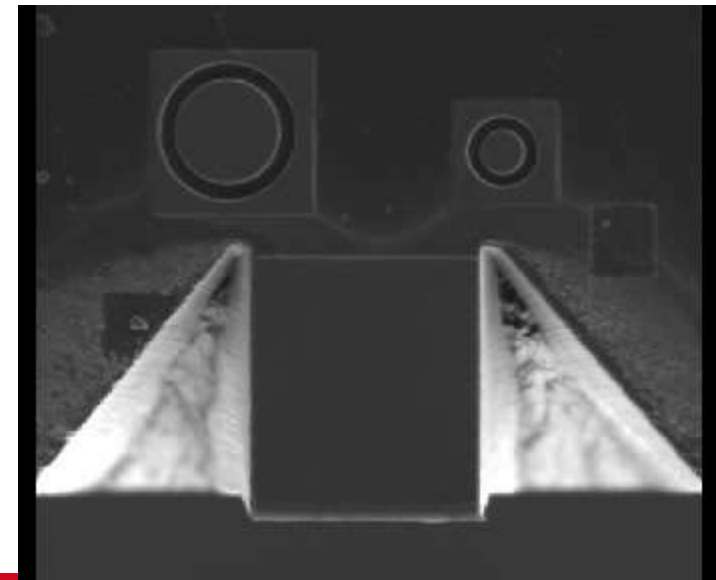
**Trenches** – to  
avoid shadowing  
of EBSD detector



**Platinum Cap** –  
to prevent  
curtaining

**Fiducial marks** – for  
re-alignment between  
stage moves. (Atop  
platinum for improved  
contrast)

FIBView



# Indexing Details

- Helpful to understand what is happening in the algorithms so an operator can better make adjustments to optimize the results
- The algorithms between the EBSD vendors differ from each other but the underlying principles are similar.

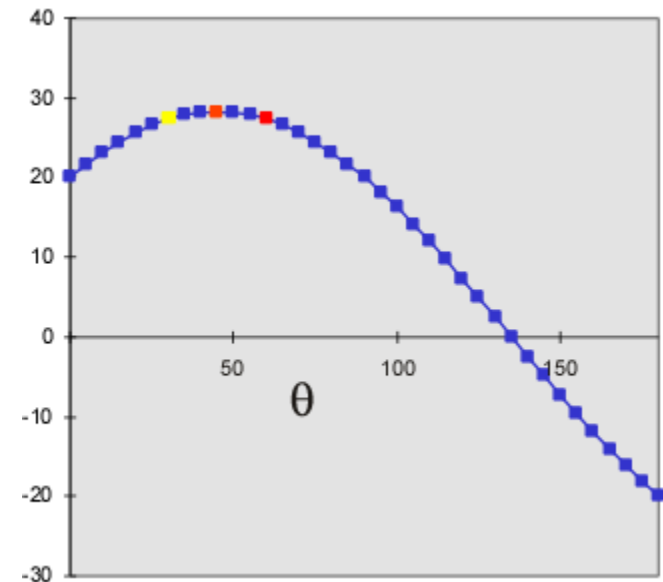
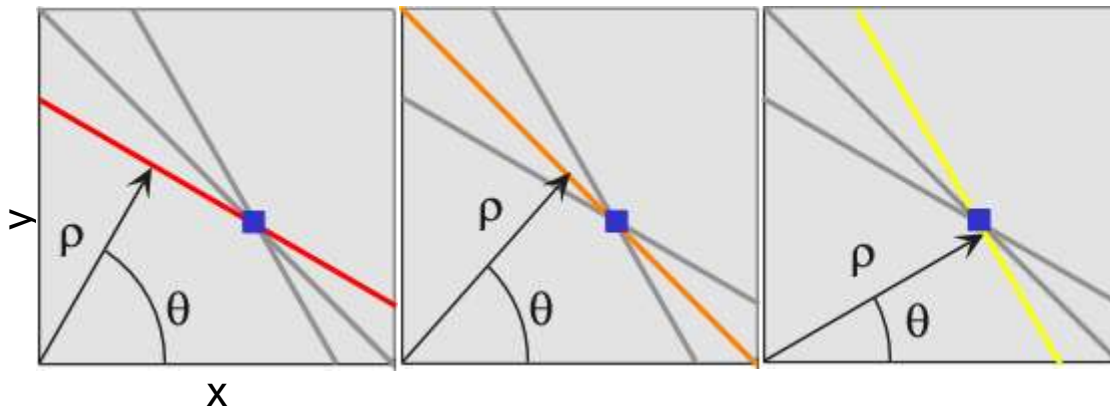
# Q. How do we find the bands? Hough Transform



# Hough Transform

A given pixel in an image could belong to an infinite set of lines. A line can be parameterized by the Hough parameters  $r$  and  $q$ . Where  $q$  describes the angle of the line and  $r$  represents the perpendicular distance of the line from the origin. The relationship between the lines passing through a pixel at a coordinate in the image of  $x, y$  can be expressed as:  $r = x \cos q + y \sin q$ . This means a point in image space transforms to a sinusoidal curve in Hough space.

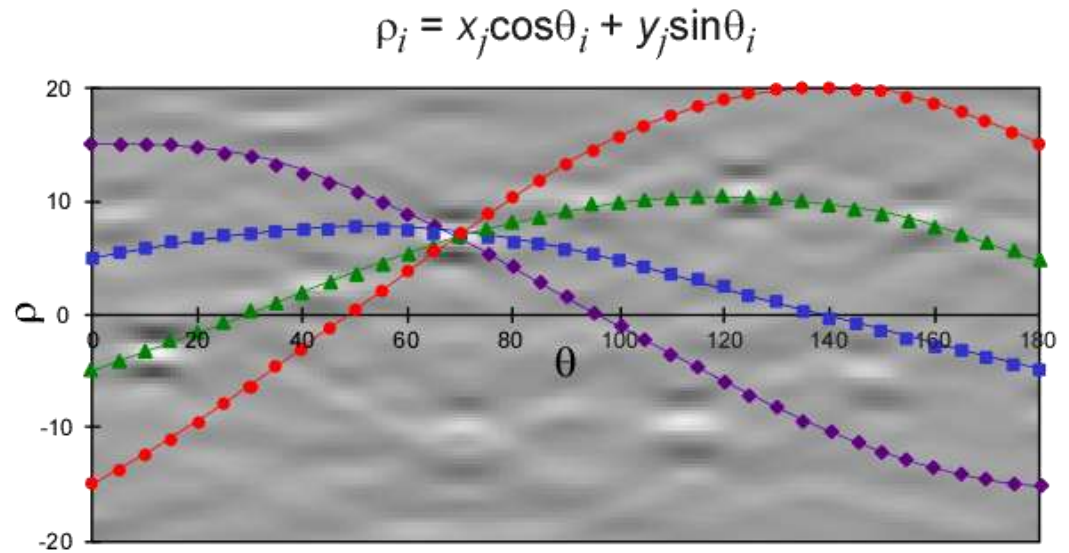
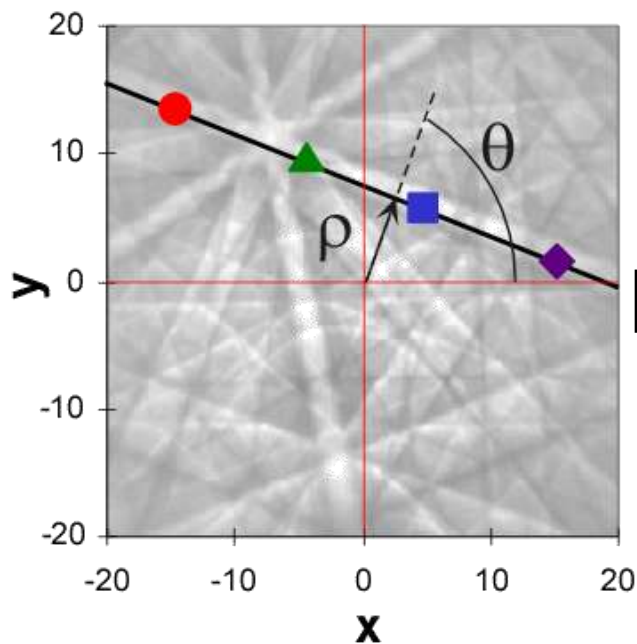
$$\rho_i = x \cos \theta_i + y \sin \theta_i$$





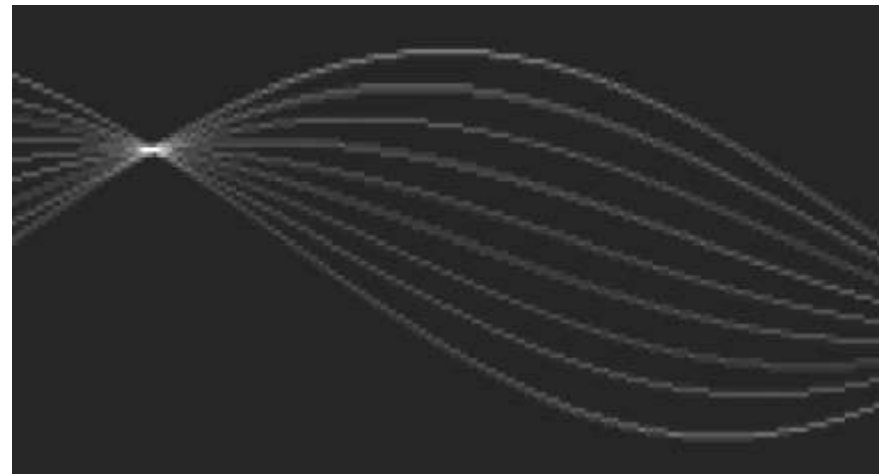
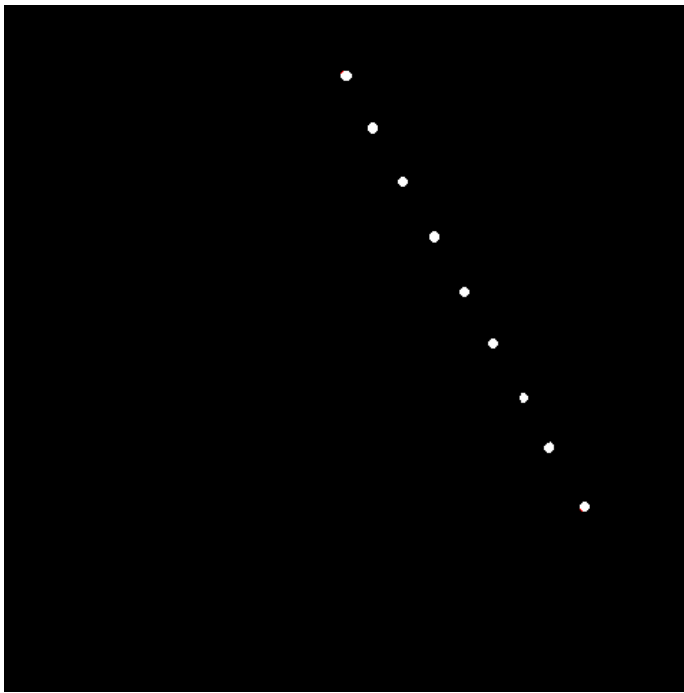
# Hough Transform

Consider 4 pixels along a line. For each pixel in the line, all possible  $\rho$  values are calculated for  $\theta$ 's ranging in values from 0 to 180 degrees using the equation:  $\rho = x\cos\theta + y\sin\theta$ . This produces 4 sinusoidal curves. These curves intersect at a point at a  $\rho, \theta$  coordinate corresponding to the angle of the line ( $\theta$ ) and its position relative to the origin ( $\rho$ ). Thus, a line in the pattern space transforms to a point in Hough Space.



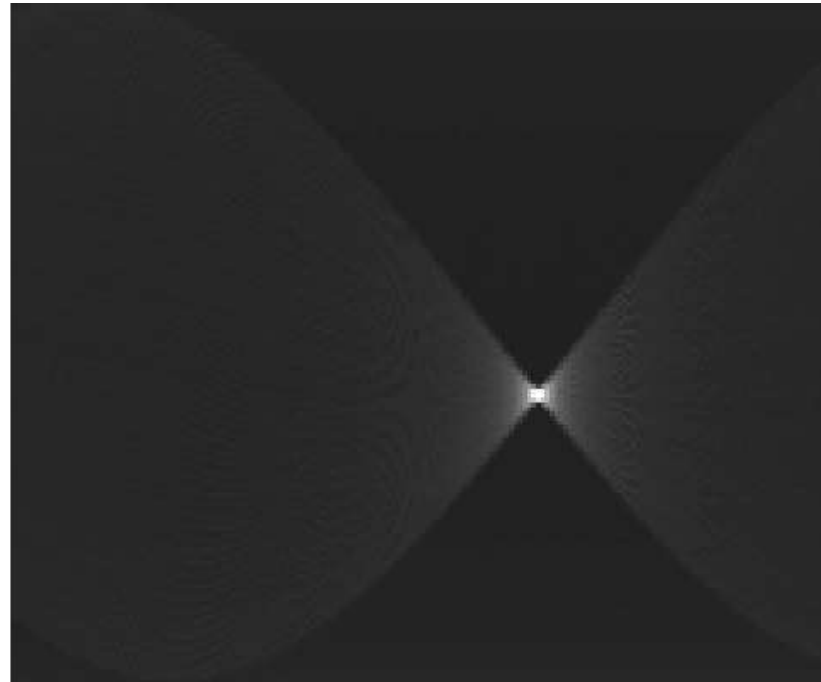
# Hough Transform

An entire image can be transformed into Hough Space by building an accumulator array  $H(\rho, \theta)$  where, for each pixel in the image, all possible  $r$  values are calculated for  $\theta$ 's ranging in values from 0 to 180 degrees via the equation  $\rho = x\cos\theta + y\sin\theta$ . The intensity value of the pixel at  $x, y$  is then added to the bin in the array at each corresponding  $\rho, \theta$ . (Strictly speaking the Hough Transform only applies to binary images - this adaptation is the Radon Transform).

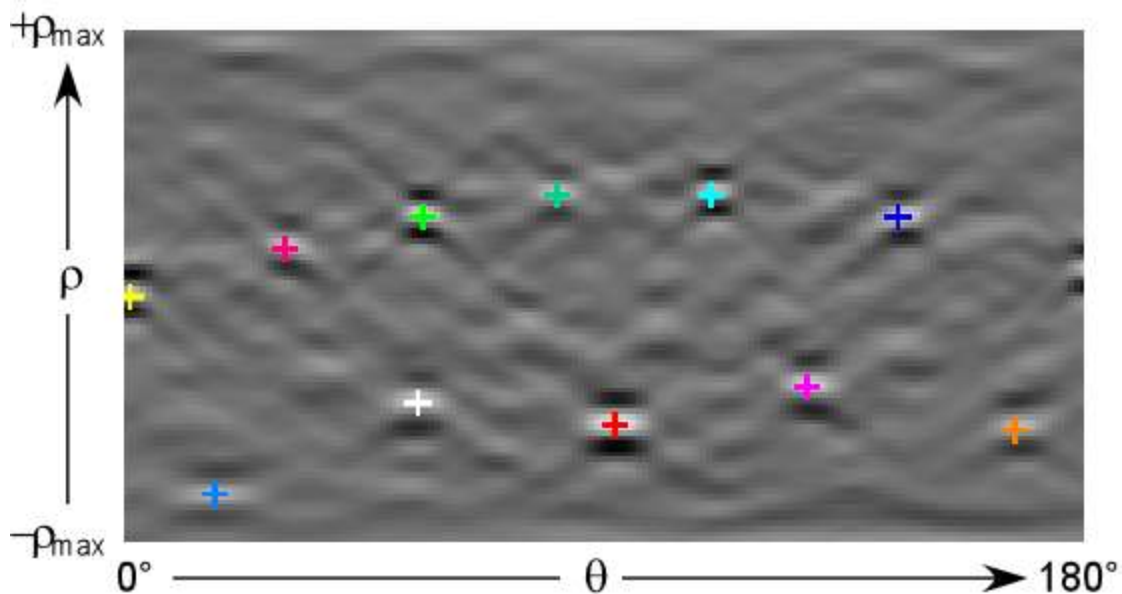
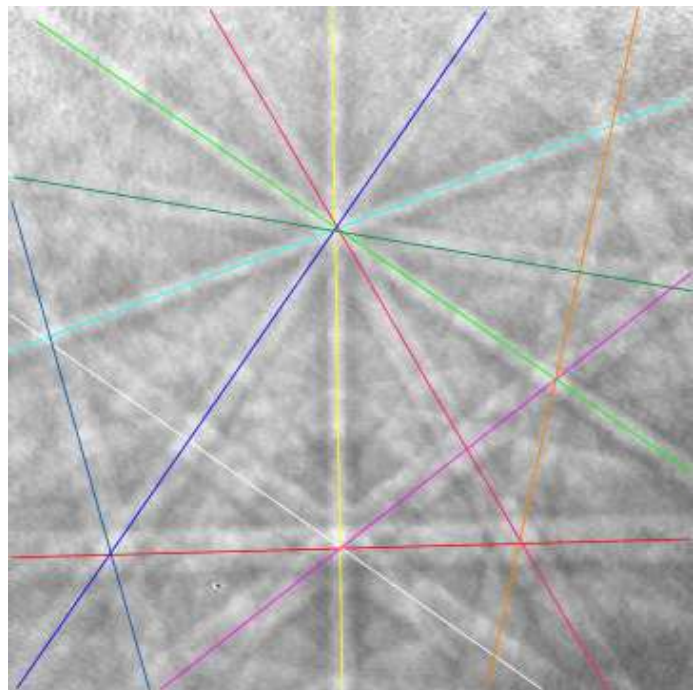


# Hough Transform

An entire image can be transformed into Hough Space by building an accumulator array  $H(\rho, \theta)$  where, for each pixel in the image, all possible  $r$  values are calculated for  $\theta$ 's ranging in values from 0 to 180 degrees via the equation  $\rho = x\cos\theta + y\sin\theta$ . The intensity value of the pixel at  $x, y$  is then added to the bin in the array at each corresponding  $\rho, \theta$ . (Strictly speaking the Hough Transform only applies to binary images - this adaptation is the Radon Transform).

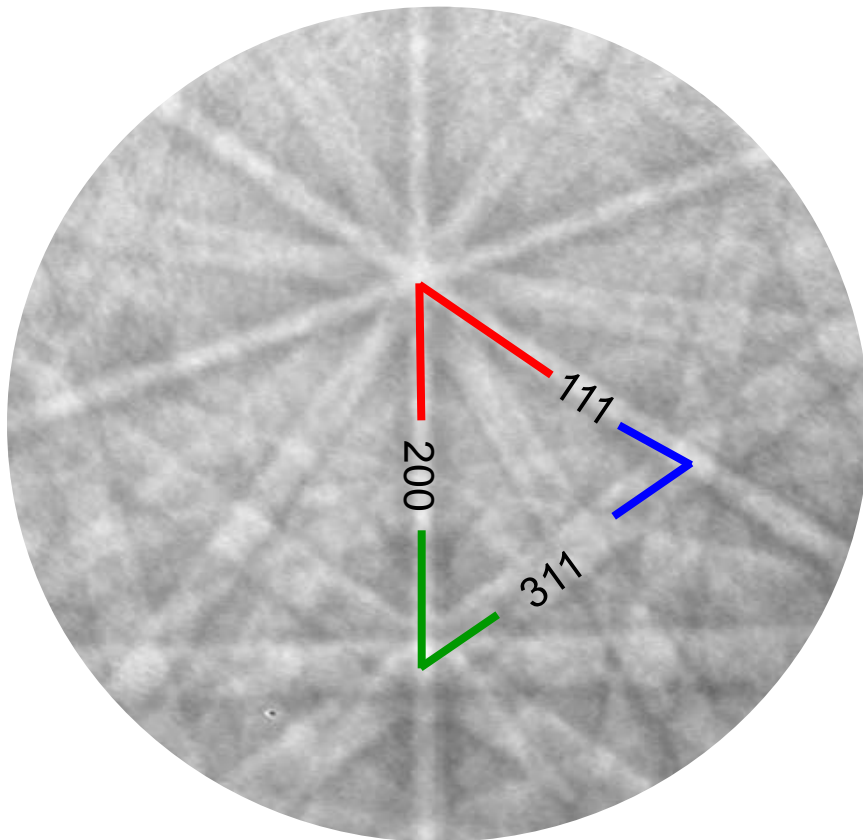


# Hough Transform



# Indexing: Bands – Triplet Solutions

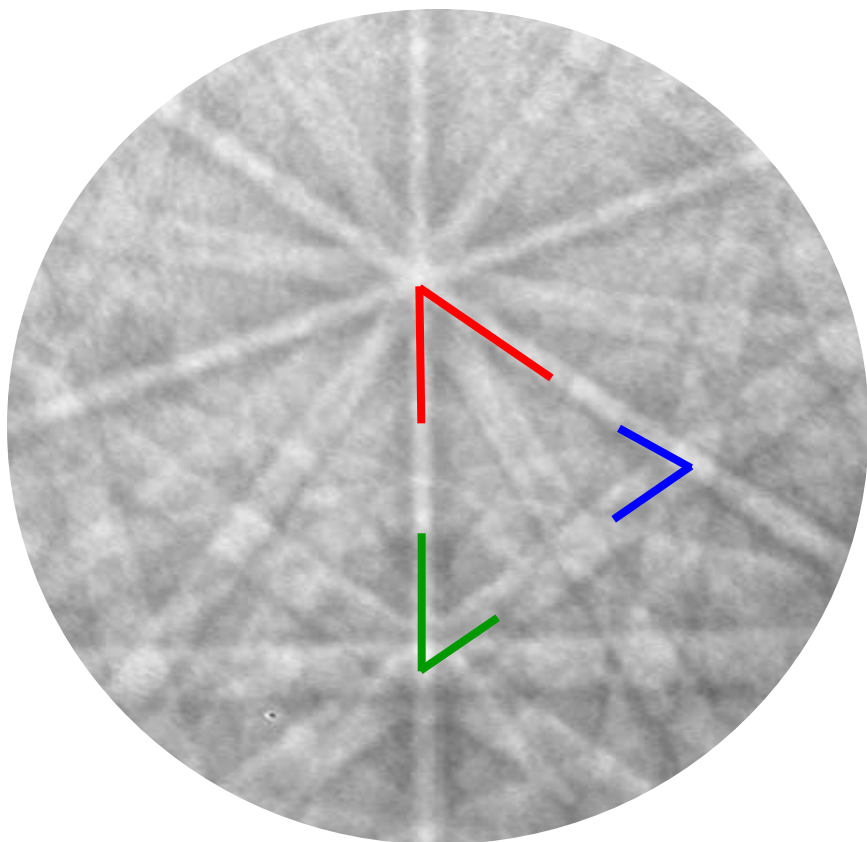
A set of orientations is obtained from a triplet of bands by comparing the interplanar angles against a look-up table.



Angle	(hkl) <sub>1</sub>	(hkl) <sub>2</sub>
25.2	200	311
29.5	111	311
31.5	220	311
35.1	311	311
35.3	111	220
45.0	200	220
50.5	311	311
54.7	111	200
58.5	111	311
60.0	220	202
63.0	311	131
64.8	220	311
70.5	111	111
72.5	200	131
80.0	111	311
84.8	311	131
90.0	111	220
90.0	200	020
90.0	200	022
90.0	220	113
90.0	220	220

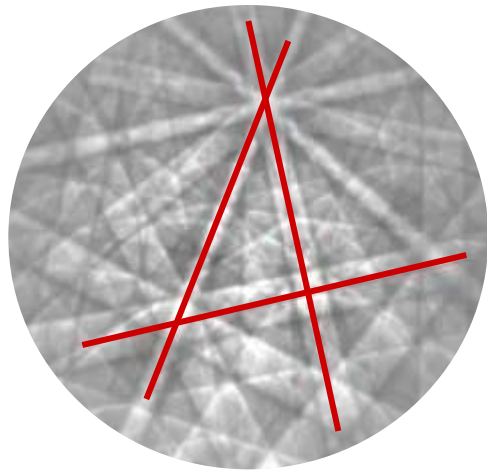
# Indexing: Bands – Triplet Solutions

Now allow a little bit of “wobble” by setting the *interplanar angle tolerance* to 5 degrees. Let the “red” angle be measured as 55 degrees, the “blue” as 59 degrees and the green as 72 degrees. Now, multiple solutions can exist for the triplet.



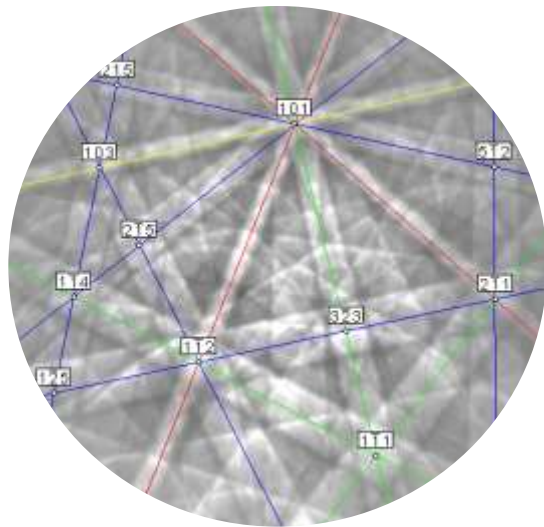
Angle	(hkl) <sub>1</sub>	(hkl) <sub>2</sub>
25.2	200	311
29.5	111	311
31.5	220	311
35.1	311	311
35.3	111	220
45.0	200	220
50.5	311	311
54.7	111	200
58.5	111	311
60.0	220	202
63.0	311	131
64.8	220	311
70.5	111	111
72.5	200	131
80.0	111	311
84.8	311	131
90.0	111	220
90.0	200	020
90.0	200	022
90.0	220	113
90.0	220	220

# Indexing: Bands – Triplet Solutions

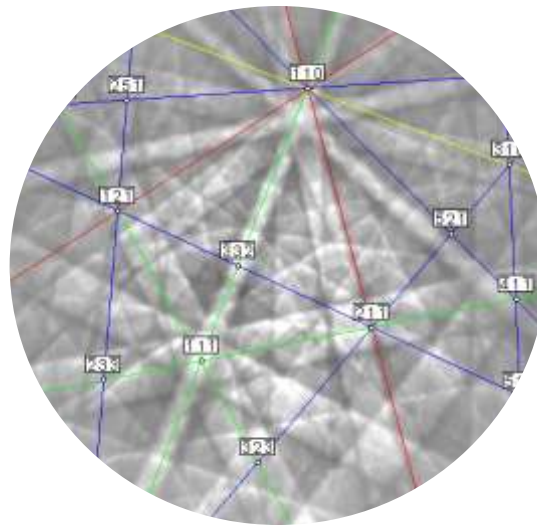


For a set of three bands, compare the interplanar angles against the LUT and determine all possible indexing solutions.

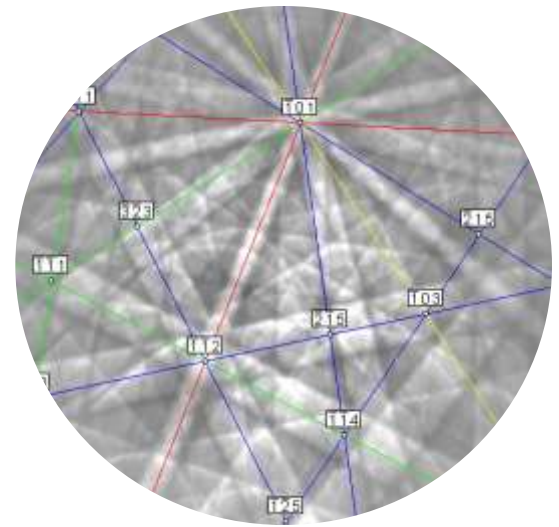
Solution 1



Solution 2



Solution 3



# Indexing: Bands – Triplet Solutions

$$\#triplets = \frac{n!}{(n-3)! \cdot 3!}$$

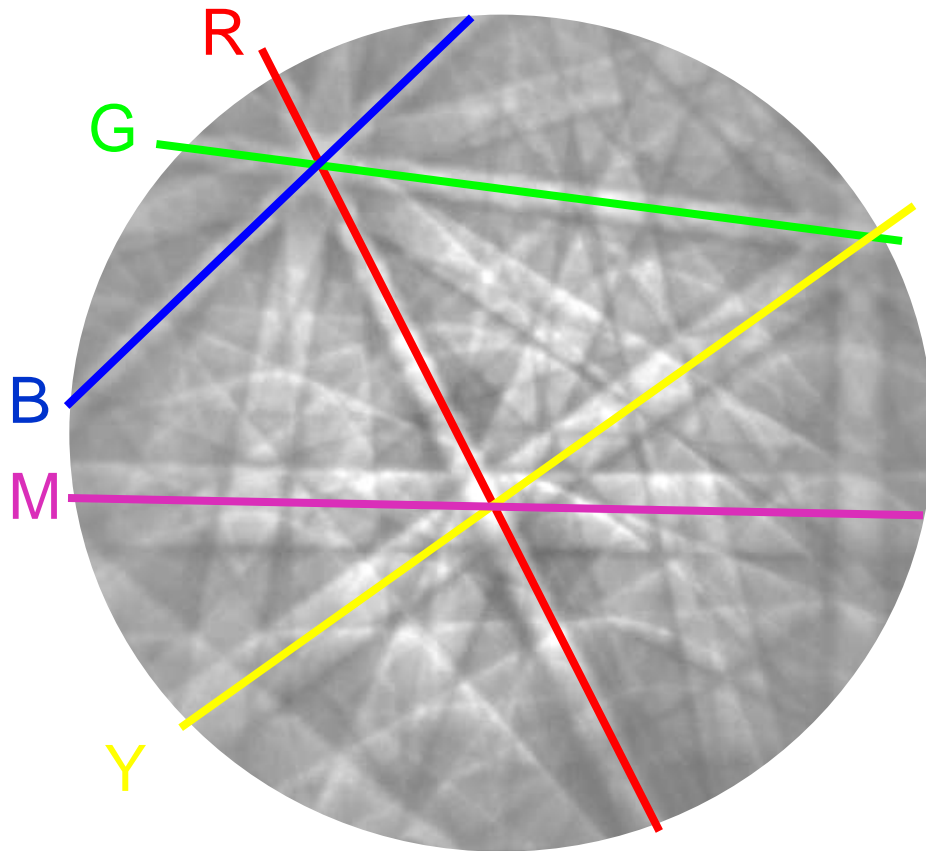
For a given number of bands,  $n$ , used for pattern indexing, the number of band triplets is determined by this formula.

Typically 7 to 9 detected bands are used for automatic indexing.

$n$	# triplets
3	1
4	4
5	10
6	20
7	35
8	56
9	84



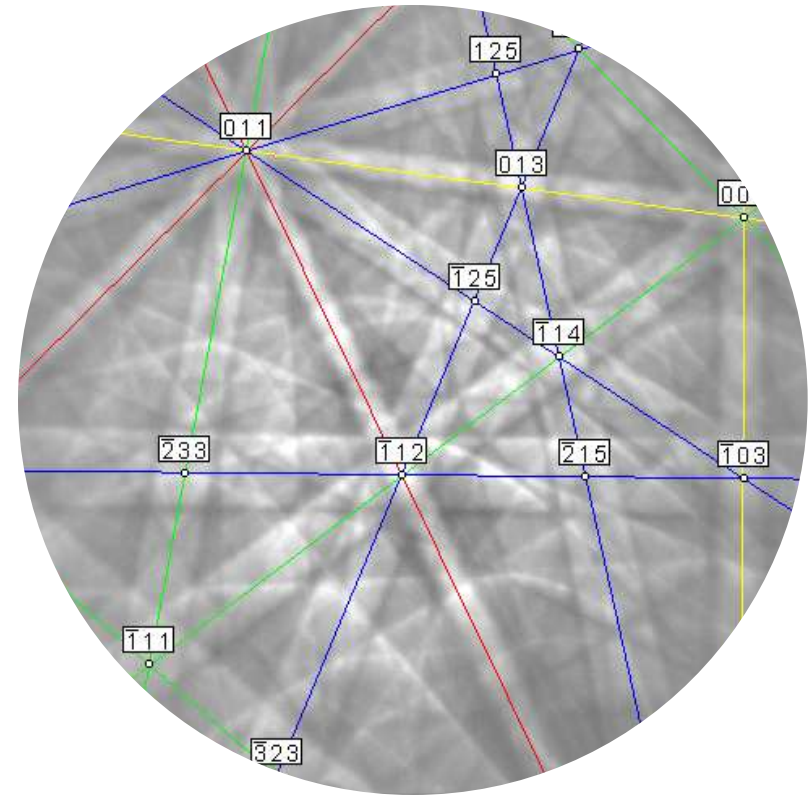
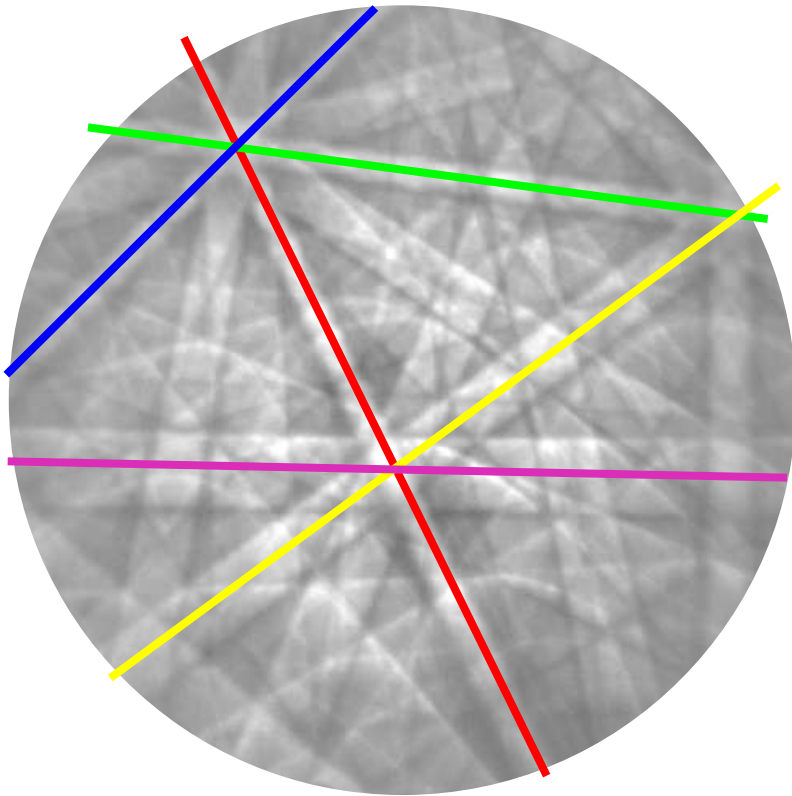
# Indexing: Bands – Triplet Voting



Triplet	Solution 1 ( $V_1$ )	Solution 2 ( $V_2$ )	Solution 3 ( $V_3$ )
R G Y	X		
R G B	X		
R G M	X		
R Y B	X		
R Y M	X	X	
R B M	X		
G Y B	X		
G Y M	X		
G B M	X		
Y B M	X		X
<b>Total</b>	<b>10</b>	<b>1</b>	<b>1</b>

For this set of 5 detected bands, 10 triplet combinations are possible. For each of these 10 triplets, solution  $V_1$  matched. Solutions  $V_2$  and  $V_3$  each matched one triplet only.

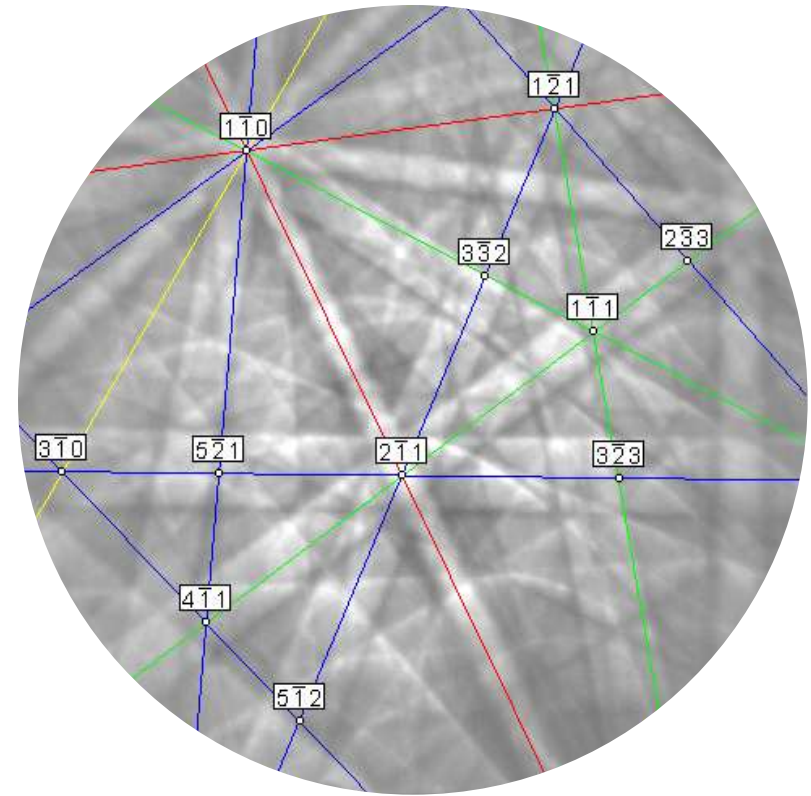
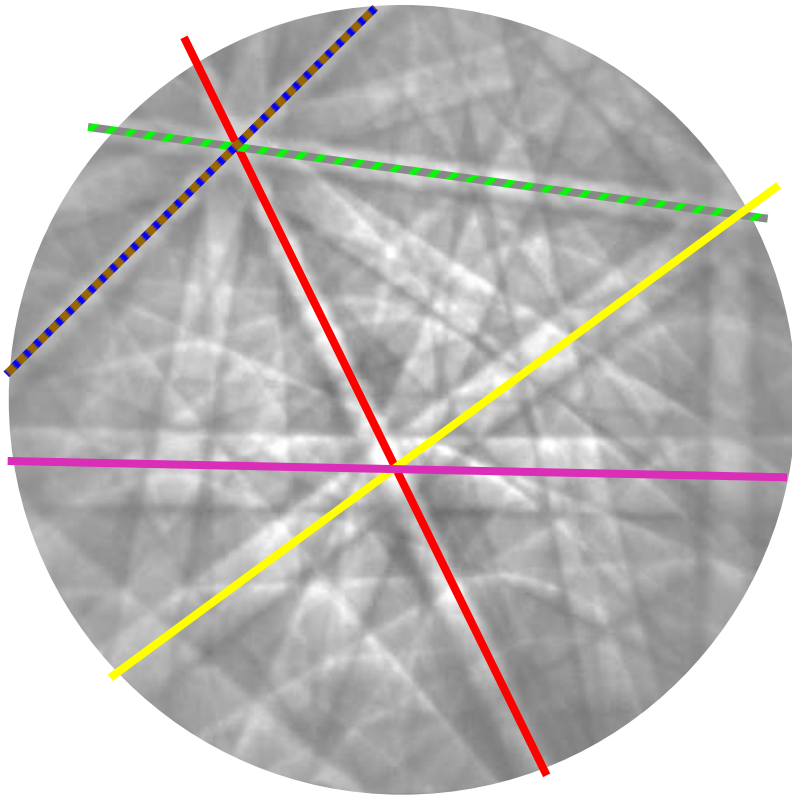
# Indexing: Bands – Triplet Voting



Fit = 0.26°

Solution  $V_1$ . Notice how each of the 5 bands match bands in the indexing solution overlay.

# Indexing: Bands – Triplet Voting



Fit = 1.30°

Solution  $V_2$ . Only the red, yellow, and magenta band triplet now match, producing a higher fit value.

# Indexing: Bands – Confidence Index

Triplet	Solution 1 (V <sub>1</sub> )	Solution 2 (V <sub>2</sub> )	Solution 3 (V <sub>3</sub> )
R G Y	X		
R G B	X		
R G M	X		
R Y B	X		
R Y M	X	X	
R B M	X		
G Y B	X		
G Y M	X		
G B M	X		
Y B M	X		X
<b>Total</b>	<b>10</b>	<b>1</b>	<b>1</b>

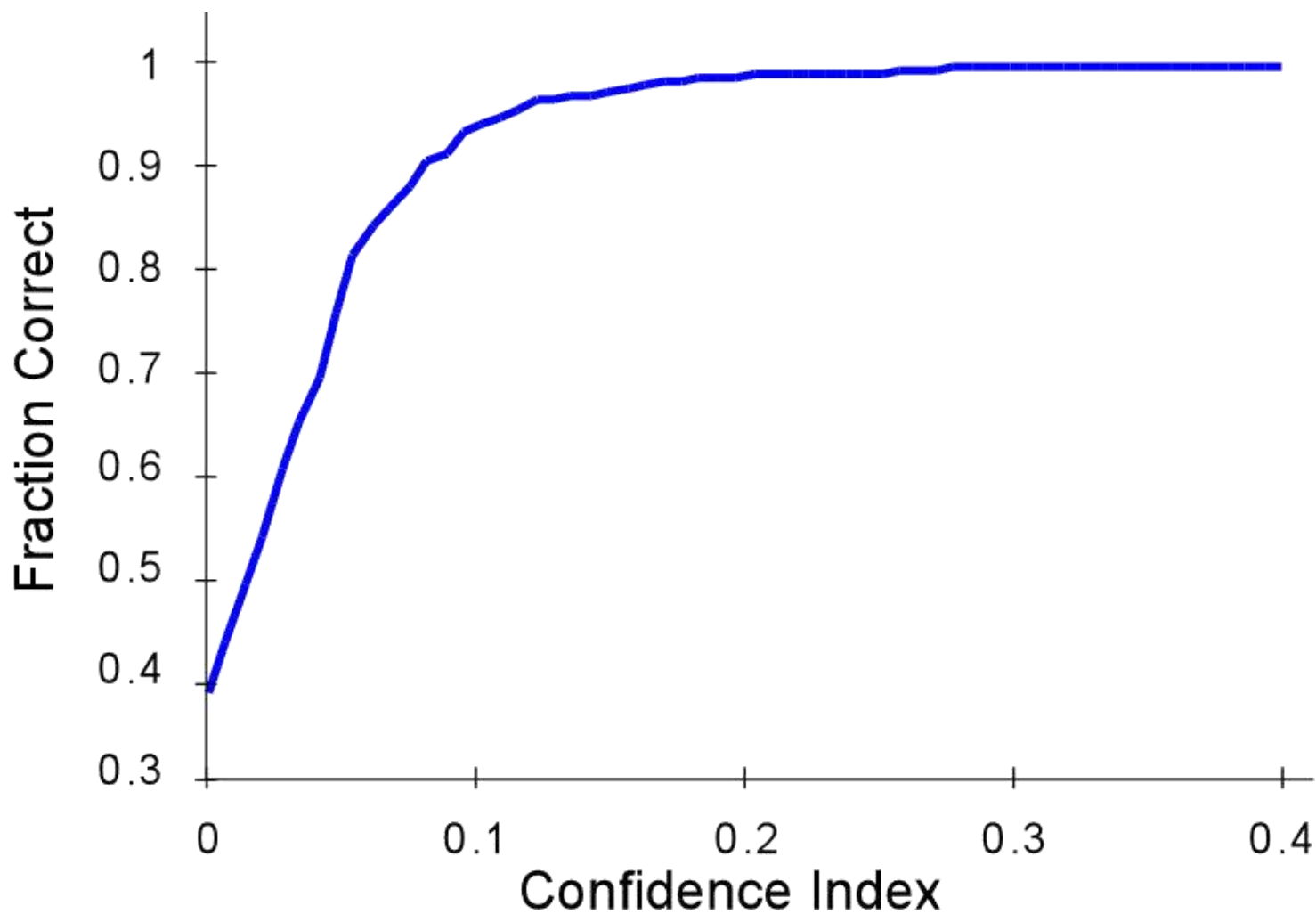
$$CI = \frac{V_1 - V_2}{V_{ideal}}$$

$$V_{ideal} = \frac{n!}{(n-3)! \cdot 3!}$$

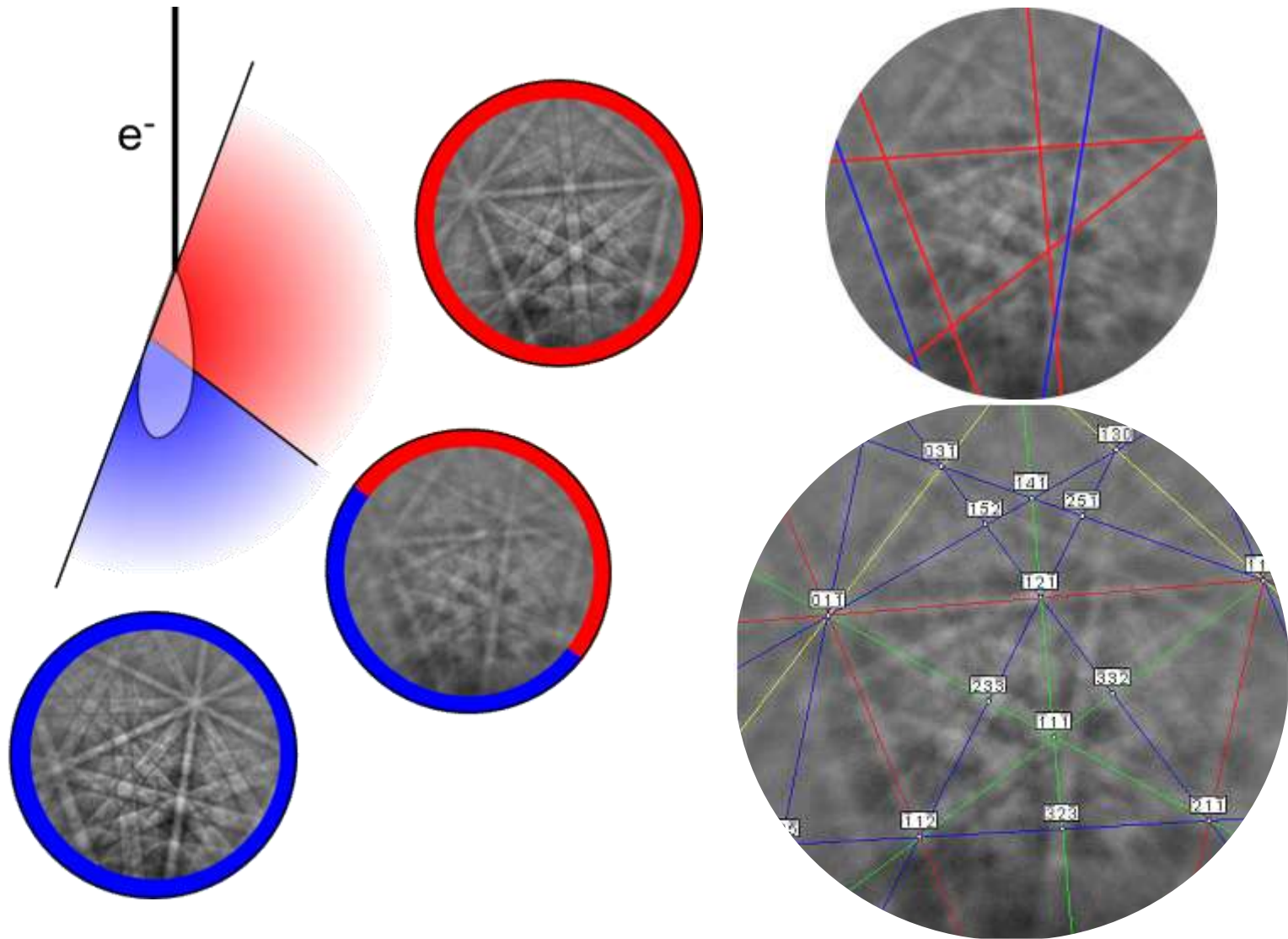
$$CI = \frac{10-1}{10} = 0.9$$

The confidence index, *CI*, measures the uniqueness of an orientation solution relative to the total number of possible votes.

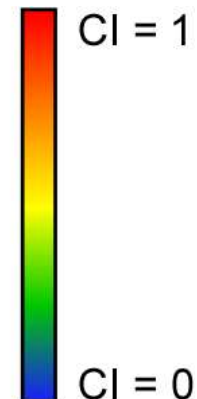
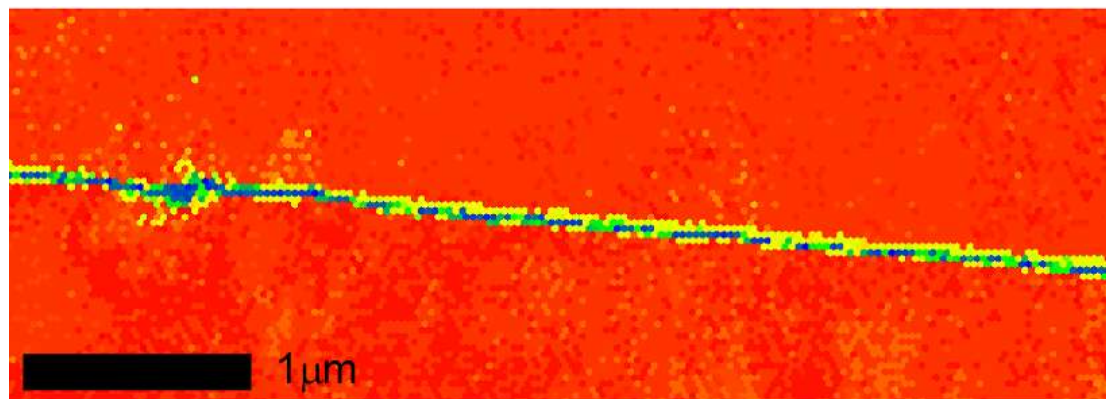
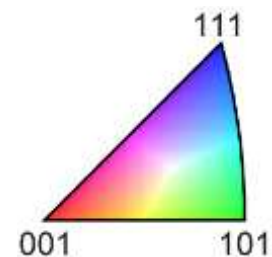
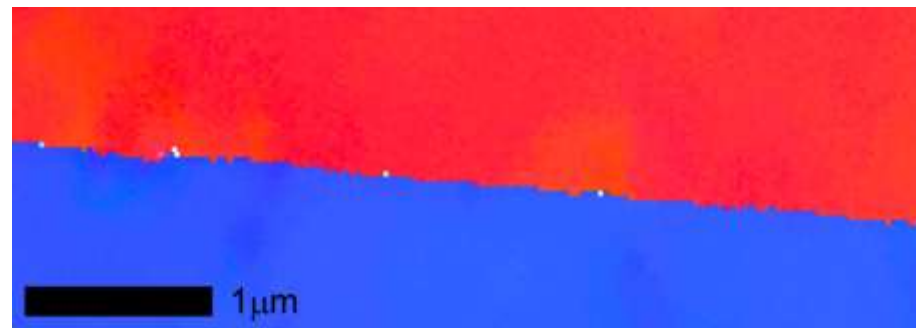
# Indexing: Bands – Confidence Index



# Indexing: Deconvoluting Patterns



# Indexing: Deconvoluting Patterns & CI



# Optimizing EBSD

- **Speed**

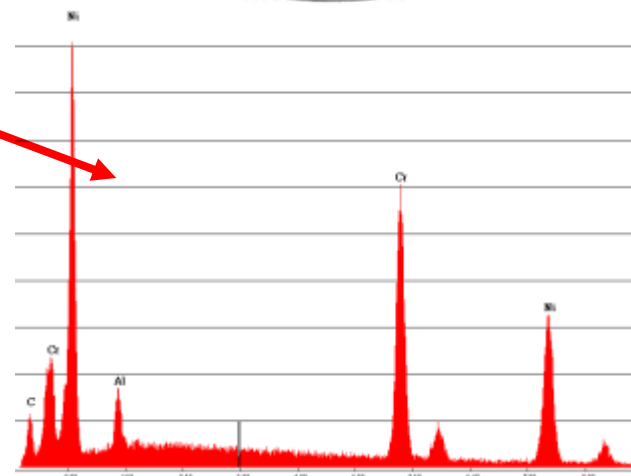
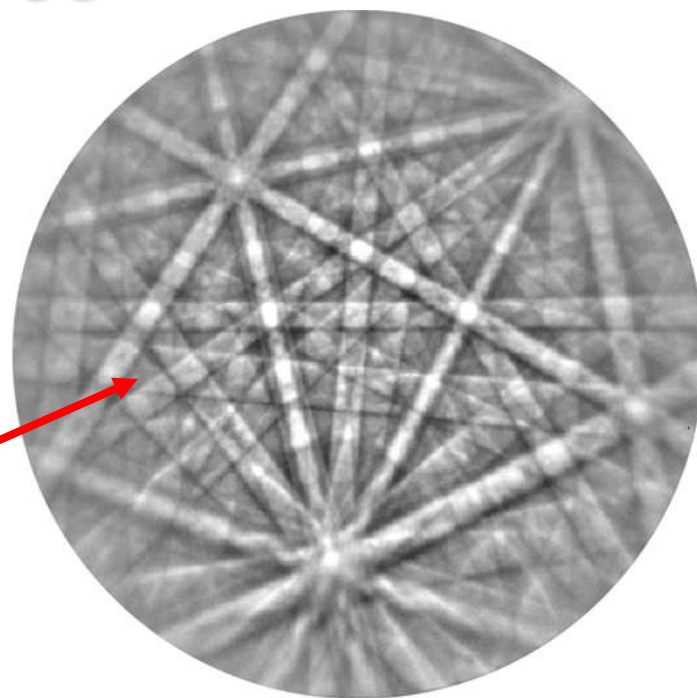
- Pretty patterns vs. scanning patterns
- Camera
  - Binning
  - Gain
- Image Processing
- Hough Settings
  - Convolution Mask
  - dTheta

- **Speed vs. Quality**

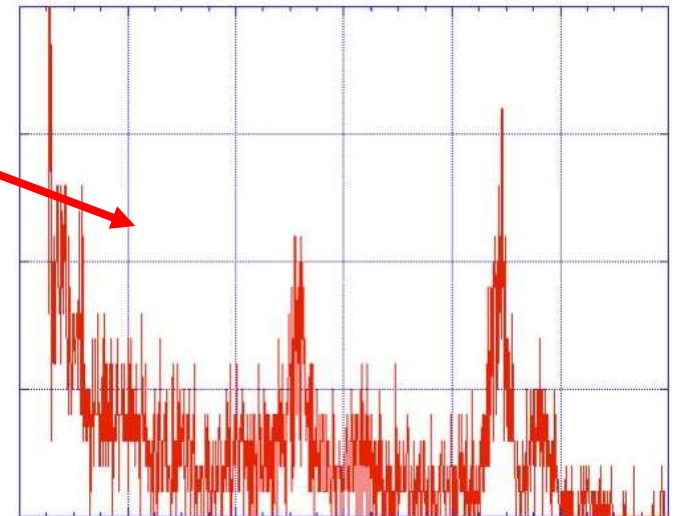
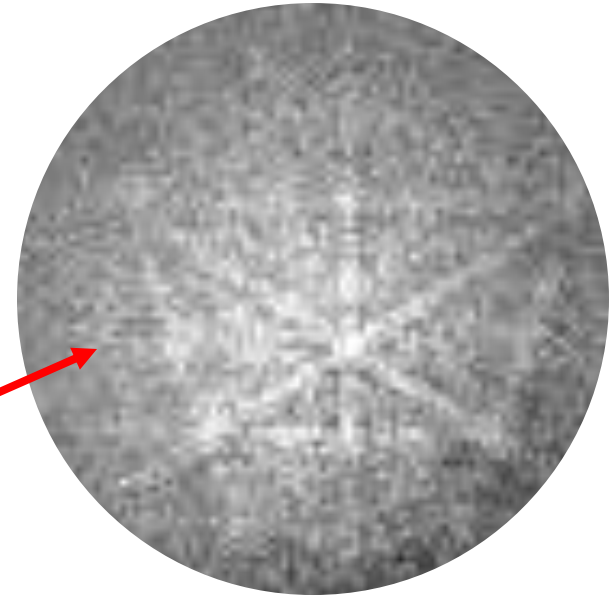
- Indexing success rate
- Precision
- Clean-up



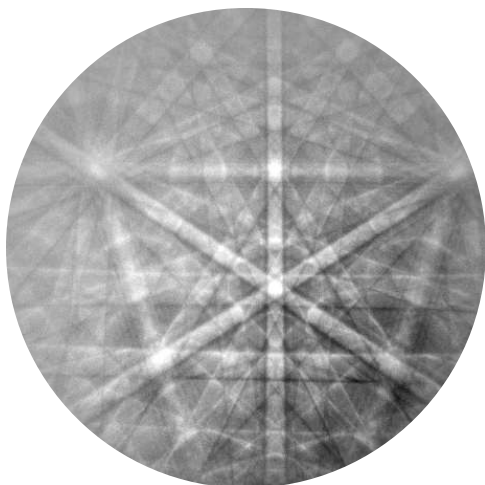
# Pretty Pictures



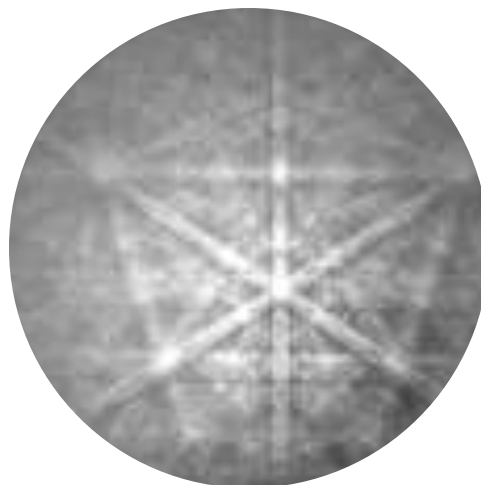
# Reality



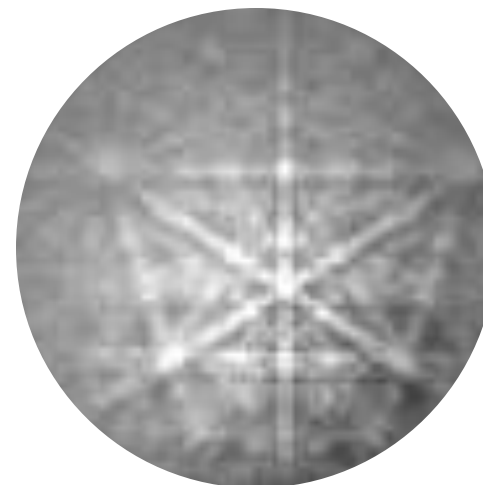
# CCD Camera Binning



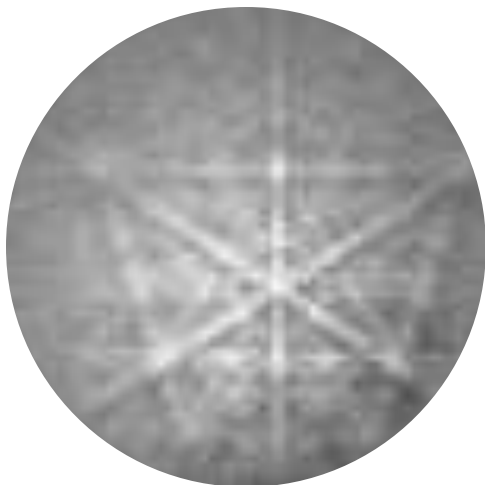
1x1 – 480x480



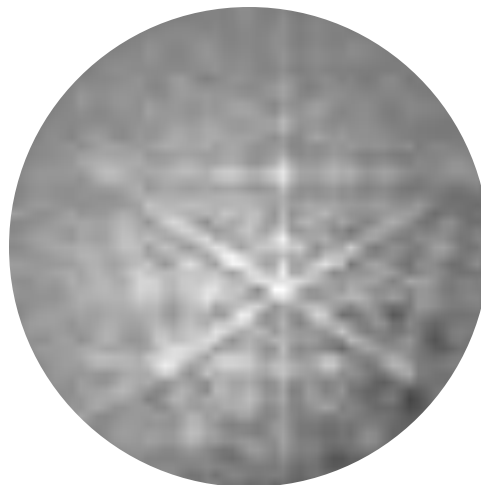
4x4 – 120x120



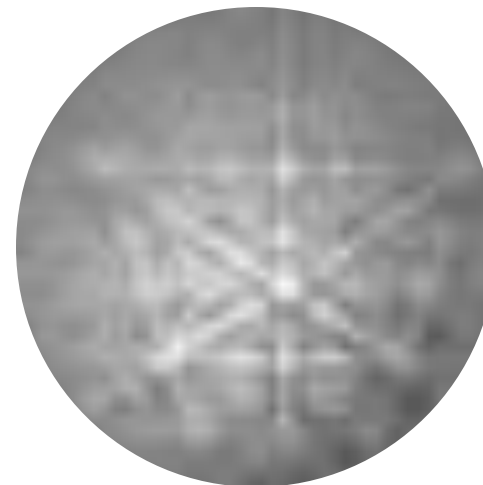
5x5 – 96x96



6x6 – 80x80



8x8 – 60x60



10x10 – 48x48

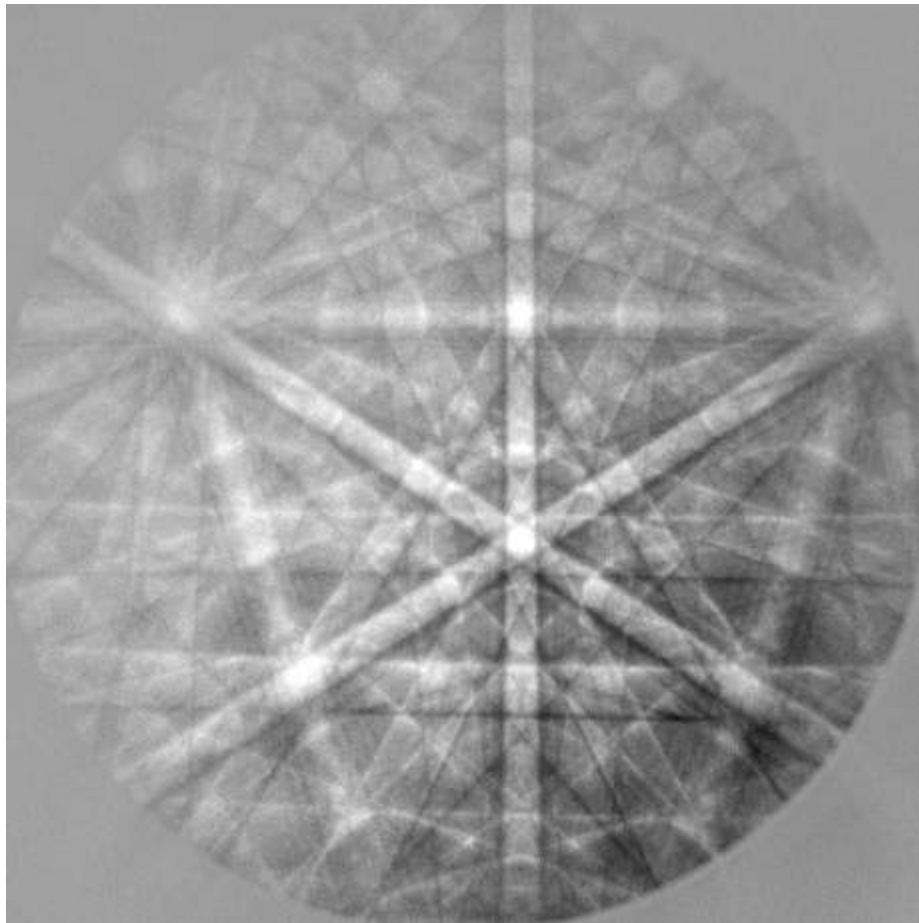
# How Binning Works

2	5	3	1	1	4	1	1
1	1	4	5	1	4	1	1
1	1	1	1	6	5	1	1
1	1	1	1	4	1	3	6
1	1	1	3	3	3	3	1
1	2	3	2	2	1	1	1
3	2	2	2	1	1	1	1
1	1	3	1	1	1	1	1

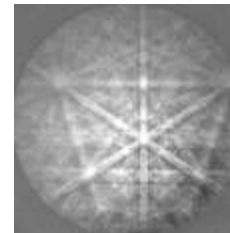
9	13	10	4
4	4	16	11
5	14	9	6
7	8	4	4

Binning reduces the number of pixels in the final image, and increases the effective image intensity.

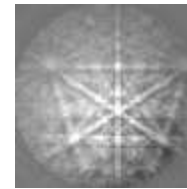
# CCD Camera Binning



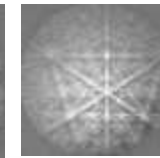
1x1



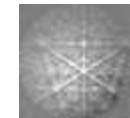
4x4



5x5



6x6



8x8



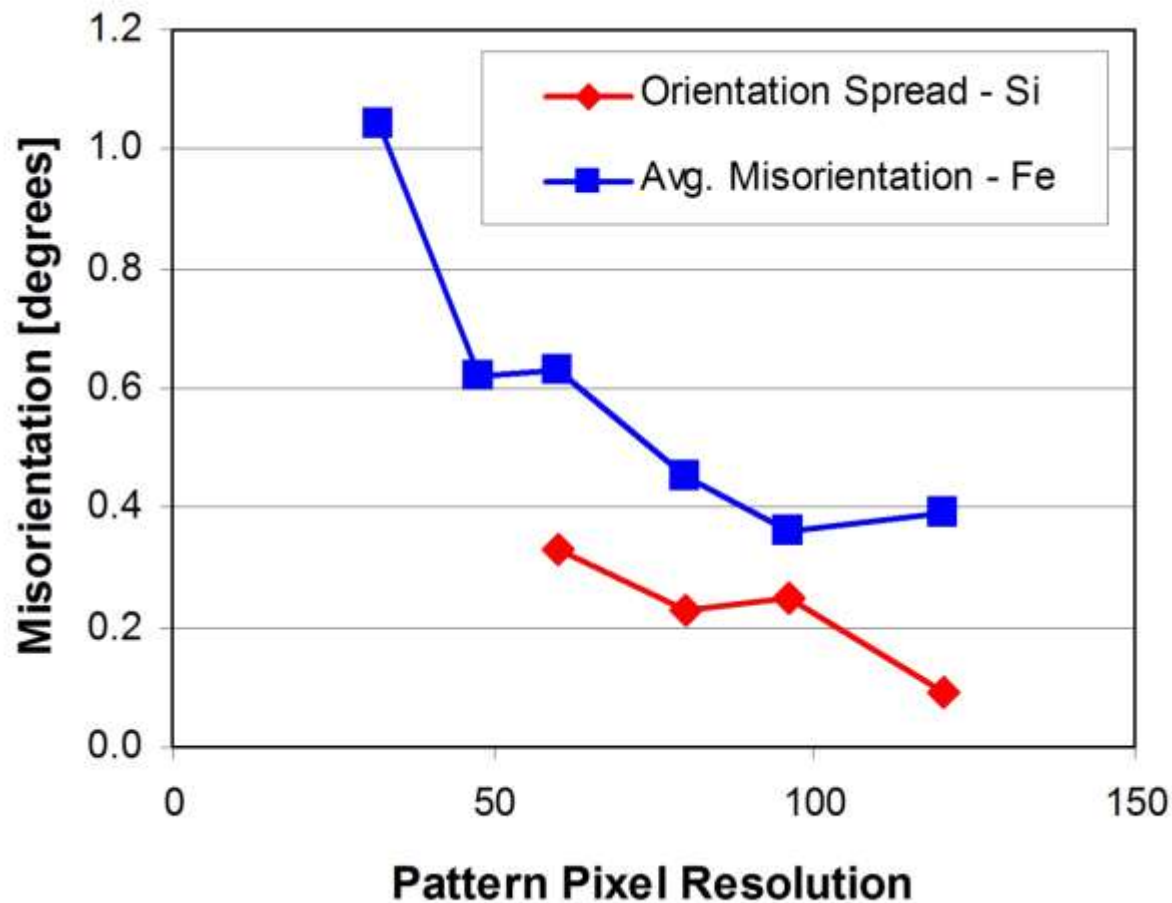
10x10

## Maximum Frame Rate (FPS)

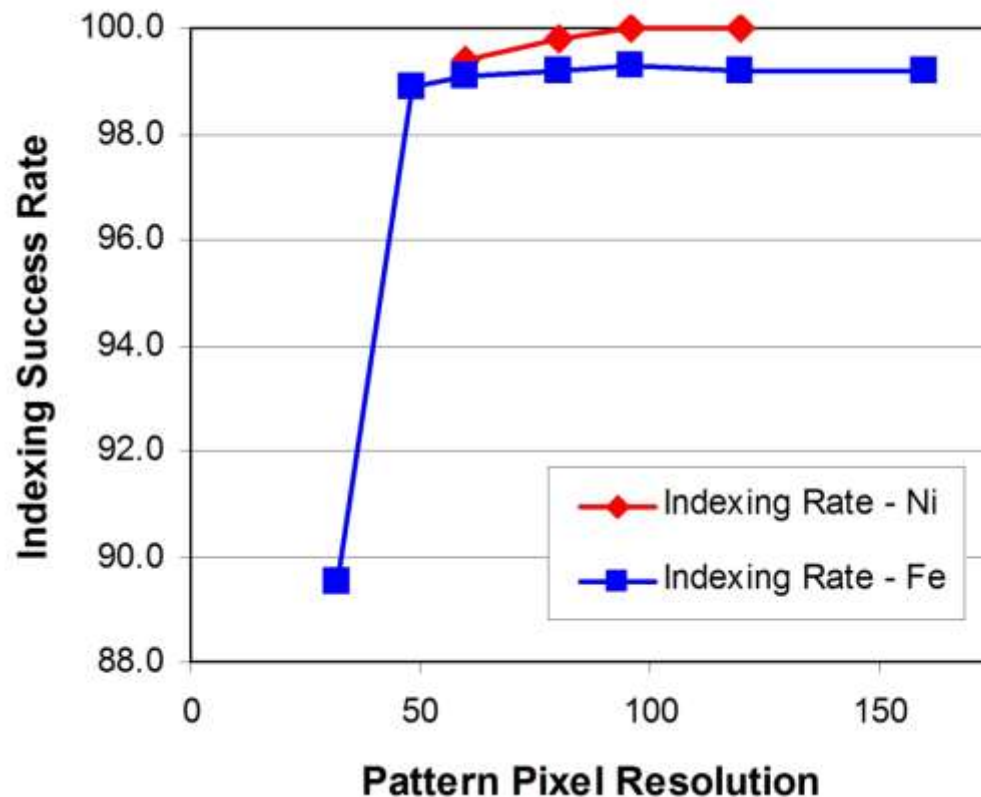
<b>4x4 (128x128)</b>	543
<b>5x5 (96x96)</b>	640
<b>6x6 (80x80)</b>	732
<b>8x8 (60x60)</b>	881
<b>10x10 (48x48)</b>	1000

Faster camera frame rates are possible with binned images.

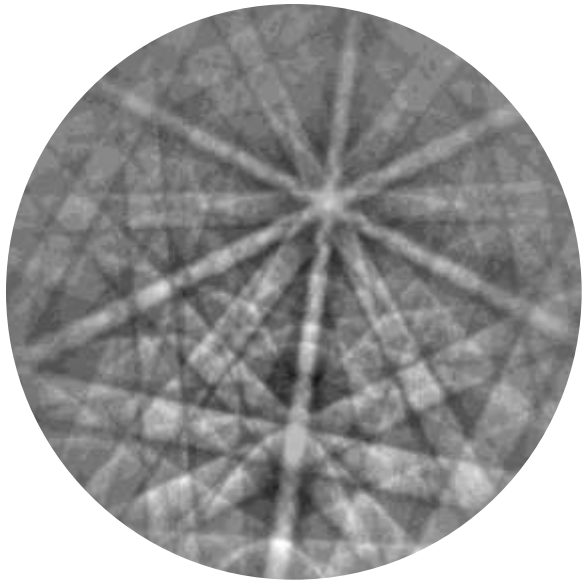
# Precision vs. Binning



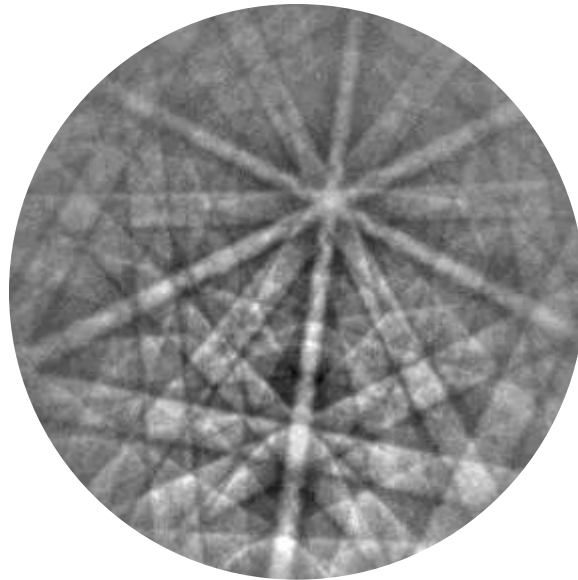
# Indexing Rate vs. Binning



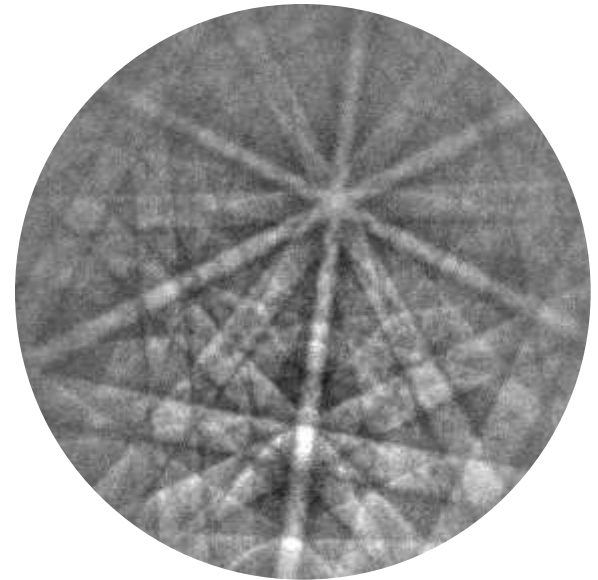
# Camera Gain



**Minimum Gain**  
**2.76 Seconds**



**Mid-Range Gain**  
**0.55 Seconds**

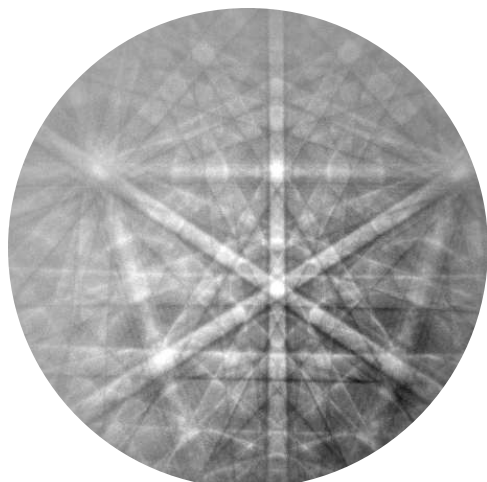


**Maximum Gain**  
**0.15 Seconds**

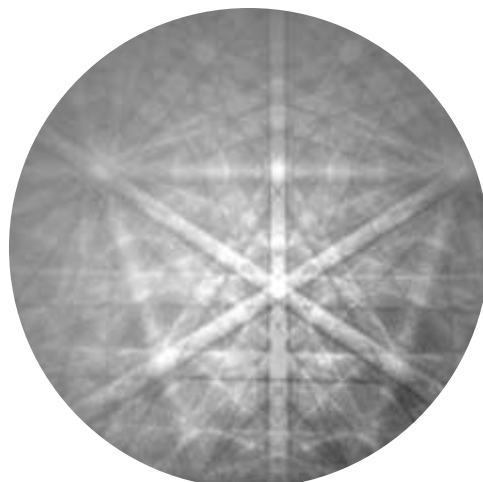
As gain goes up, exposure time goes down, however, the signal to noise ratio decreases.



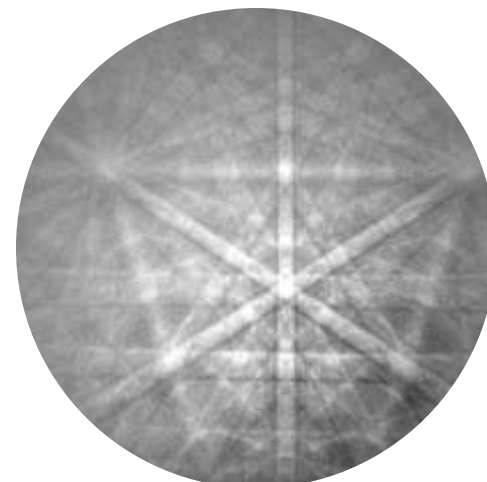
# 1x1 Binning (480x480 Pixels)



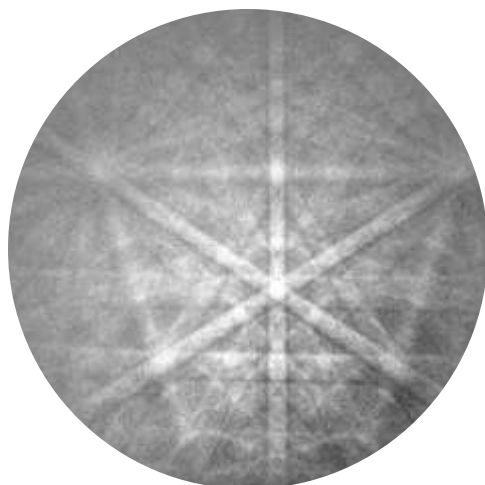
0% Gain



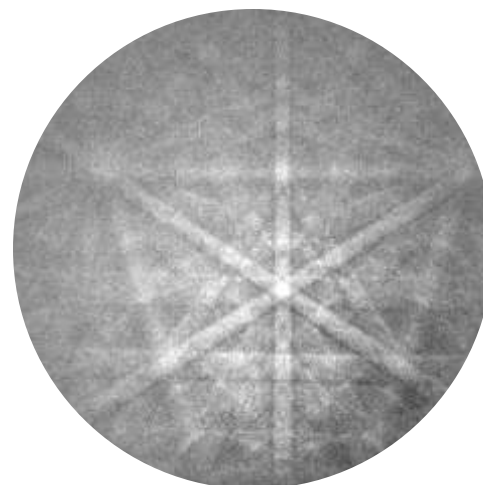
20% Gain



40% Gain

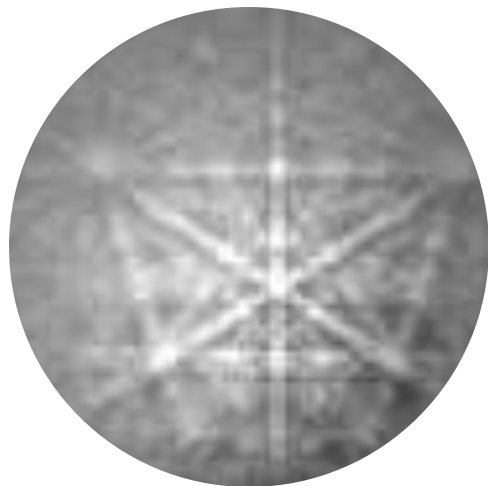


60% Gain



80% Gain

# 5x5 Binning (96x96 Pixels)



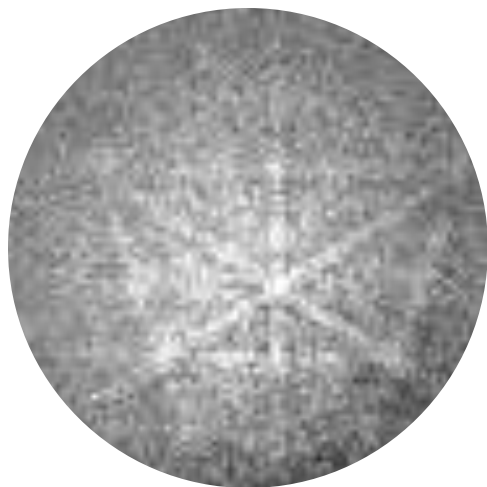
0% Gain



20% Gain



40% Gain

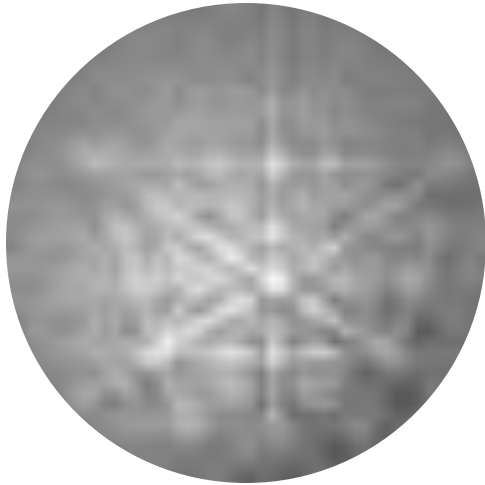


60% Gain

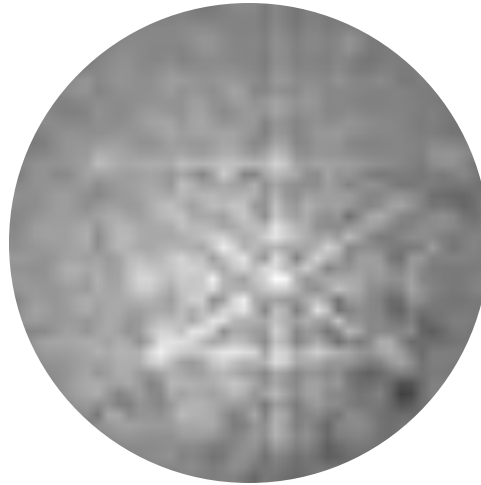


80% Gain

# 10x10 (48x48 Pixels)



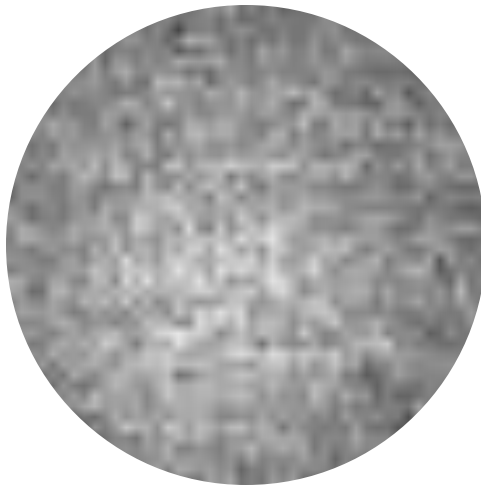
0% Gain



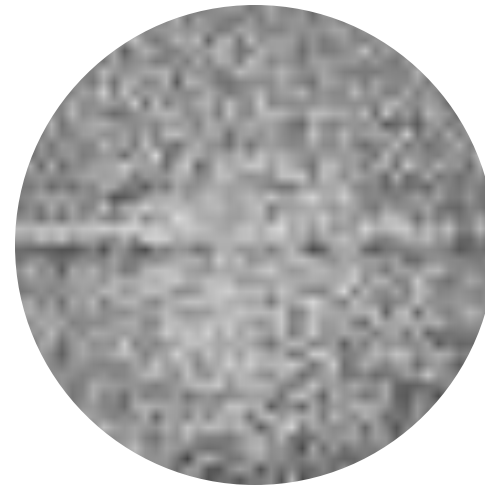
20% Gain



40% Gain



60% Gain

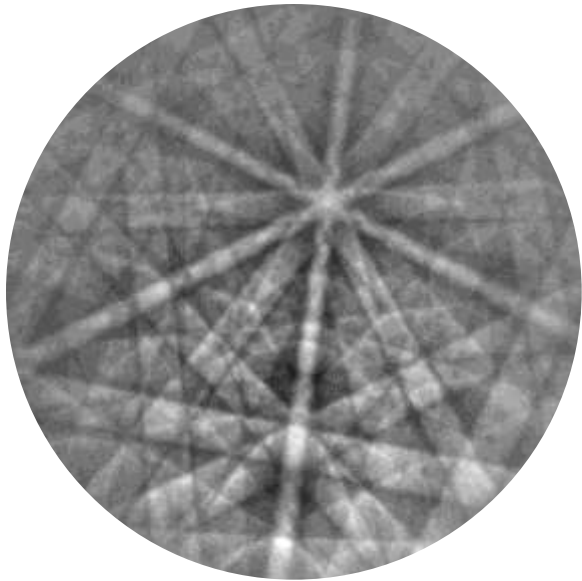


80% Gain

# 5x5 (96x96 Pixels) Spread & Indexing Rate

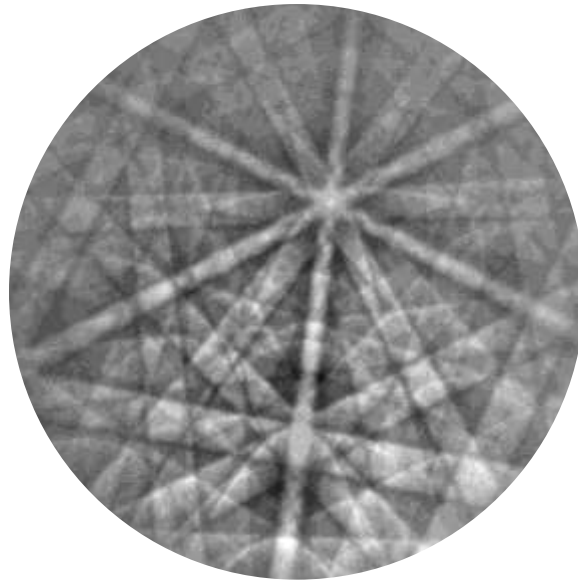
	0% Gain	20% Gain	40% Gain	60% Gain	80% Gain
Orientation Spread	0.12°	0.13°	0.18°	0.26°	0.40°
Indexing Success Rate	100%	100%	97.1%	24.8%	0.8%

# SEM Settings – Beam Current



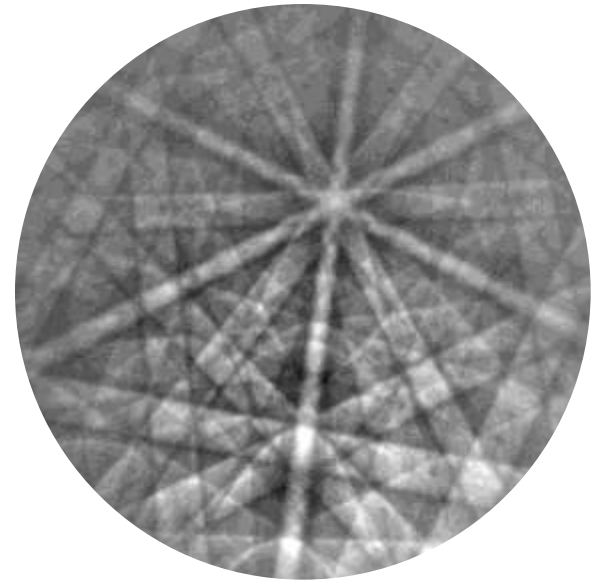
**0.6nA Beam Current**

**4.62 Seconds**



**2.4nA Beam Current**

**1.56 Seconds**

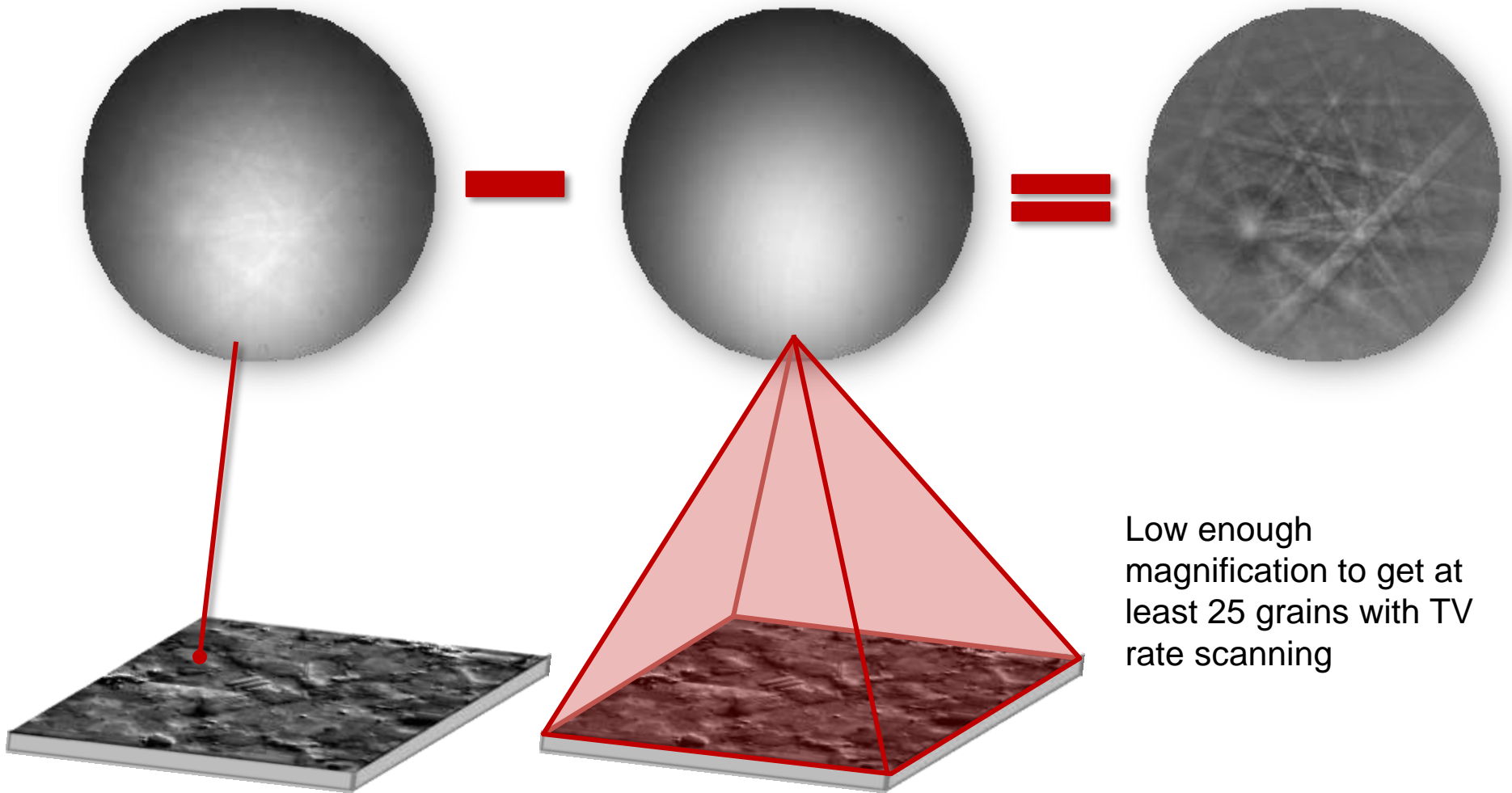


**9.45nA Beam Current**

**0.6 Seconds**

Note however that the gain required for an EBSD pattern is inversely related to the amount of incident beam current.

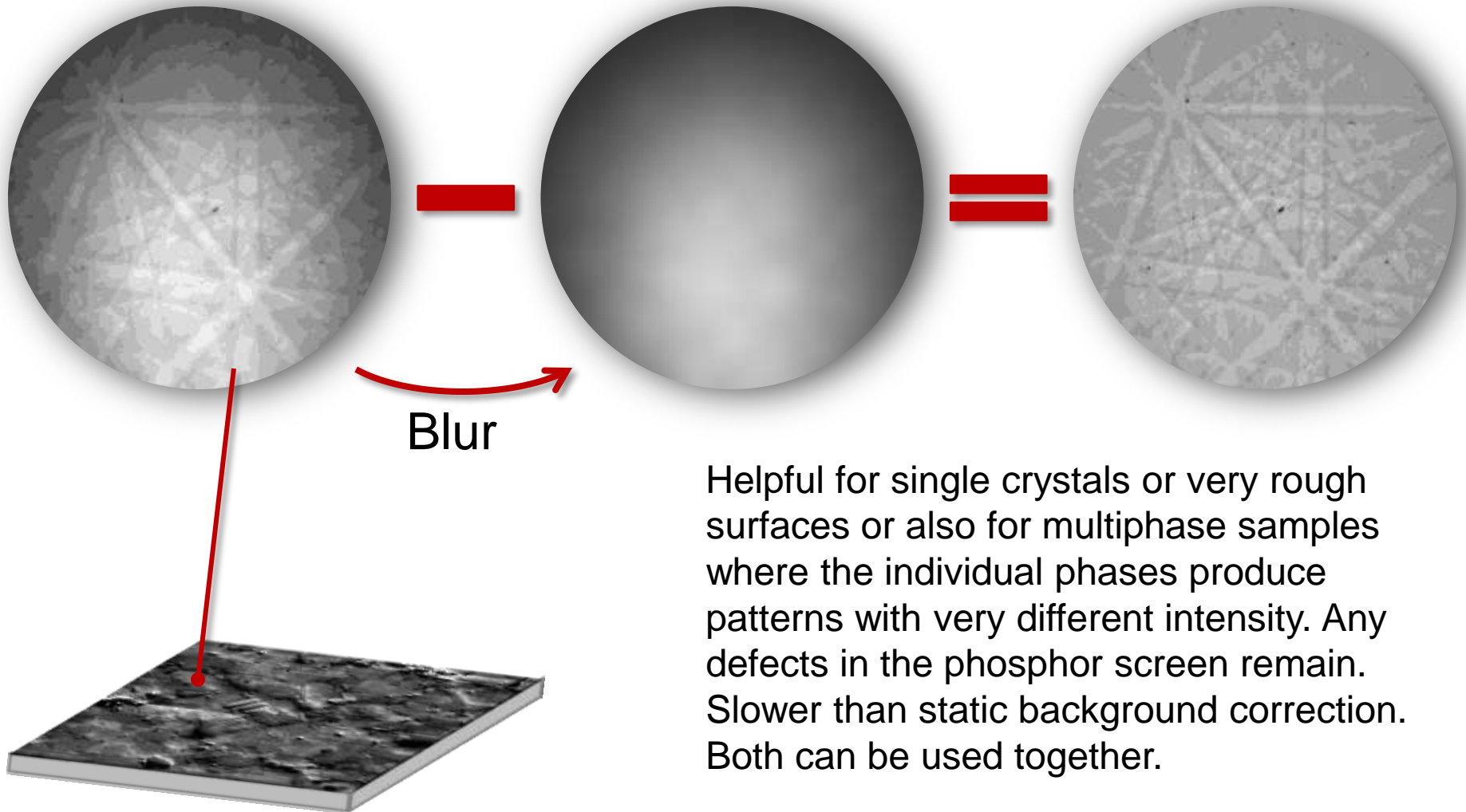
# Image Processing



Low enough magnification to get at least 25 grains with TV rate scanning

We usually have to do some image processing of the patterns to get uniform intensity across the pattern so that the automated band detection algorithms can do a good job.

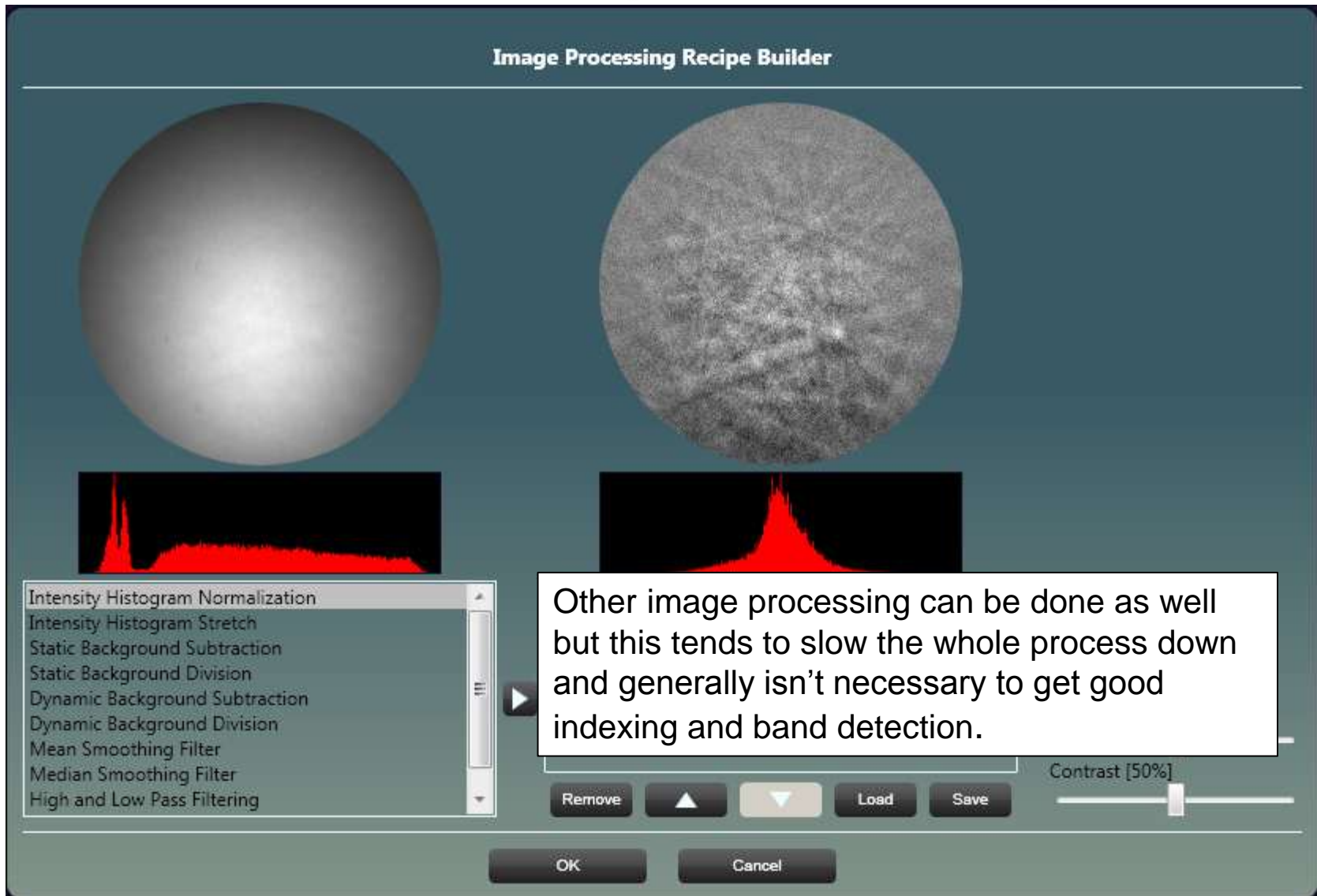
# Dynamic Background Correction



Helpful for single crystals or very rough surfaces or also for multiphase samples where the individual phases produce patterns with very different intensity. Any defects in the phosphor screen remain. Slower than static background correction. Both can be used together.

# Other Image Processing

Image Processing Recipe Builder



Intensity Histogram Normalization  
Intensity Histogram Stretch  
Static Background Subtraction  
Static Background Division  
Dynamic Background Subtraction  
Dynamic Background Division  
Mean Smoothing Filter  
Median Smoothing Filter  
High and Low Pass Filtering

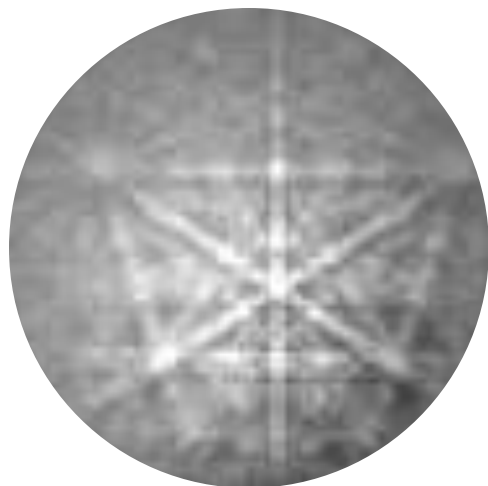
Other image processing can be done as well but this tends to slow the whole process down and generally isn't necessary to get good indexing and band detection.

Remove ▲ ▼ Load Save Contrast [50%]

OK Cancel



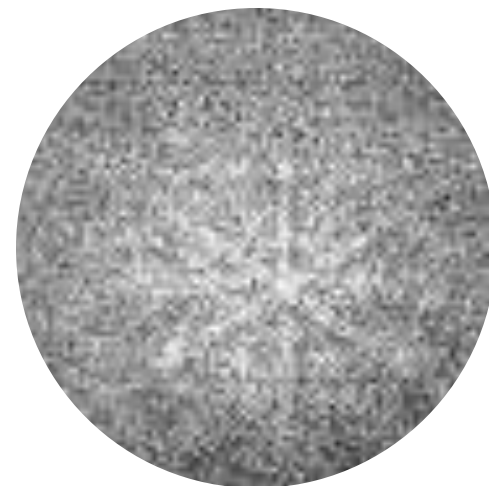
# 5x5 (96x96 Pixels) With Frame Averaging



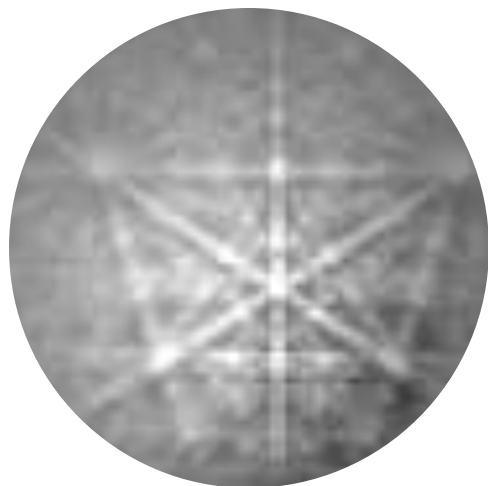
0% Gain



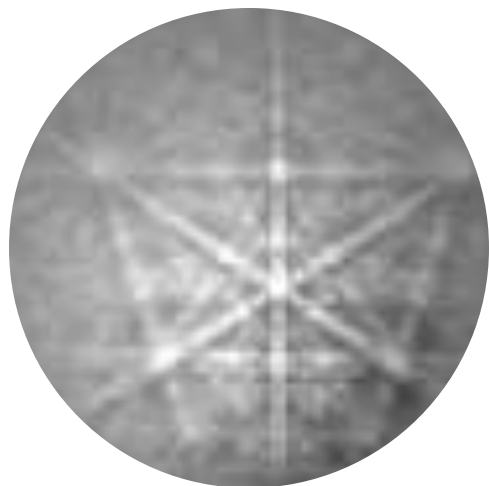
40% Gain



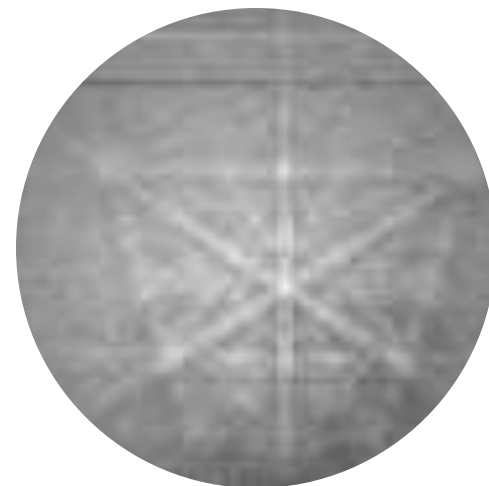
80% Gain



0% Gain Avg

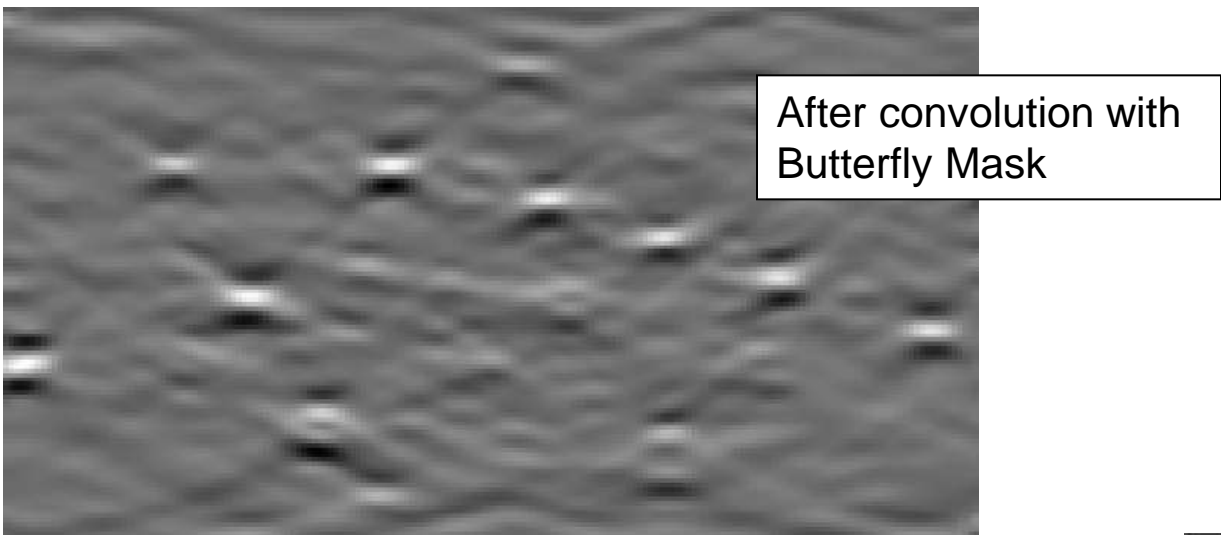
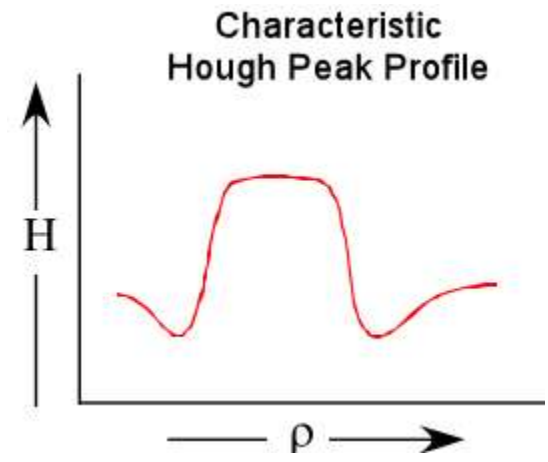
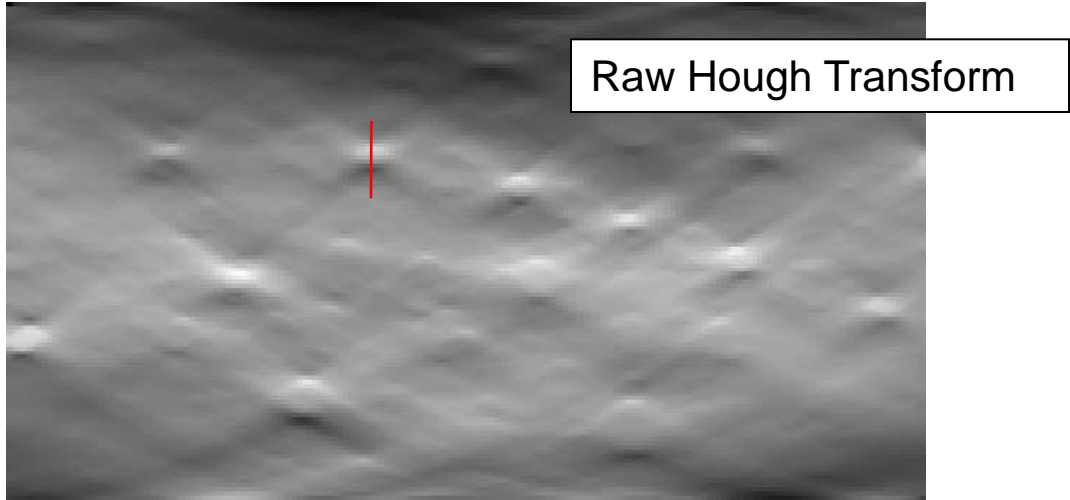


40% Gain Avg

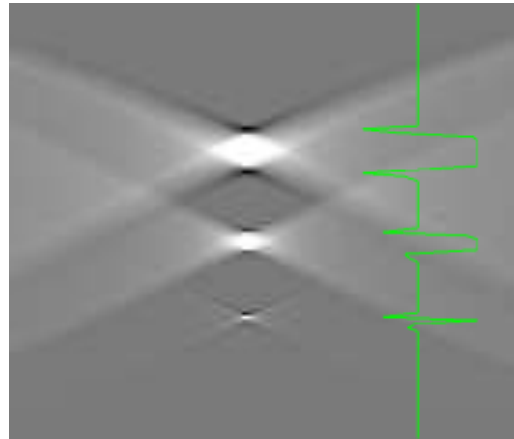


80% Gain Avg

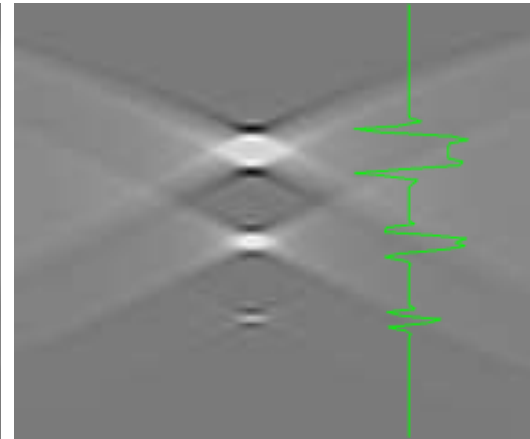
# Hough Transform



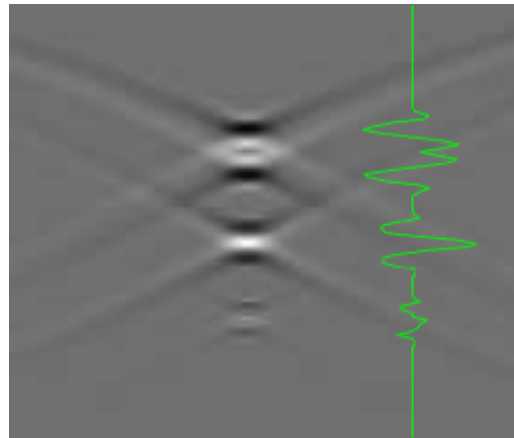
# Hough Transform



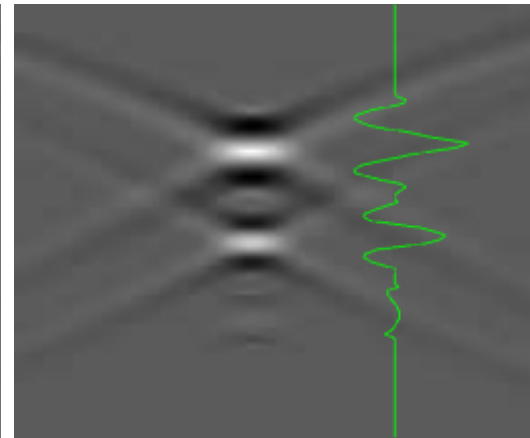
No Mask



Small Mask (5x5)



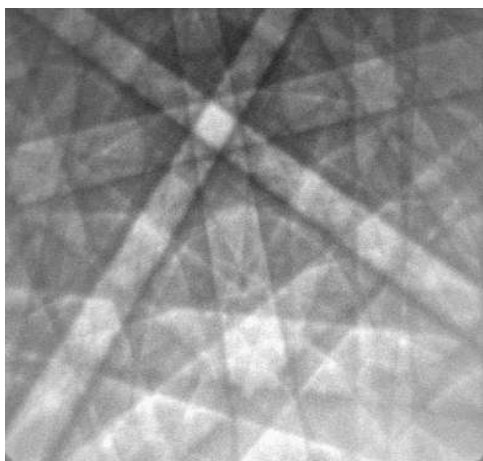
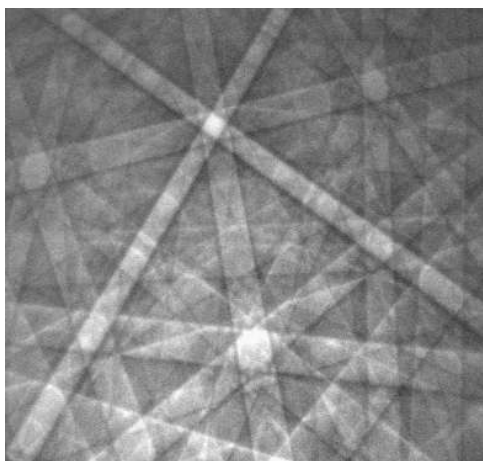
Medium Mask (9x9)



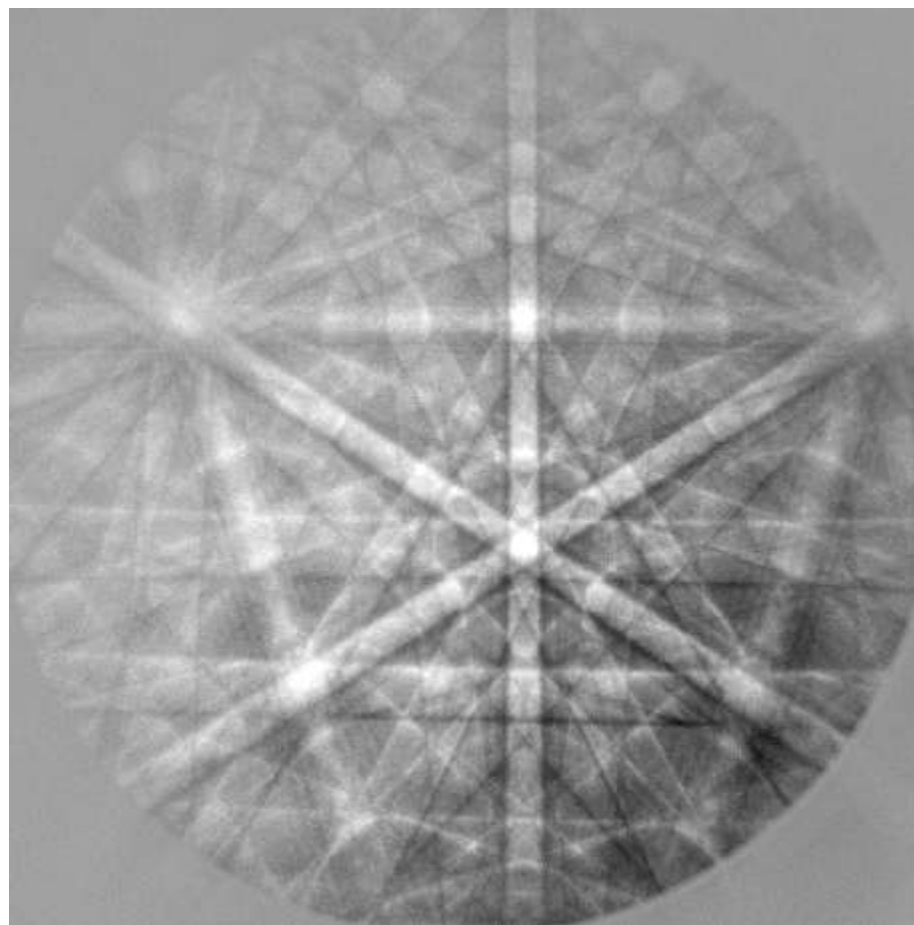
Large Mask (13x13)

One weakness of the butterfly mask is that it enhances peaks from bands a given width. It can be difficult to find the right size of mask when there is considerable variance in the width of the bands as is often the case in low-symmetry materials. This also biases any band width measurements.

# Mask Size & Band Width



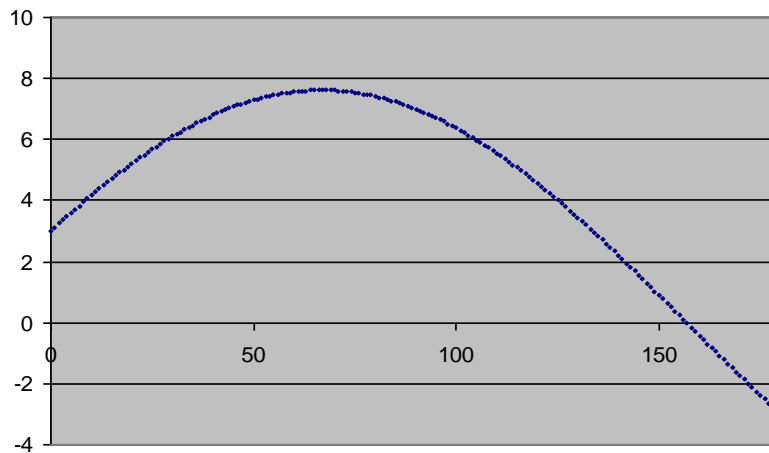
Voltage



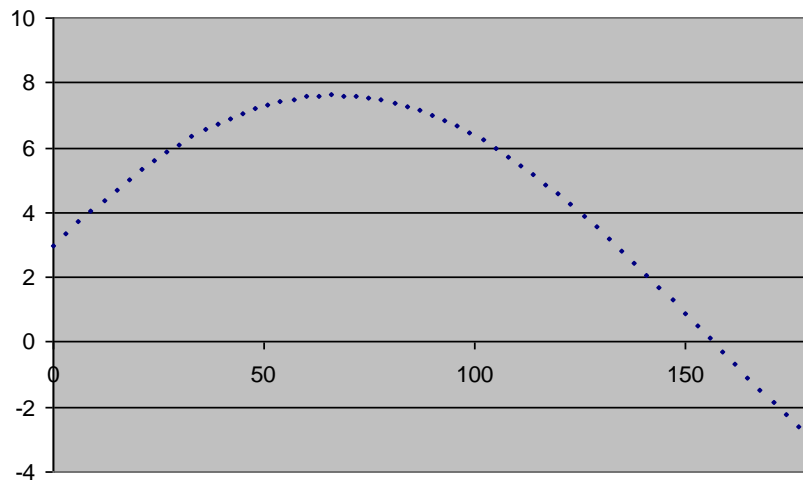
Binning

# Hough Theta Step

1°



3°



# 5x5 (96x96 Pixels) Average Orientation Spread

	0% Gain	20% Gain	40% Gain	59% Gain Max FPS	60% Gain	80% Gain
0.5° $\theta$	0.12°	0.13°	0.18°	0.27°	0.26°	0.40°
1° $\theta$	0.16°	0.17°	0.20°	0.27°	0.27°	0.40°
2° $\theta$	0.17°	0.20°	0.29°	0.38°	0.38°	0.48°
3° $\theta$	0.13°	0.13°	0.22°	0.33°	0.34°	0.57°

Less than 60% gain is necessary for maximum FPS

# 10x10 (48x48 Pixels) Orientation Spread

	0% Gain	20% Gain	40% Gain	41% Gain Max FPS	60% Gain	80% Gain
0.5° $\theta$	0.21°	0.26°	0.41°	0.42°	0.93°	NA
1° $\theta$	0.21°	0.27°	0.48°	0.49°	0.99°	NA
2° $\theta$	0.26°	0.28°	0.50°	0.46°	1.09°	NA
3° $\theta$	0.29°	0.44°	0.78°	0.80°	1.55°	NA

Near 40% gain is necessary for maximum FPS

# 10X10 Points Correctly Indexed

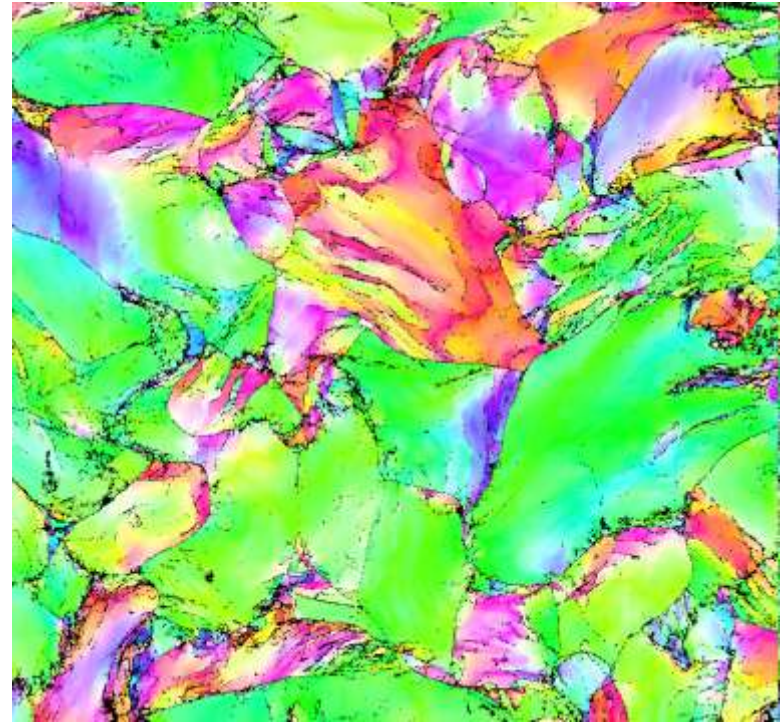
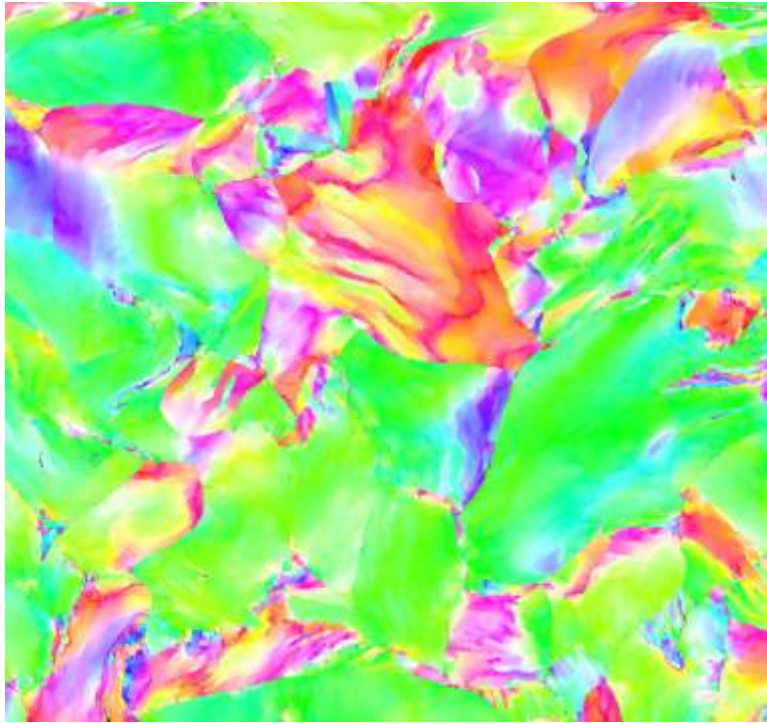
	0% Gain	20% Gain	40% Gain	41% Gain Max FPS	60% Gain	80% Gain
0.5° $\theta$	100%	100%	96.7%	97.4%	28.7%	0.9%
1° $\theta$	100%	100%	97.1%	95.3%	24.8%	0.8%
2° $\theta$	100%	99.9%	93.1%	92.2%	12.1%	0.7%
3° $\theta$	100%	99.1%	72.7%	75.2%	3.9%	0.8%



# Summary and Conclusions

- High frame rates are achieved through camera binning and camera gain settings.
- Hough settings must be optimized for a given camera setting. Set up Hough parameters for patterns obtained under scan conditions, not for patterns obtained in a high quality pattern mode.
- There is a trade-off between speed and orientation precision but not as severe as one might assume. Trust the Hough!
- Evaluate the quality of the results on the quality of the scan data not how the pattern looks.
- Note that these results have been shown in diamond cubic silicon and FCC nickel.
- Non-cubic patterns may require higher resolution, to achieve accurate band detection.
- Recently submitted a manuscript on precision with a particular focus on precision at grain boundaries.

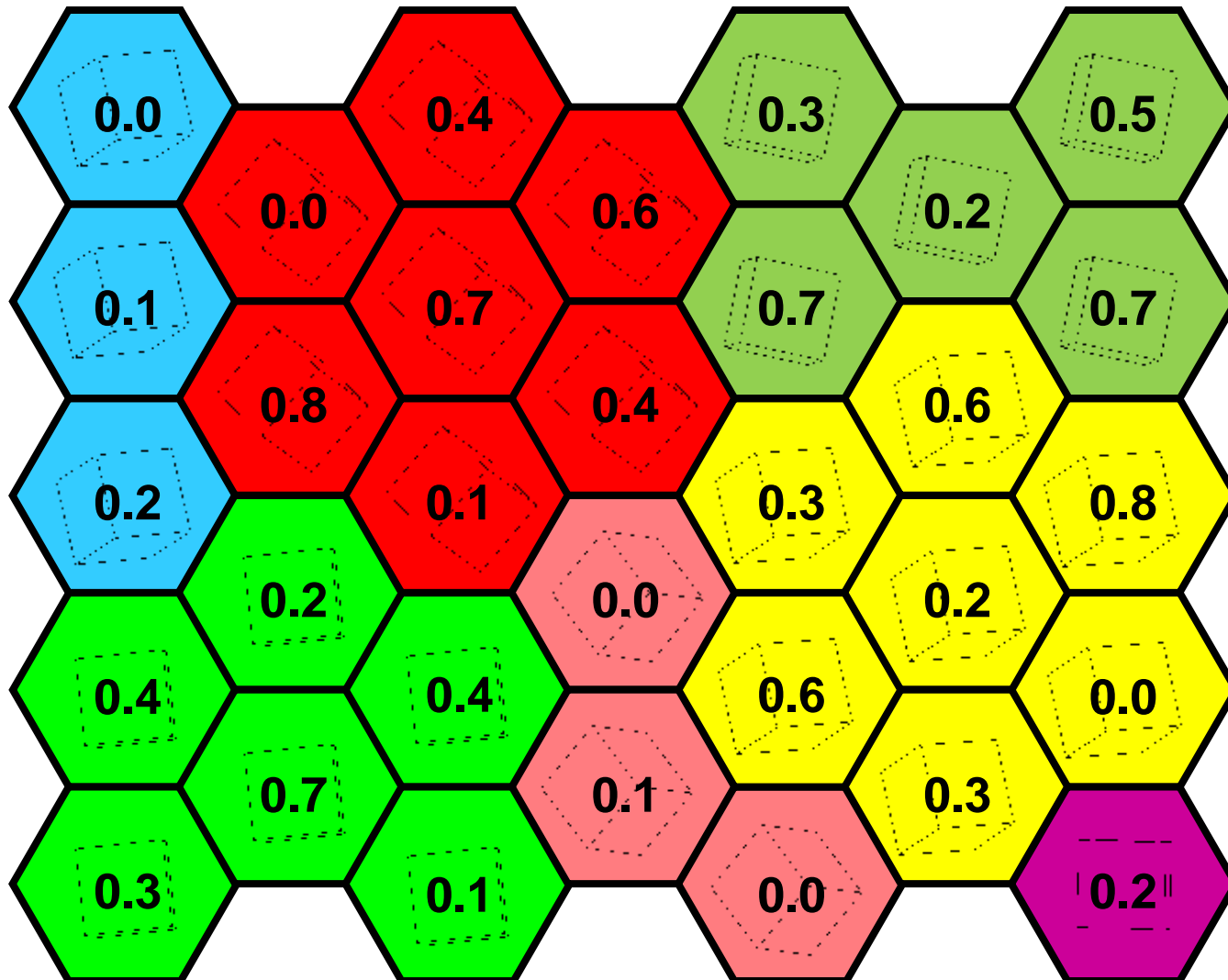
# Confidence Index Partitioning or Filtering



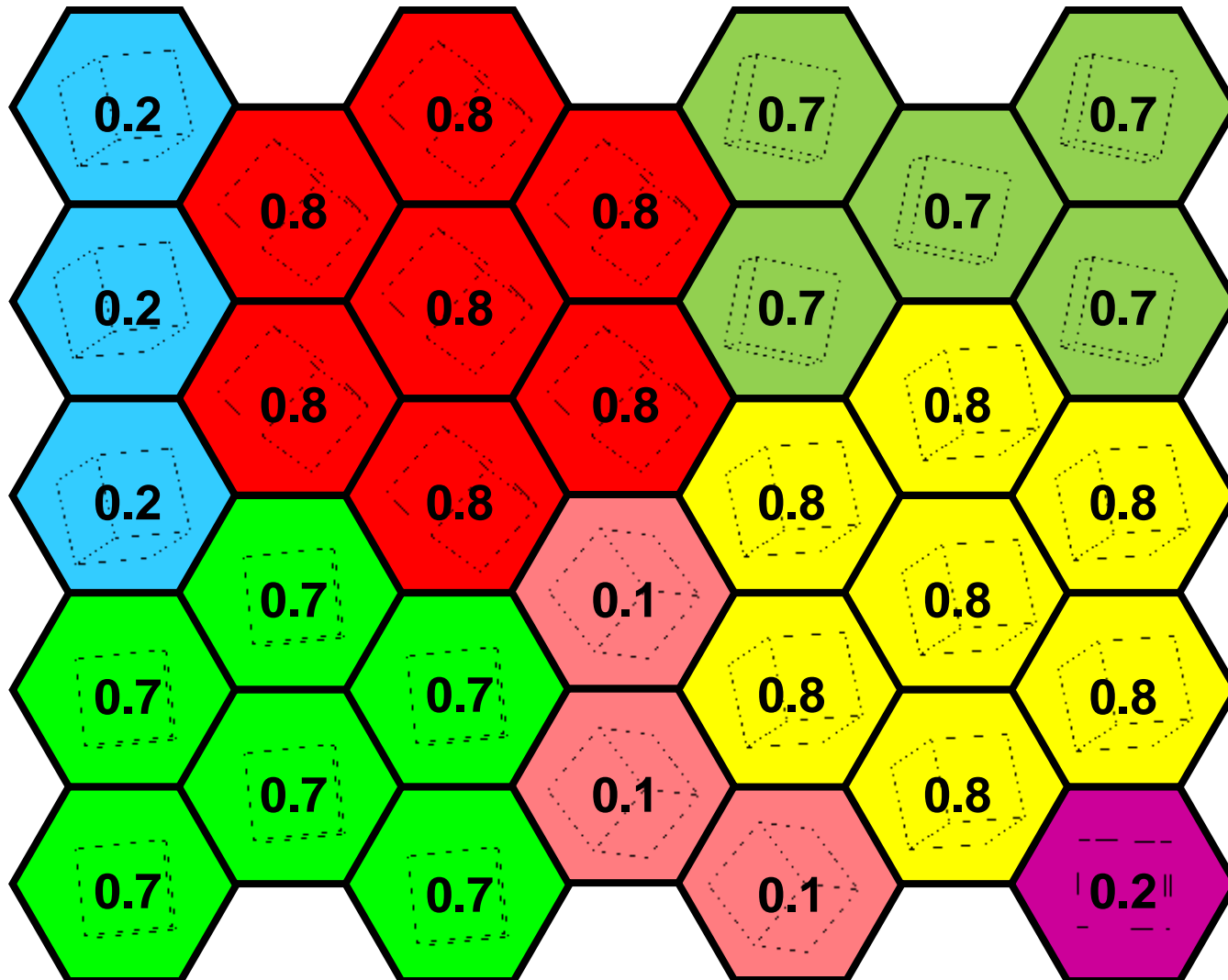
CI>0.3

We can filter the data based on CI or Fit or other metrics to mask data we do not have confidence in from our subsequent analysis. However, some of these low CI points are actually valid orientations. How can we recover them?

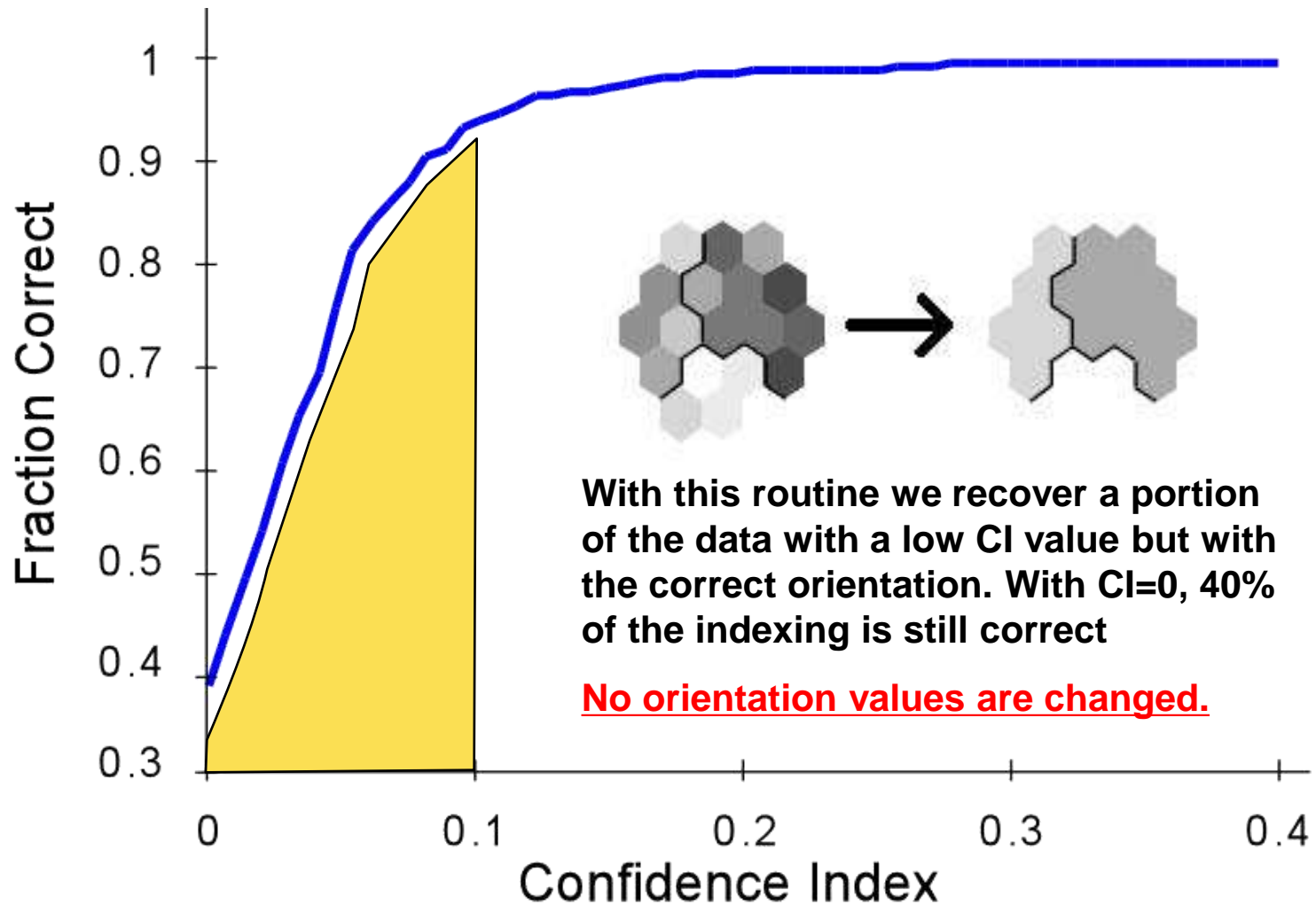
# Initial Confidence Index Distribution



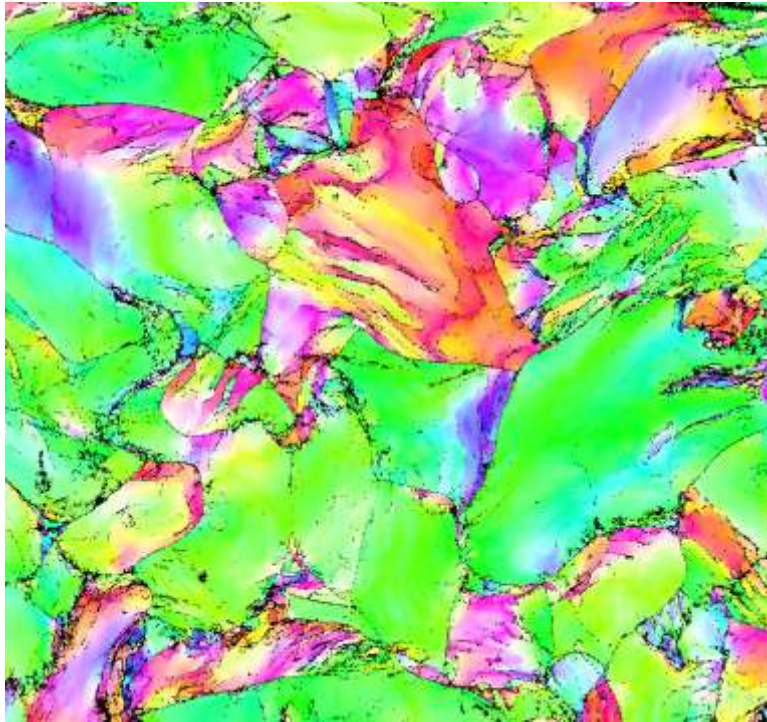
# Confidence Index Standardization



# Confidence Index Standardization

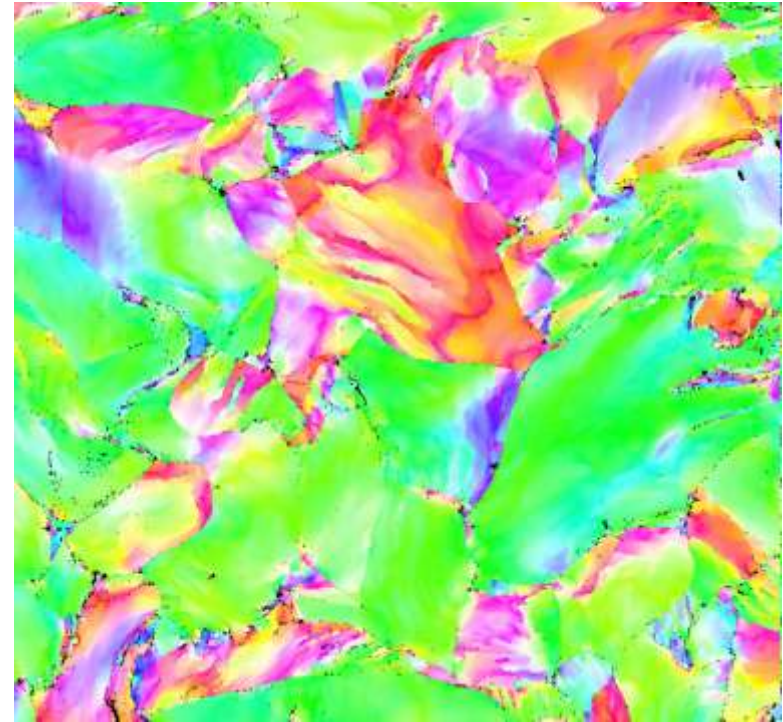


# Confidence Index Standardization



As-collected

CI>0.3 91.8%



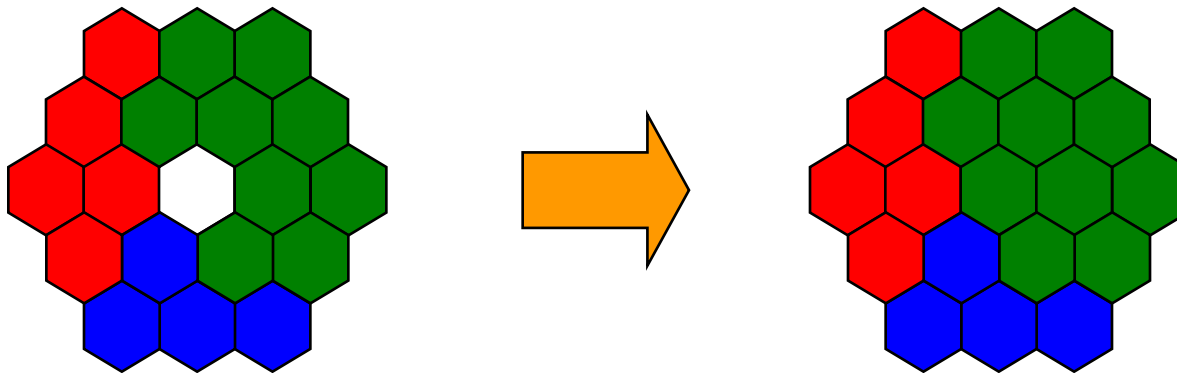
Grain CI Standardization

CI>0.3 98.7%

The confidence index standardization (CIS) routine identifies the maximum fraction of correctly indexed points. Use it always.

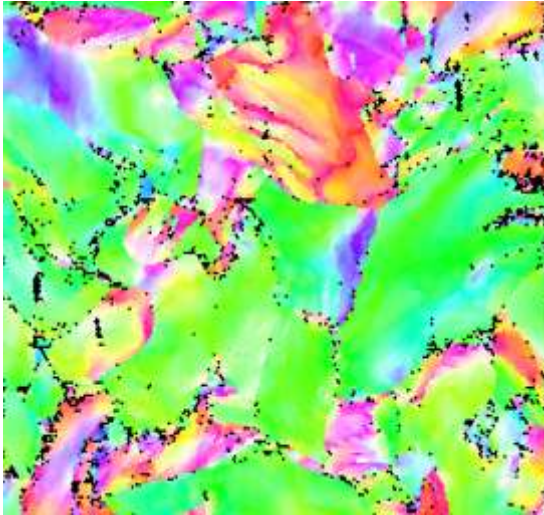
# Grain Dilation

This clean up method is an iterative method. The routine only acts on points that do not belong to any grains; yet have neighboring points, which do belong to grains. A point may not belong to any grain due to the point either not being indexed or due to it belonging to a grain group having fewer members than the Minimum Grain Size. If the majority of neighbors of a particular point belong to the same grain then the orientation of the particular point is changed to match that of the majority grain - otherwise the orientation is randomly changed to match any of the neighboring points, which belong to grains. This process is repeated until each point in the data set becomes a member of a grain. (Alternatively, the user may set the code to only perform a single iteration.) In the schematic below, in the left hand figure the data point in white is not part of any grains. After dilation it's orientation is changed to match that of the neighboring member of the green grain with the highest CI.

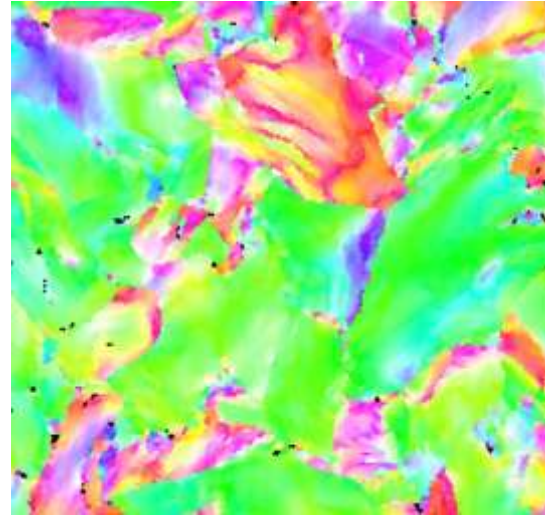


# Grain Dilation

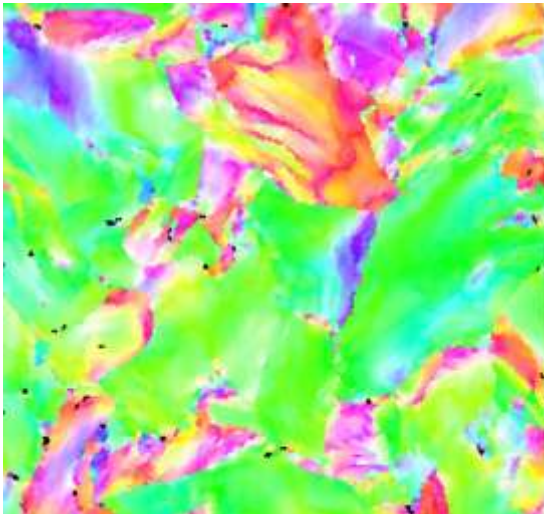
CI > 0.1 - 95.5%



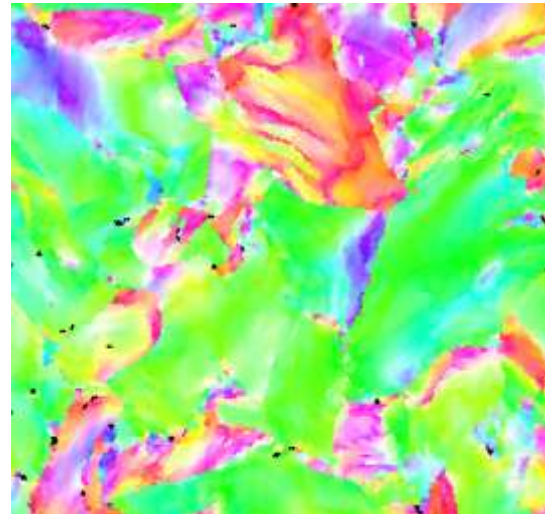
1 Iteration -  
99.6% **+4.1%**



2 Iterations -  
99.7% **+4.2%**



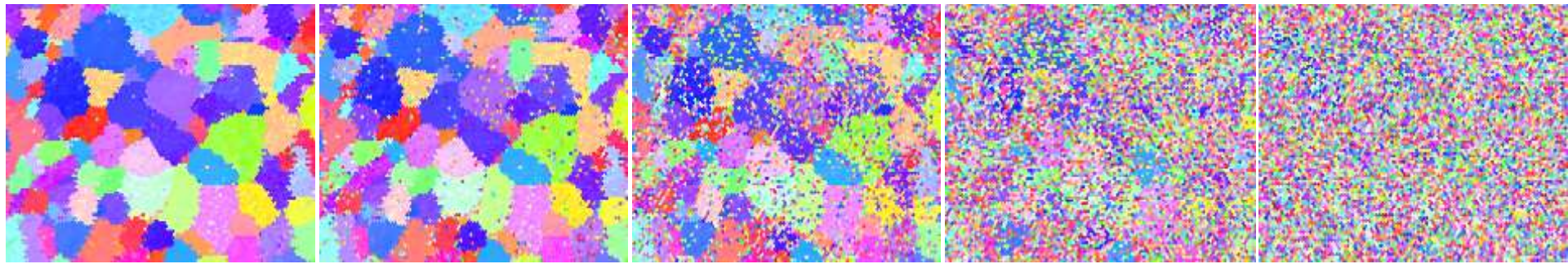
3 Iterations -  
99.7% **+4.2%**





# Grain Dilation

Grain Dilation can actually work too well. Plausible microstructures can be generated out of nearly random data.



**Full Grain Dilation Cleanup**



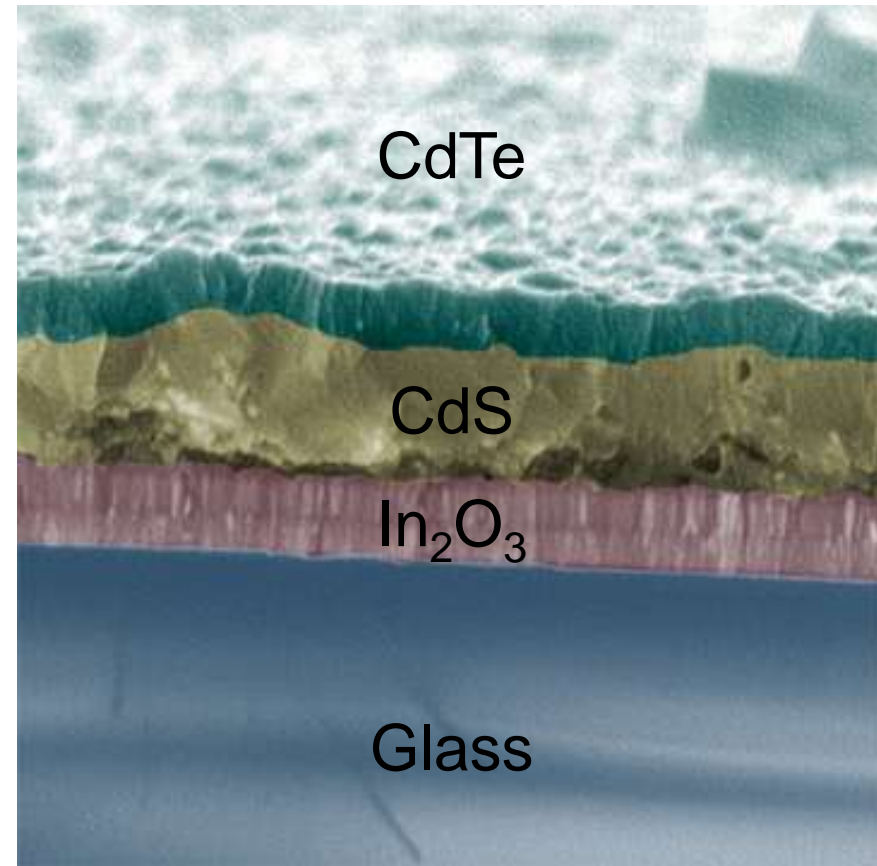
# Summary

- There are a wide range of tools available to identify the correctly indexed points and remove noise within the data maps.
- The Grain CI Standardization feature should always be used in conjunction with a CI filter to determine indexing quality.
- Cleanup routines should be used with care to improve data quality and results and not introduce artificial data.
- It is good practice to report the cleanup used during analysis and the fraction of points cleaned.
- Start with the best data possible!

# Case Study

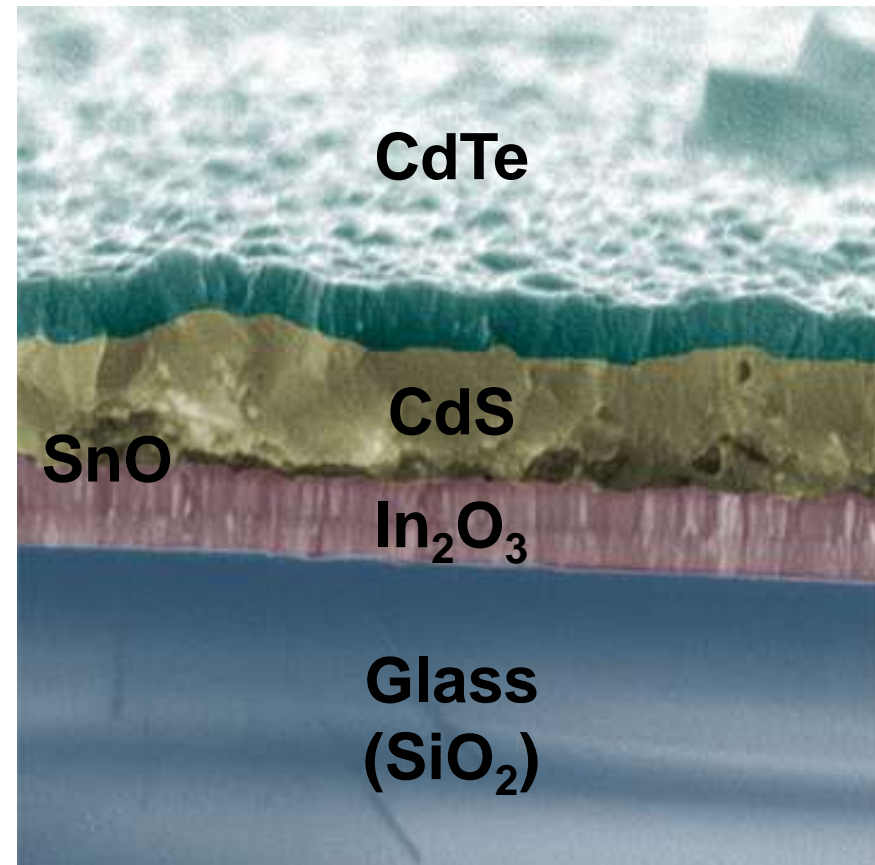
# Photovoltaic Materials

- Energy demand and prices are expected to increase and the demand for alternative sources is high.
- Photovoltaic thin film solar cells provide a commercially viable technology. Thin film photovoltaic devices use only a small fraction of the raw material when compared to traditional cells.
- Polycrystalline CdTe and CIGS thin films have higher efficiencies than single crystal devices.
- The performance of these thin films is influenced by the crystallographic structure, grain boundary character, and grain size.



# Photovoltaic Materials

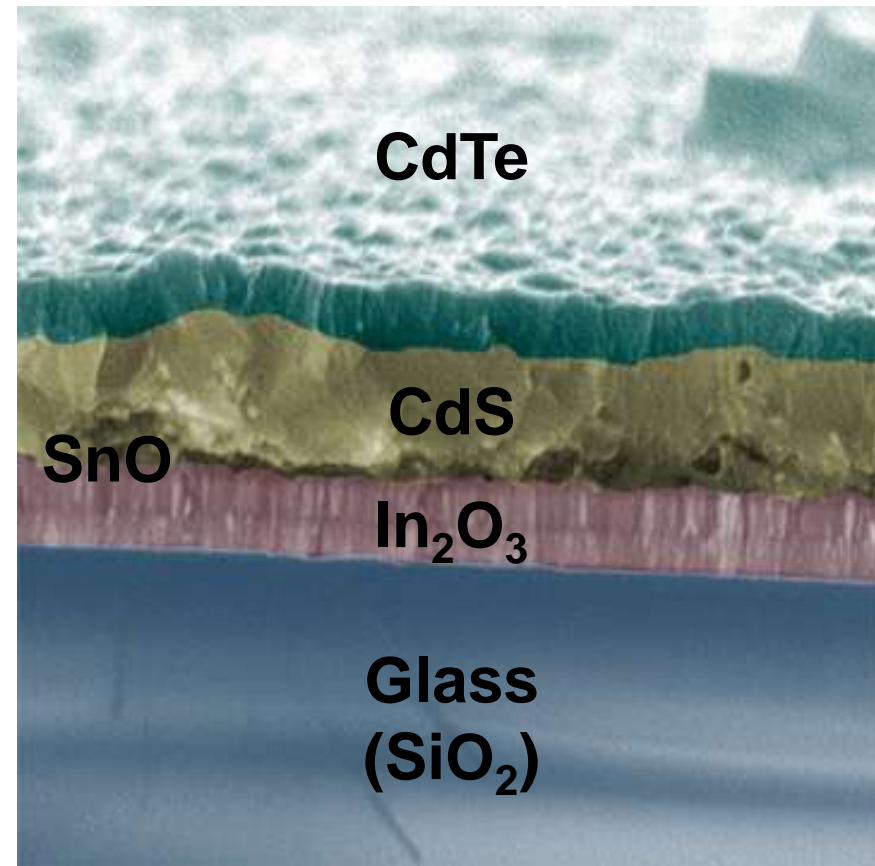
- Cadmium Telluride
  - Cubic (FCC)
- Cadmium Sulfide
  - Cubic (FCC)
  - Hexagonal
- Indium Oxide
  - Cubic (FCC)
- Tin Oxide
  - Tetragonal
- Silicon Oxide
  - Trigonal











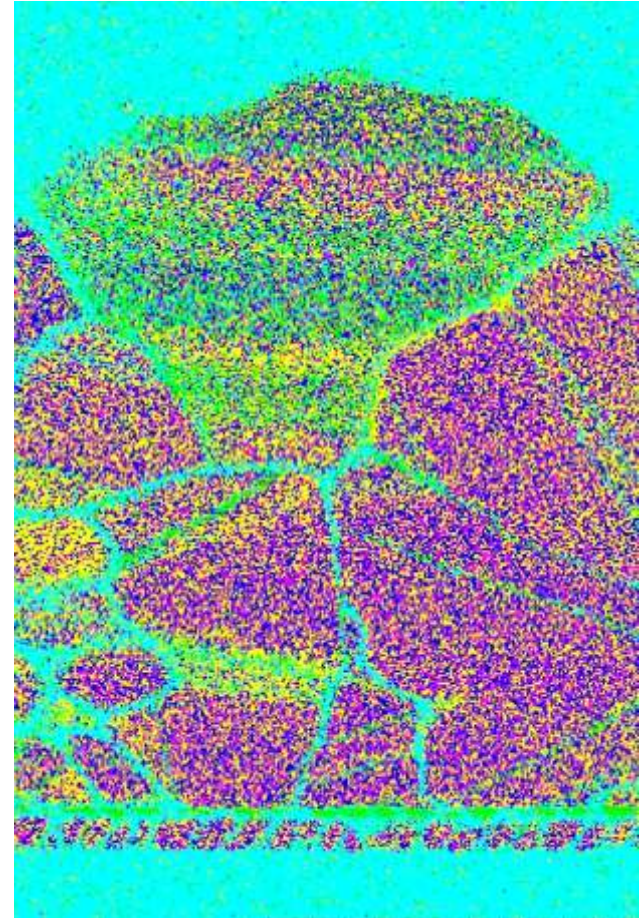
# Photovoltaic Materials

- Cadmium Telluride
  - Cubic (FCC)
- Cadmium Sulfide
  - Cubic (FCC)
  - Hexagonal
- Indium Oxide
  - Cubic (FCC)
- Tin Oxide
  - Tetragonal
- Silicon Oxide
  - Amorphous



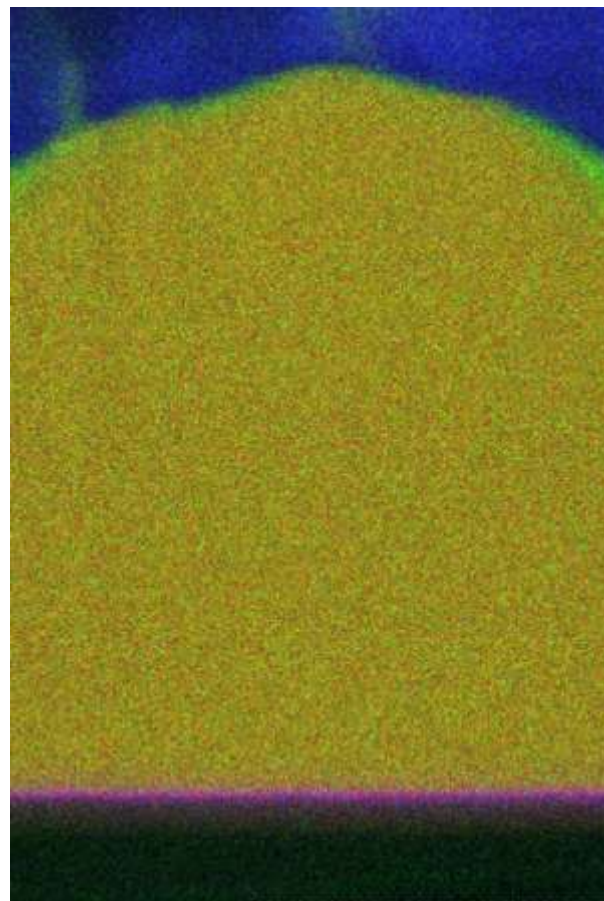
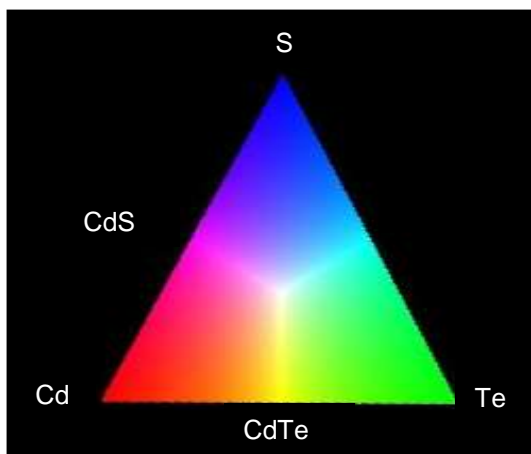
# EBSD Phase Mapping

-  Gold
  - Cubic (FCC)
-  Cadmium Telluride
  - Cubic (FCC)
-  Cadmium Sulfide
  - Cubic (FCC)
  - Hexagonal
-  Indium Oxide
  - Cubic (FCC)
-  Tin Oxide
  - Tetragonal
-  Silicon Oxide
  - Trigonal

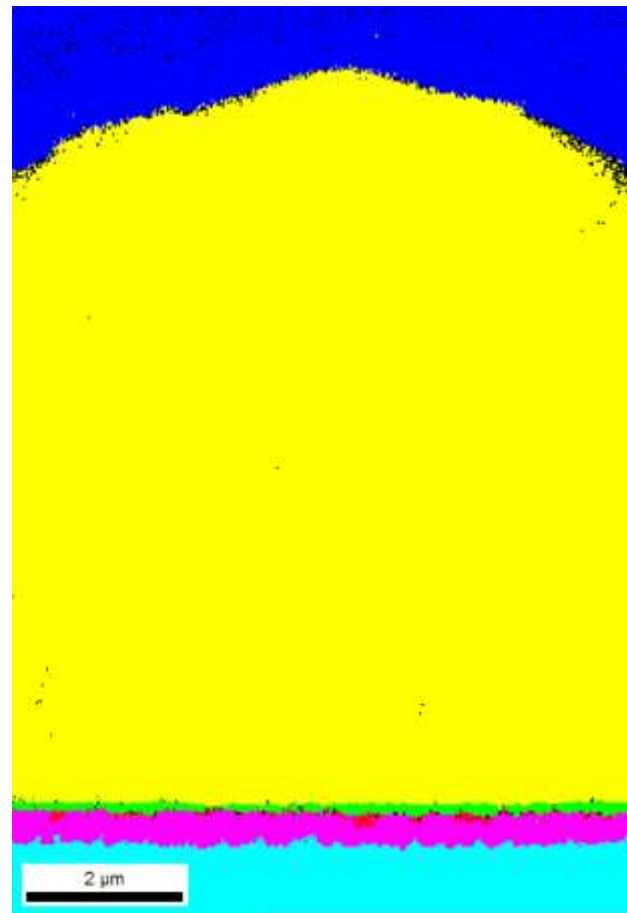
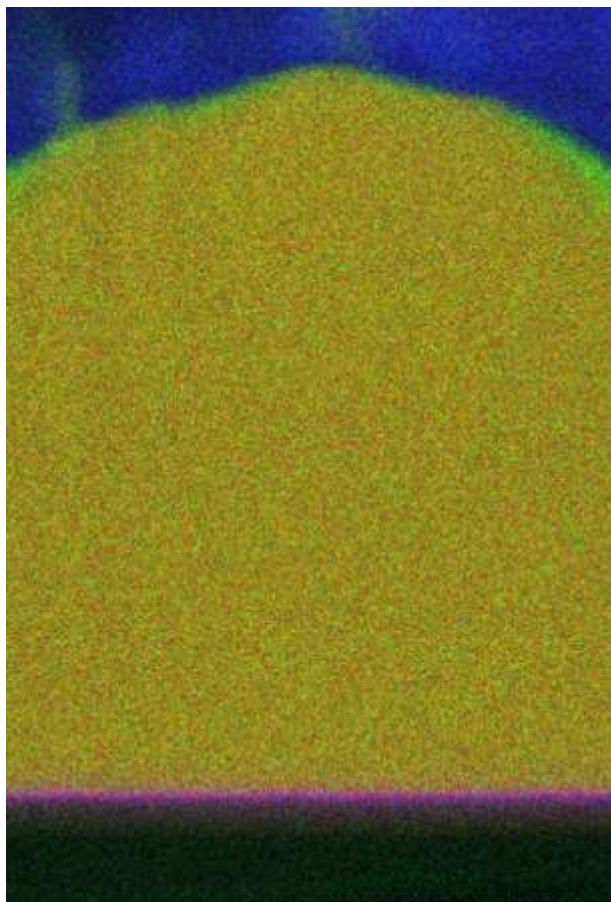




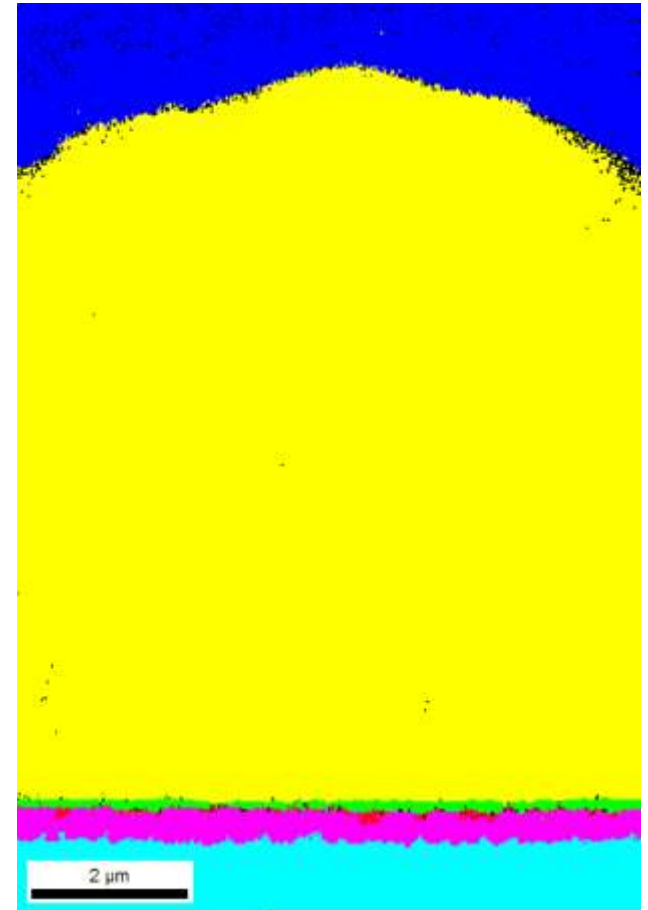
# EDS RGB Element Mapping



# EDS Phase Mapping

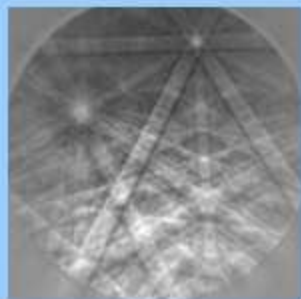
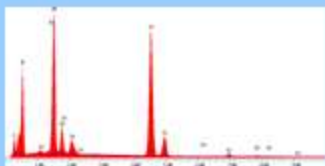


# EDS Phase Mapping

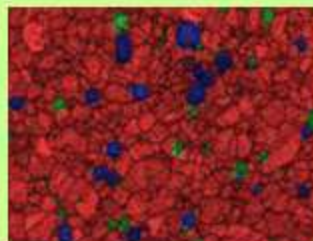


# Chemistry assisted indexing

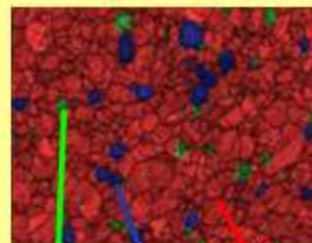
Collect EDS & EBSD data



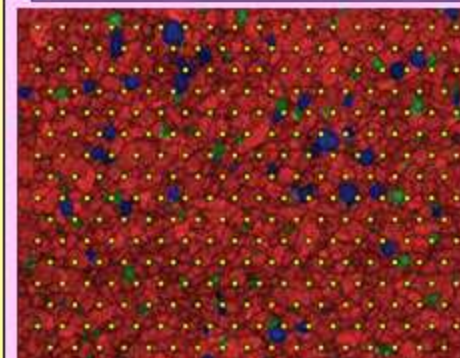
Perform Component Analysis on the EDS data.



Assign each scan point a crystallographic phase(s) according to the component to which it belongs.

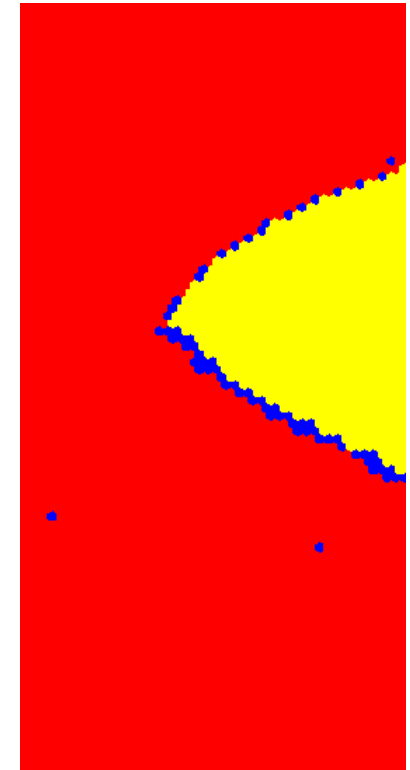
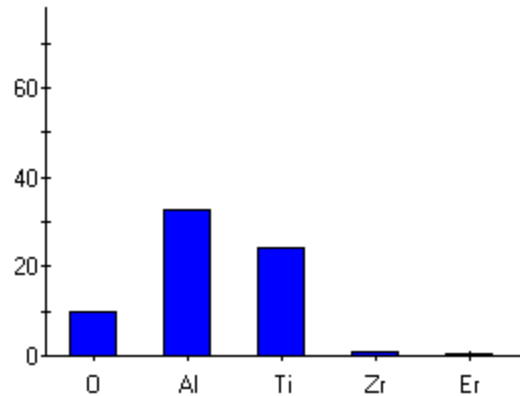
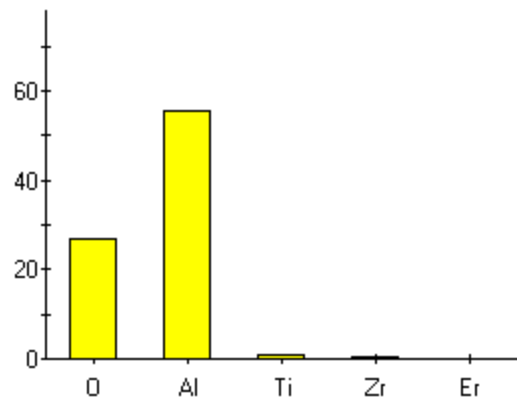
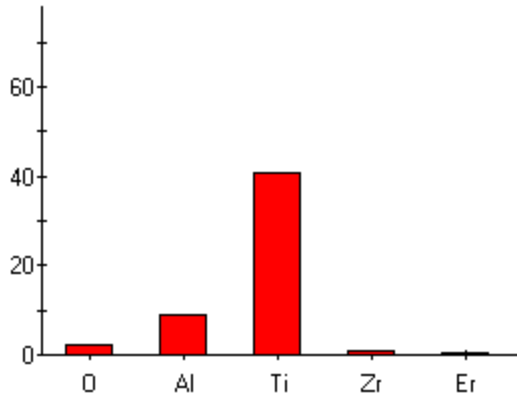


Index EBSPs at each scan point using the phase information assigned to each scan point

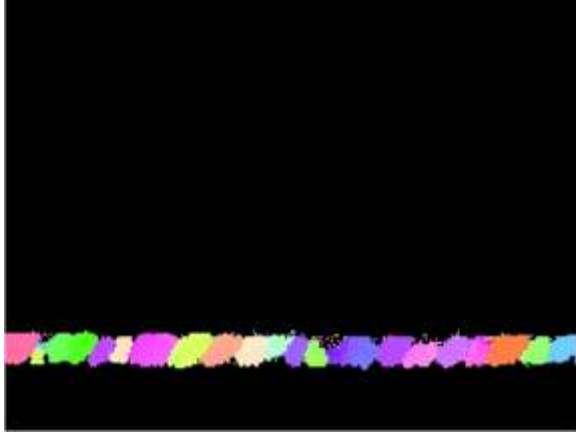
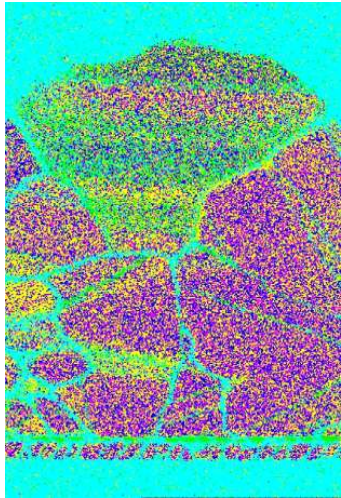


# Chemistry assisted indexing - Notes

- Using the counts of individual elemental ROIs is faster than using full spectral analysis and is effective and efficient on these materials.
- Need to resolve “intermediary” phases
- High-Tilt EDS



# EBSD & EDS phase mapping



$\text{In}_2\text{O}_3$



CdS (Hexagonal)



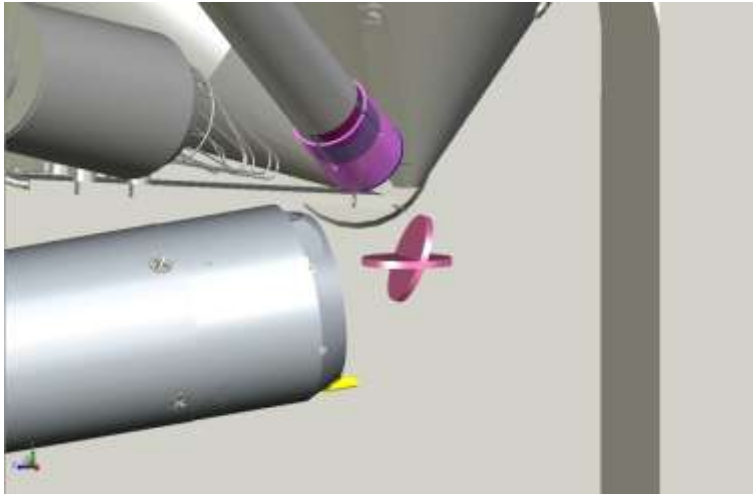
CdTe

# EDS at EBSD Geometry

Azimuthal Angle (°)	Sample Tilt (°)	CPS (Normalized)
NA	0	1.00
0	70	1.44
15	70	1.36
30	70	1.31
45	70	1.22

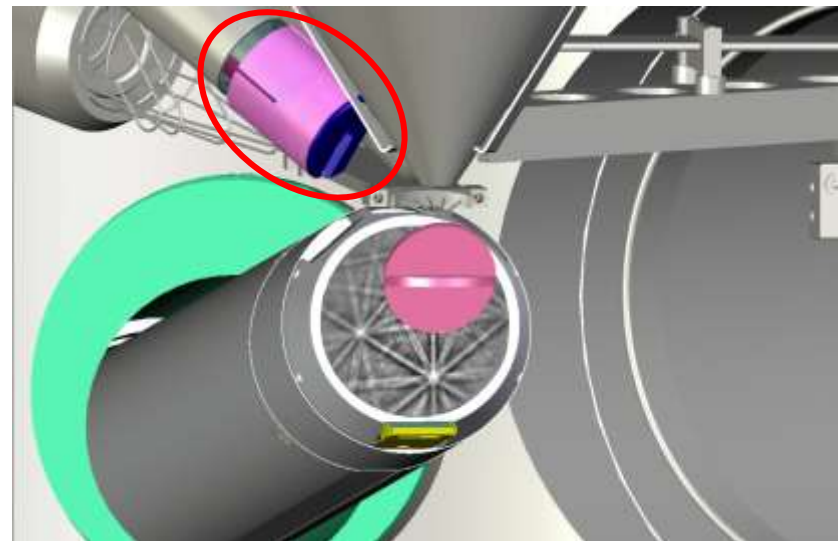
- Tilting the specimen increases the EDS count rate due to reduced absorption
- The improved count rate is often beneficial for simultaneous EDS-EBSD mapping

# Effect of Working Distance on EDS Signal



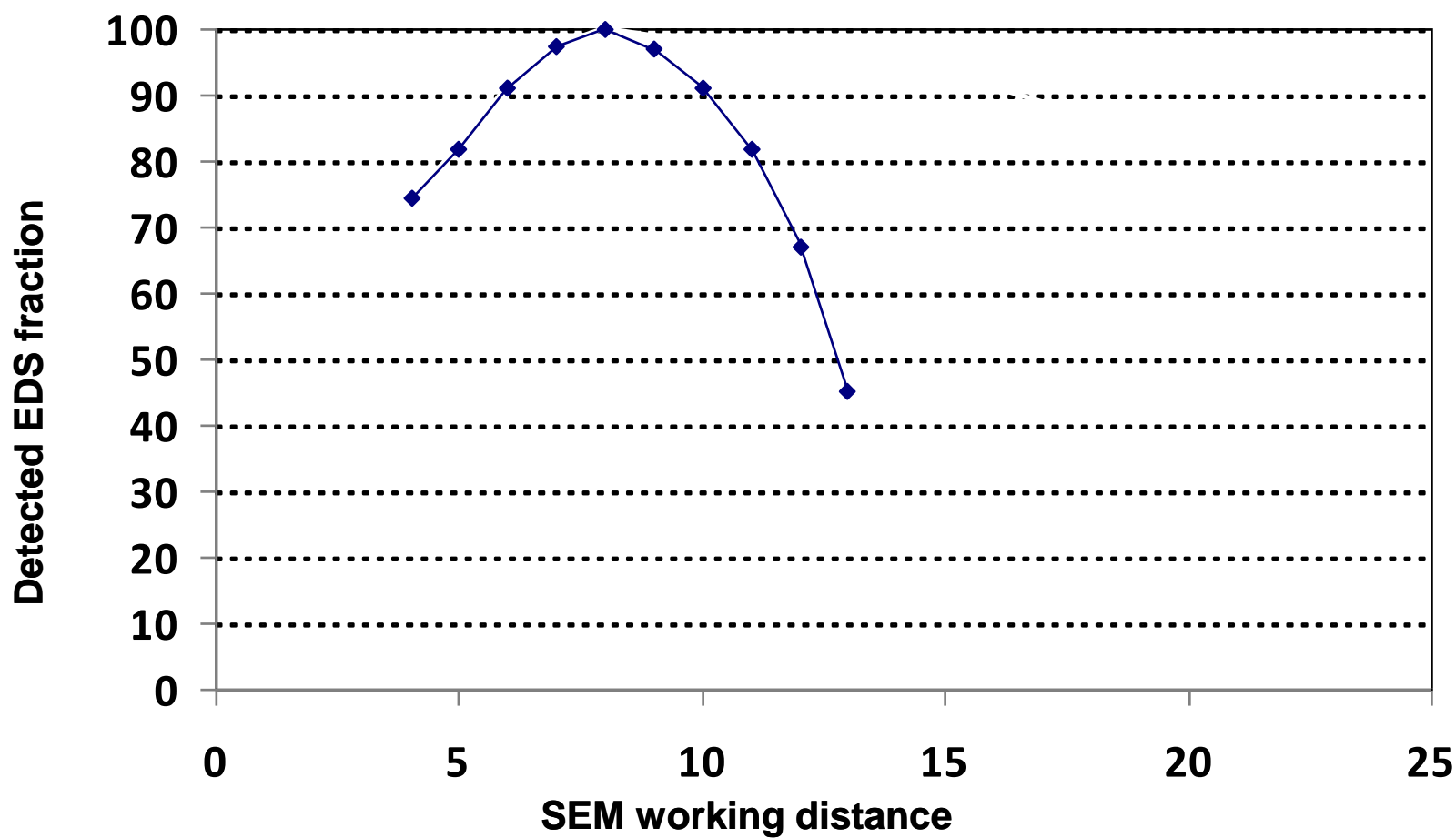
- For simultaneous EDS and EBSD data collection, both detectors need to be aimed at the same working distance.
- EBSD data is often collected away from the optimal EDS analytical working distance depending on the sample size and geometry

- A special EDS electron trap was designed to allow EDS detection over an extended WD range.

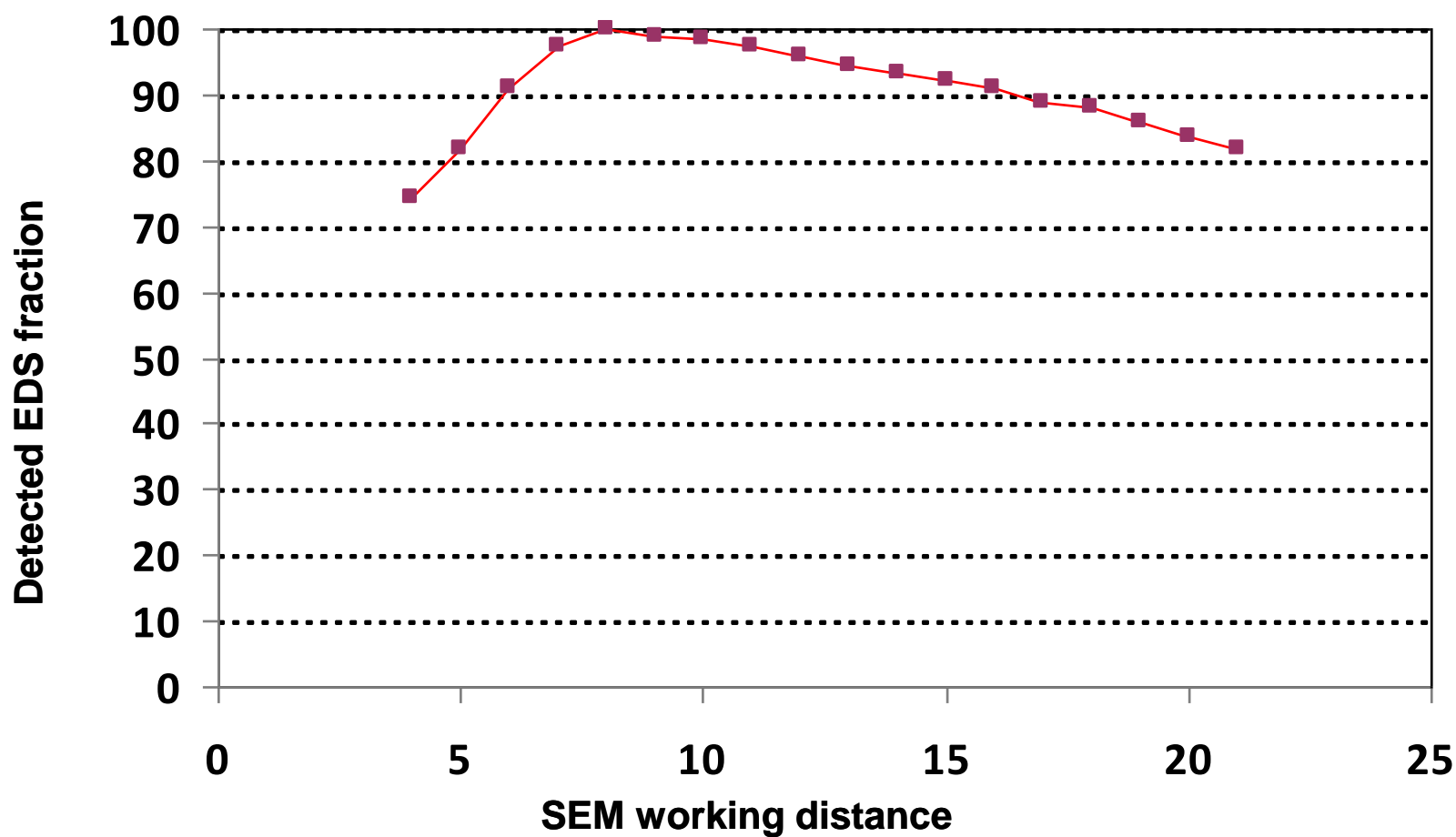




# EDS Detection Improvement



# EDS Detection Improvement

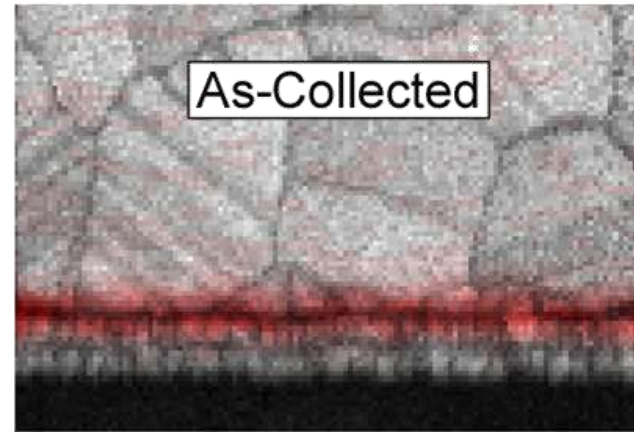
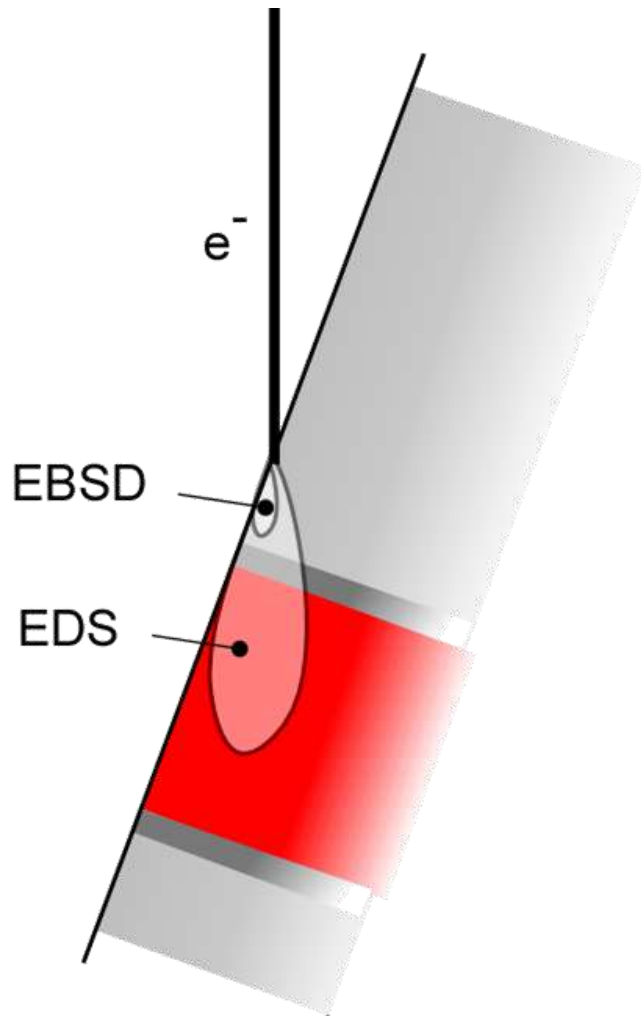


# eZAF Quantification Results

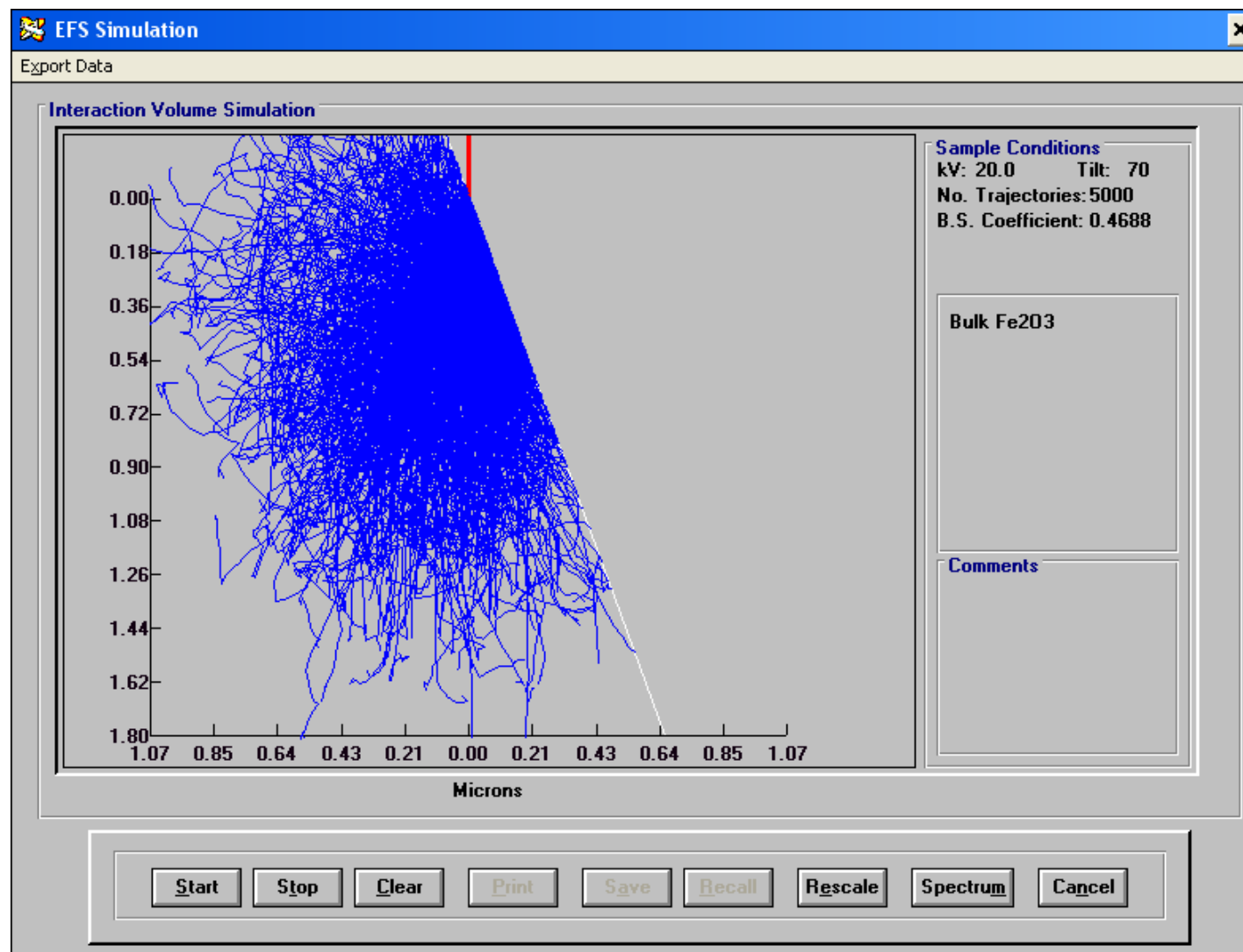
RMS (%) Fe <sub>2</sub> O <sub>3</sub>	0°	15°	30°	45°
ZAF	6.3	6.5	7.6	9.1
eZAF	1.4	1.3	0.3	1.1

- Significant improvement for “difficult” sample compared to old ZAF routine

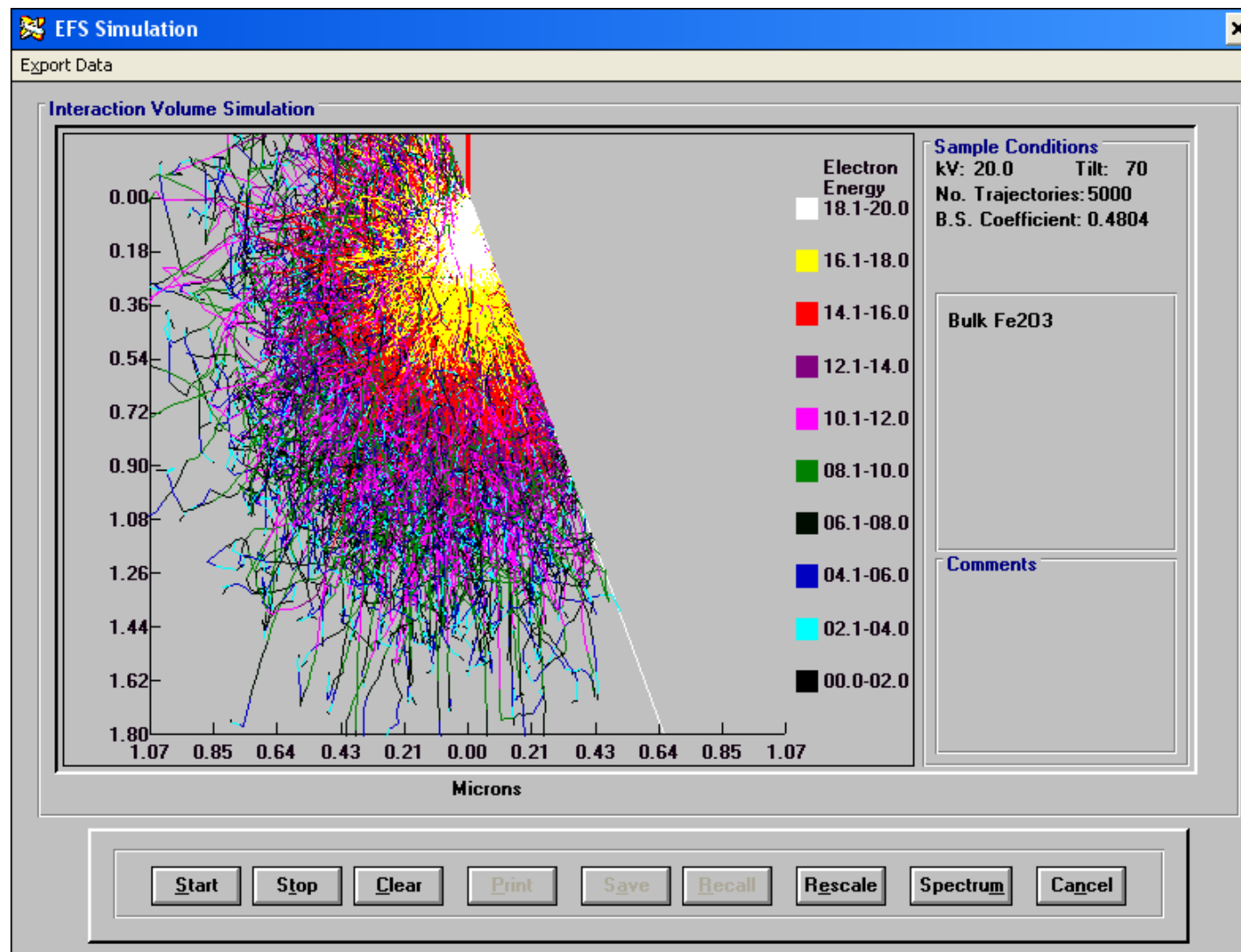
# EBSD and EDS interaction volumes



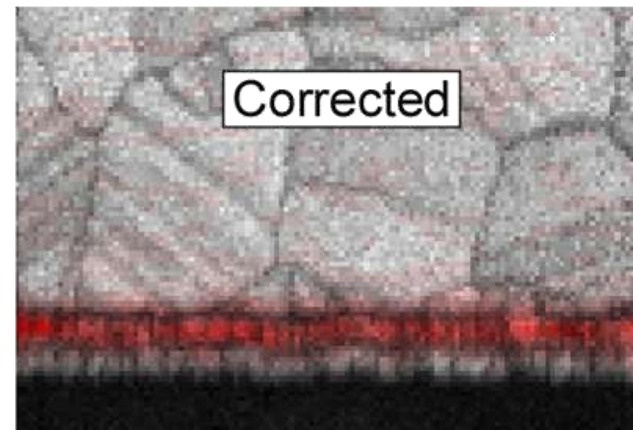
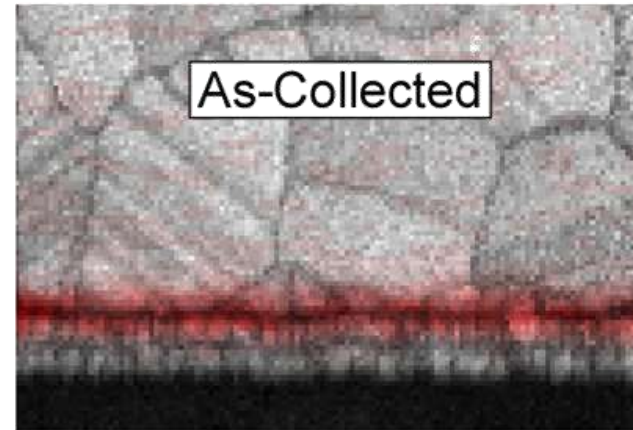
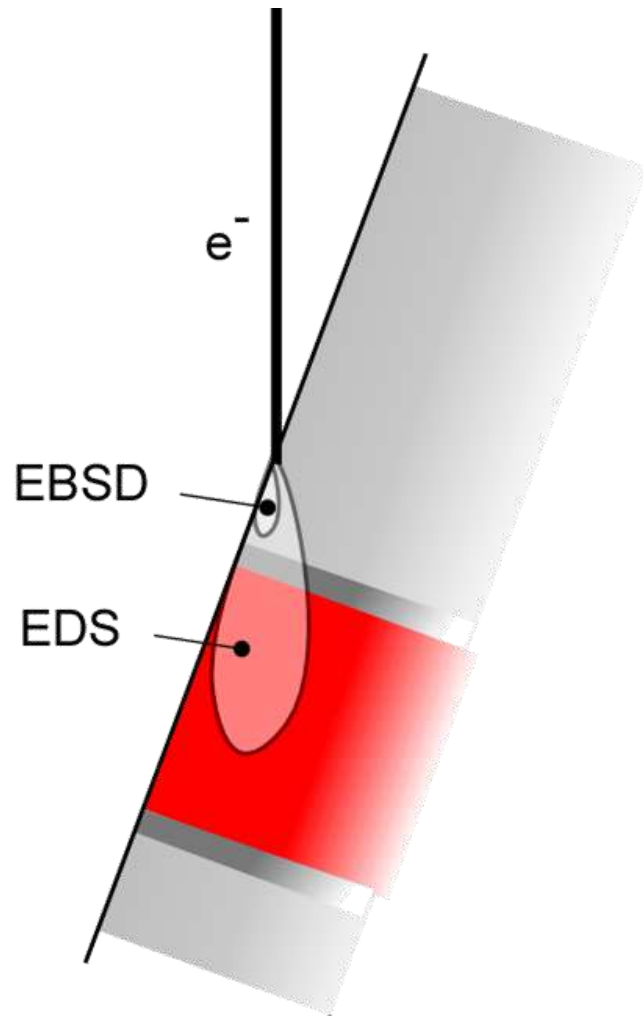
# EDS and EBSD Interaction Volume



# EDS and EBSD Interaction Volume

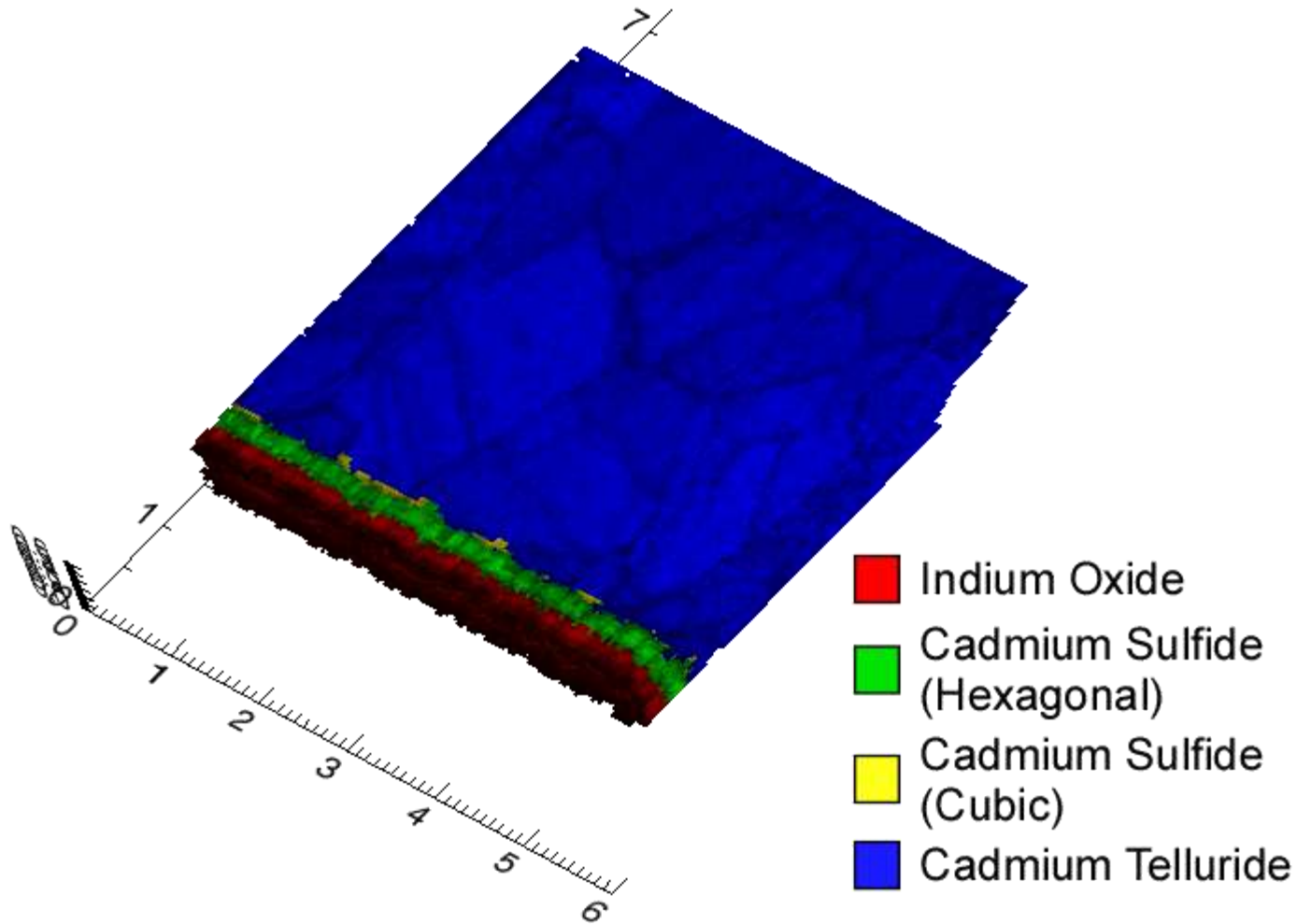


# EBSD and EDS interaction volumes



1  $\mu\text{m}$

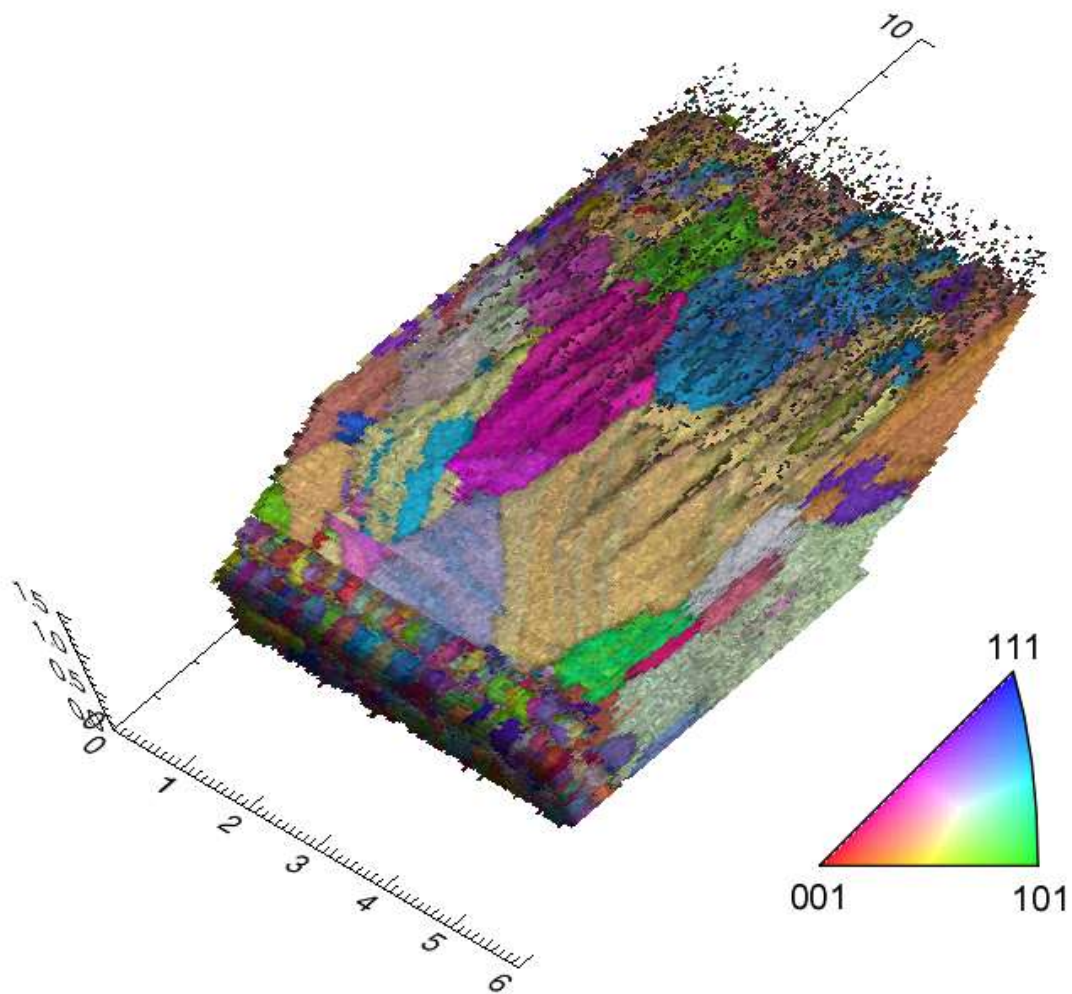
# 3D Reconstruction – Phase map (EDS)



Data collected on a Zeiss Ariga with the help of Hubert Schulz



# 3D Reconstruction – Orientation Map (EDS + EBSD)





Thank you for  
your attention